## Crystallography course SCHEDULE

## Topic

1. Wed 28.02. Lec-1: Introduction
2. Mon 04.03. Lec-2: Crystal Chemistry \& Tolerance parameter
3. Mon 04.03. EXERCISE 1
4. Wed 06.03. Lec-3: Crystal Chemistry \& BVS
5. Fri 08.03. Lec-4: Symmetry \& Point Groups
6. Mon 11.03. EXERCISE 2
7. Wed 20.03. Lec-5: Crystallography \& Space Groups (Linda) Ke3
8. Fri 22.03. Lec-6: XRD \& Reciprocal lattice (Linda) Ke4
9. Mon 25.03. EXERCISE 3 (Linda) Ke4
10. Thu 04.04. Lec-7: Rietveld (Linda) 12:15-14, Ke3
11. Fri 05.04. EXERCISE 4: Rietveld (Linda)

Mon 08.04. EXERCISE 4: Rietveld (Linda)
12. Thu 11.04. Lec-8: ND \& GI-XRD 12:15-14, Ke3
13. Fri 12.04. Lec-9: XRR (Topias)
14. Mon 15.04. EXERCISE 5: XRR (Topias)

Wed 17.04. EXERCISE 5: XRR (Topias)
15. Mon 22.04. Lec-10: Synchrotron radiation \& XAS \& EXAFS
16. Thu 25.04. Mössbauer 12:15-14, Ke3
17. Fri 26.04. EXERCISE 6
18. Mon 29.04. Seminars:
19. Fri 03.05. Seminars:
20. Mon 06.05. ADDITIONAL DISCUSSION/QUESTION POSSIBILITY

## LINDA'S LECTURES \& EXERCISES

- 20.03. WEDNESDAY: 14.15-16 (Ke3)
- 22.03. FRIDAY: 12:15-14 (Ke3)
- 25.03. MONDAY: 14.15-16 (Ke4) Exercise session
- Exercise deadline 27.3 at 13:00


## Easter holidays break! ()

- 04.04. THURSDAY: 12:15-14 (Ke3)
- 05.04. FRIDAY: 12:15-14 (?) Rietveld exercise 1
- Getting started on FullProf
- 08.04. MONDAY: 14:30-16 (?) Rietveld exercise 2
- Help with finishing the exercise
- Exercise deadline on 10.4. at 13:00


## SEMINARS

- IR
- Raman
- XPS
- SEM
- AFM
- HRTEM
- ED
- EELS Miklos Nemesszeghy


## INSTRUCTIONS for SEMINAR PRESENTATIONS

- Topics: IR, Raman, XPS, SEM, AFM, HRTEM, ED, EELS
- Seminar presentation is mandatory
- Presentation slides will be put up in MyCourses afterwards
- Seminars are part of the course content and it is likely that there will be questions in the exam related to these seminars
- Given independently or in a group of two students
- Evaluated in the scale: 10 ~ 20 points
- Presentation: 25+5 minutes
- Rough content of the presentation:
- principle of the technique(s)
- type of information gained
- interpretation of the measured data
- pros \& cons
- two to four research examples
(you will be given some relevant research papers for an example)



## LECTURE 5: CRYSTALLOGRAPHY BASICS

- From "point-like" molecules to 3D crystals
- Translation in 3D crystals $\rightarrow$ NEW SYMMETRY OPERATIONS (glide planes \& screw axes)
- From Point groups to Space groups
- Crystal lattice, lattice points \& unit cell
- International Tables of Crystallography


## Symmetry elements (Schönflies / Hermann-Mauguin)

## RECALL FROM PREVIOUS LECTURE:

Point / Molecular symmetry $\square$ One point remains unchanged

Identity $E$


Symmetry/inversion center il


Rotation axis $C_{n} / 1,2,3, \ldots$


Reflection/mirror plane $\sigma / m$


Improper rotation axis $S_{n} /(\overline{1}, \overline{2}), \overline{3}, \overline{4}, \overline{6}$


NOTE:

- Improper rotation axis $\overline{1}$ and inversion center $\overline{1}$ are equivalent
- Improper rotation axis 2 and mirror plane $m$ are equivalent


## CRYSTAL 2D

## FROM MOLECULES TO CRYSTALS

There are two things which make macroscopic (infinite) crystals different from discrete molecules in terms of symmetry: Translation \& Space-filling

Translation in crystals

- Translation: move from one point to another (the entire object)
- This does not exist in molecules, but is the essence of macroscopic crystals exhibiting long-range order

- Crystal lattice: regular, infinite pattern


## FROM MOLECULES TO CRYSTALS

There are two things which make macroscopic (infinite) crystals different from discrete molecules in terms of symmetry: Translation \& Space-filling

## Space-filling $\rightarrow 5$ lattices

- Macroscopic crystals need to continuosly fill the space
- For molecules 5 -fold rotation is possible, but not for crystals
- Quasicrystals can have 5-fold rotation: 2-component lattice



## FROM MOLECULES TO CRYSTALS

The crystal lattice is the geometrical 'drawing board', which is then filled with constituents that build a pattern: lattice points, basis, and unit cell

## Basis:

- The atom (atom group) that is repeated


## Lattice points

- The positions filled by the basis as a function of the symmetry operations for the lattice
- Each lattice point has identical environment + symmetry properties (=point group)



## FROM MOLECULES TO CRYSTALS

The crystal lattice is the geometrical 'drawing board', which is then filled with constituents that build a pattern: lattice points, basis, and unit cell

## Unit cell

- Smallest possible part of the crystal lattice;
> that repeats itself periodically;
> to completely fills the lattice volume;
> and is enough to describe the entire lattice perfectly

- Choice of the unit cell not always unambiguous: several options


## 2D $\rightarrow$ 3D

## LATTICES IN 3D: CRYSTAL SYSTEMS (7)

No information on the positions of atoms


Cubic
$\mathrm{a}=\mathrm{b}=\mathrm{c}$
$\alpha=\beta=\gamma=90^{\circ}$
$\mathrm{NaCl}, \mathrm{MgAl}_{2} \mathrm{O}_{4}$


Tetragonal
$\mathrm{a}=\mathrm{b} \# \mathrm{c}$
$\alpha=\beta=\gamma=90^{\circ}$ $\mathrm{TiO}_{2}, \mathrm{~K}_{2} \mathrm{NiF}_{4}$


Orthorhombic a \# b \# c
$\alpha=\beta=\gamma=90^{\circ}$ $\mathrm{YBa}_{2} \mathrm{Cu}_{3} \mathrm{O}_{7}$


Rhombohedral
$\mathrm{a}=\mathrm{b}=\mathrm{c}$
$\alpha=\beta=\gamma \# 90^{\circ}$
$\mathrm{BaTiO}_{3}$ (low-T)

Hexagonal

$\mathrm{a}=\mathrm{b} \# \mathrm{c}$
$\alpha=\beta=90, \gamma=120^{\circ}$ $\mathrm{LiNbO}_{3}$


Monoclinic

$$
\mathrm{a} \# \mathrm{~b} \# \mathrm{c}
$$

$$
\alpha=\gamma=90^{\circ}, \beta \# 90^{\circ}
$$

$$
\mathrm{KH}_{2} \mathrm{PO}_{4}
$$



Triclinic
$\mathrm{a} \# \mathrm{~b} \# \mathrm{c}$
$\alpha \# \beta \# \gamma \# 90^{\circ}$

## Each LATTICE POINT should be able to be

 described with the vector $Q_{[u v w]}=u a+v b+w c$, where $u, v$ and $w$ all are INTEGER values

Unit cell: YES


Unit cell: NO


Lattice points = unit cell corners

What we need to tell about the UNIT CELL
(basis)

- Shape \& size of the unit cell plus the atomic positions in the unit cell
- Shape \& size are given by:
- Lattice parameters: a, b ja c
- Angles between the axes: $\alpha, \beta$ ja $\gamma$


## LATTICES IN 3D: BRAVAIS LATTICES (14)

Basic stacking of lattice sites included (basis, not atoms)


| Centering | Lattice <br> sites/cell | Abbre- <br> viation |
| :--- | :--- | :--- |
| Primitive | 1 | P |
| Base (A, B, or C) <br> centered | 2 | A, B, C |
| Body centered | 2 | I |
| Hexagonal, <br> rhombohedral | 3 | h / R |
| Face centered | 4 | F |

## EXAMPLE

- What is the Bravais lattice type of NaCl : Cubic F (basis: $\mathrm{Na}-\mathrm{Cl}$ )


## Rock-salt ( NaCl ) structure



## Your EXERCISE question

- What is the Bravais lattice type of CsC



## Counting atoms

## NUMBER of FORMULA UNITS in UNIT CELL (Z)

= basis sets



Simple cubic

> Typically $Z=1-6$, but can be tens or even hundreds!


## EXAMPLE

- How many NaCl formula units in unit cell?
- 

answer...


- $\mathrm{Cl}^{-}$
- $\mathrm{Na}^{+}$

| If the atom is placed... | ...it belongs to: |
| ---: | ---: |
| inside unit cell | one unit cell |
| on unit cell face | two unit cells |
| unit cell edge | four unit cells |
| unit cell corner | eight unit cells |

# Lattice symmetry: POINT GROUPS 

## The 2D point groups

Crystallography fundamental rule of translation: units must stack without gaps!


## The 2D point groups

Crystallography fundamental rule of translation: units must stack without gaps!

5 options for rotational symmetry

| + | 1 | 2 | 3 | 4 |
| :---: | :---: | :---: | :---: | :---: |
| 5 more options with <br> mirror symmetry |  |  |  |  |
| $=10$ point groups | $m(1 \mathrm{~m})$ | 2 mm | 3 m | 4 mm |

Point group graphical symbol shown in red

## New symmetry in 3D

## Additional translation symmetry elements in INFINITE LATTICES

Combining translation with other symmetry operations/elements
$\rightarrow$ new symmetry operations/elements: glide planes \& screw axes

## Screw axis

- Rotation plus translation
$n_{m}\left(2_{1}, 3_{1}, 3_{2}, 4_{1}, 4_{2}, 4_{3}, 6_{1}, 6_{2}, 6_{3}, 6_{4}, 6_{5}\right)$
For example: $2_{1}$ : rotation $180^{\circ}$ and translation $1 / 2(\mathrm{~m} / \mathrm{n})$


## Glide plane

- Reflection against a mirror plane plus (half) translation parallel to the plane

Axis glide plane: $\quad a, b, c$ (translations by $1 / 2 a, 1 / 2 b, 1 / 2 c$ to each glide plane direction)
Diagonal glide plane:

$$
\begin{aligned}
& n[1 / 2(a+b), 1 / 2(b+c), 1 / 2(c+a)] \\
& d[1 / 4(a+b), 1 / 4(b+c), 1 / 4(c+a)] \text { (so-called diamond glide plane) }
\end{aligned}
$$

## GLIDE PLANE c ( $\perp \mathrm{b}$ )

Reflection (m) through ac-plane, followed by (half) translation (t) along c-axis


## Symmetry elements in 3D

| Element | What it does | Possible in crystal system |
| :---: | :---: | :---: |
| Identity $(1)$ | - | All |
| Inversion $(\overline{1})$ | Inversion | All |
| Mirror plane $(\mathrm{m})$ | Mirror | All but triclinic |
| 2- fold rotation $(2)$ | Rotate $180^{\circ}$ | All but triclinic |
| 3-fold rotation $(3)$ | Rotate $120^{\circ}$ | Trigonal, Hexagonal and Cubic |
| 4-fold Rotation (4) | Rotate $90^{\circ}$ | Tetragonal and Cubic |
| 6-fold Rotation $(6)$ | Rotate $60^{\circ}$ | Hexagonal |


| $\boldsymbol{m}$ | Mirror plane |
| :---: | :--- |
| $\mathbf{2}$ | 2-fold rotation |
| $\mathbf{3}$ | 3-fold rotation |
| 4 | 4-fold rotation |
| $\mathbf{6}$ | 6-fold rotation |

- $\overline{\mathbf{1}} \quad$ 1-fold improper rotation = inversion point
3-fold improper rotation
(1) $\overline{4}$ 4-fold improper rotation
(-) $\overline{6}$ 6-fold improper rotation


## POINT GROUPS <br> (from historical reasons) DIFFERENT SYMMETRY SYMBOLS

Schoenflies (S) symbols

- were developed first
- in molecular symmetry \& spectroscopy

Hermann-Mauguin (H/M) symbols

- in crystallography
- long and short forms


## Graphical symbols



3D point group graphical symbols
Table 1.1 Symmetry elements of crystal point groups.

| System | Point group symmetry |  | Symmetry elements | Numberofopera-tions |
| :---: | :---: | :---: | :---: | :---: |
|  | $S$ | $H / M$ |  |  |
| Triclinic | $\begin{aligned} & C_{1} \\ & C_{i}=S_{2} \end{aligned}$ | $\frac{1}{1}$ | $\begin{aligned} & I=C_{1} \\ & I, i\left(=S_{2}\right) \end{aligned}$ | $\begin{aligned} & 1 \\ & 2 \end{aligned}$ |
| Monoclinic | $\begin{aligned} & C_{2} \\ & C_{s}=C_{1 \mathrm{~h}}=C_{1 v} \\ & C_{2 \mathrm{k}} \end{aligned}$ | $\begin{aligned} & 2 \\ & m \\ & 2 / m \end{aligned}$ | $\begin{aligned} & I, C_{2} \\ & I, \sigma \\ & I, C_{2}, \sigma_{n}, i \end{aligned}$ | $\begin{aligned} & 2 \\ & 2 \end{aligned}$ |
| Orthorhombic | $\begin{aligned} & C_{21} \\ & D_{2} \\ & D_{2 n} \end{aligned}$ | $\begin{aligned} & \mathrm{mm} 2 \\ & 222 \\ & \mathrm{mmm} \end{aligned}$ | $\begin{aligned} & I, C_{2}, 2 \sigma \\ & I, 3 C_{2} \\ & I, 3 C_{2}, 3 \sigma, i \end{aligned}$ | $\begin{aligned} & 4 \\ & 4 \\ & 8 \end{aligned}$ |
| Tetragonal | $\begin{aligned} & C_{4} \\ & S_{4} \\ & D_{4} \\ & C_{4 v} \\ & C_{4 k} \\ & D_{24} \\ & D_{4 k} \end{aligned}$ | 4 <br> 422 <br> 4 mm <br> 4/m <br> $\overline{4} 2 \mathrm{~m}$ <br> $4 / \mathrm{mmm}$ | I, $C_{4}$ <br> I, $S_{4}\left(=C_{2}\right)$ <br> I, $C_{4}\left(-C_{2}\right), 2 C_{2}^{\prime}, 2 C_{2}^{B}$ <br> $I, C_{4}, 2 \sigma_{b}, 2 \sigma_{6}$ <br> $I, C_{4}\left(=S_{4}\right), \sigma_{k}, i$ <br> $I, S_{4}\left(=C_{2}\right), 2 C_{2}^{\prime}, 2 \sigma_{4}$ <br> $I, C_{4}\left(=S_{4}\right), 2 C_{2}^{\prime}, 2 C_{2}^{\prime \prime}$, | $\begin{aligned} & 4 \\ & 4 \\ & 8 \\ & 8 \\ & 8 \\ & 8 \end{aligned}$ |
| Trigonal | $\begin{aligned} & C_{3} \\ & C_{34} \\ & D_{3} \\ & C_{3 \mathrm{~s}} \\ & D_{3 \mathrm{~d}} \end{aligned}$ | $\begin{aligned} & \frac{3}{3} \\ & 32 \\ & \frac{3}{3} m \\ & \frac{3}{3} m \end{aligned}$ | I, $C_{3}$ <br> I, $S_{6}\left(=C_{3}\right), i$ <br> I, $C_{3}, 3 C_{2}$ <br> $I, C_{3}, 3 \sigma_{0}$ <br> $I, S_{6}\left(-C_{3}\right), 3 C_{2}, 3 \sigma_{\mathrm{d}}, i$ | $\begin{array}{r} 3 \\ 6 \\ 6 \\ 6 \\ 12 \end{array}$ |
| Hexagonal | $\begin{aligned} & C_{6} \\ & C_{3 h} \\ & D_{6} \\ & D_{3 k} \\ & C_{6 h} \\ & C_{64} \\ & D_{6 k} \end{aligned}$ | 6 <br> 6 <br> 622 <br> 6 m 2 <br> $6 / m$ <br> 6 mm <br> $6 / \mathrm{mmm}$ | I, $C_{6}$ <br> I, $S_{3}\left(=C_{3}\right), \sigma_{4}$ <br> I, $C_{6}, 3 C_{2}^{\prime}, 3 C_{2}^{\prime}$ <br> $1, C_{3}\left(=S_{3}\right), 3 C_{2}, 3 \sigma_{6}, \sigma_{4}$ <br> $I, C_{6}\left(=S_{6}\right), \sigma_{k}, i$ <br> I, $C_{6}, 3 \sigma_{0}, 3 \sigma_{d}$ <br> $I, C_{6}\left(=S_{6}\right), 3 C_{2}^{\prime}, 3 C_{2}^{*}$, <br> $3 \sigma_{v}, 3 \sigma_{d}, \sigma_{k}, i$ | $\begin{array}{r} 6 \\ 6 \\ 12 \\ 12 \\ 12 \\ 12 \\ 24 \end{array}$ |
| Cubic | $\begin{aligned} & T \\ & T_{\mathrm{b}} \\ & T_{d} \\ & O \\ & O_{\mathrm{b}} \end{aligned}$ | $\begin{aligned} & 23 \\ & m 3 \\ & \frac{m 3 m}{43 m} \\ & 432 \\ & m 3 m \end{aligned}$ | I, $3 C_{2}, 4 C_{3}$ <br> $I, 3 C_{2}, 4 C_{3}\left(=S_{6}\right), 3 \sigma_{6}, i$ <br> $I, 3 C_{2}\left(=S_{4}\right), 4 C_{3}, 6 \sigma_{d}$ <br> $I, 3 C_{2}, 4 C_{3}, 3 C_{4}$ <br> $I, 3 C_{2}, 4 C_{3}\left(=S_{6}\right)$, <br> $3 C_{4}\left(=S_{4}\right), 3 \sigma_{\mathrm{h}}, 6 \sigma_{\mathrm{d}}, i$ | $\begin{aligned} & 12 \\ & 24 \\ & 24 \\ & 24 \\ & 48 \end{aligned}$ |

## Hermann-Mauguin (H/M) symbols

## CRYSTAL CLASSES <br> i.e. POINT GROUPS

The 32 Point Groups

| 1 | 4 | $\overline{3}$ | 6 mm |
| :---: | :---: | :---: | :---: |
| $\overline{1}$ | $\overline{4}$ | 32 | $\overline{6} m 2$ |
| 2 | $4 / \mathrm{m}$ | 3 m | $6 / \mathrm{mmm}$ |
| m | 422 | $\overline{3} m$ | 23 |
| $2 / \mathrm{m}$ | 4 mm | 6 | $m \overline{3}$ |
| 222 | $\overline{4} 2 m$ | $\overline{6}$ | 432 |
| mm 2 | $4 / \mathrm{mmm}$ | $6 / \mathrm{m}$ | $\overline{4} 3 m$ |
| mmm | 3 | 622 | $m \overline{3} m$ |

The 32 Point Groups (Schoenflies)

| $1\left(\mathrm{C}_{1}\right)$ | $4\left(\mathrm{C}_{4}\right)$ | $\overline{3}\left(\mathrm{C}_{3 \mathrm{i}}\right)$ | $6 \mathrm{~mm}\left(\mathrm{C}_{6 \sigma \mathrm{v}}\right)$ |
| :---: | :---: | :---: | :---: |
| $\overline{1}\left(\mathrm{C}_{\mathrm{i}}=\mathrm{S}_{2}\right)$ | $\overline{4}\left(\mathrm{~S}_{4}\right)$ | $32\left(\mathrm{D}_{3}\right)$ | $\overline{6} m 2\left(\mathrm{D}_{3 \sigma \mathrm{~h}}\right)$ |
| $2\left(\mathrm{C}_{2}\right)$ | $4 / \mathrm{m}\left(\mathrm{C}_{4 \sigma \mathrm{~h}}\right)$ | $3 \mathrm{~m}\left(\mathrm{C}_{3 \sigma \mathrm{v}}\right)$ | $6 / \mathrm{mmm}\left(\mathrm{D}_{6 \sigma \mathrm{~h}}\right)$ |
| $\mathrm{m}\left(\mathrm{C}_{\sigma}\right)$ | $422\left(\mathrm{D}_{4}\right)$ | $\overline{3} m\left(\mathrm{D}_{3 \mathrm{~d}}\right)$ | $23(\mathrm{~T})$ |
| $2 / \mathrm{m}\left(\mathrm{C}_{2 \sigma \mathrm{~h}}\right)$ | $4 \mathrm{~mm}\left(\mathrm{C}_{4 \sigma \mathrm{v}}\right)$ | $6\left(\mathrm{C}_{6}\right)$ | $m \overline{3}\left(\mathrm{~T}_{\mathrm{h}}\right)$ |
| $222\left(\mathrm{D}_{2}\right)$ | $\overline{4} 2 m\left(\mathrm{D}_{2 \mathrm{~d}}\right)$ | $\overline{6}\left(\mathrm{C}_{3 \sigma \mathrm{~h}}\right)$ | $432(\mathrm{O})$ |
| $2 \mathrm{~mm}\left(\mathrm{C}_{2 \sigma \mathrm{v}}\right)$ | $4 / \mathrm{mmm}\left(\mathrm{D}_{4 \mathrm{~h}}\right)$ | $6 / \mathrm{m}\left(\mathrm{C}_{6 \sigma \mathrm{~h}}\right)$ | $\overline{4} 3 m\left(\mathrm{~T}_{\mathrm{d}}\right)$ |
| $\mathrm{mmm}\left(\mathrm{D}_{2 \sigma \mathrm{~h}}\right)$ | $3\left(\mathrm{C}_{3}\right)$ | $622\left(\mathrm{D}_{6}\right)$ | $m \overline{3} m\left(\mathrm{O}_{\mathrm{h}}\right)$ |

## Which point groups are possible for each lattice type?

| System | Minimum Requirements |
| :---: | :---: |
| Cubic | Four 3-fold rotation axis |
| Tetragonal | One 4-fold rotation (or RI) axis |
| Orthorhombic | Three perpendicular 2-fold axis |
| Rhombohedral | One 3-fold rotation (or RI) axis |
| Hexagonal | One 6 fold rotation (or RI) axis |
| Monoclinic | One 2 fold rotation axis or mirror plane |
| Triclinic | none |
| System | Point groups |
| Cubic | 23, $m \overline{3}, 432, \overline{4} 3 m, m \overline{3} m$ |
| Tetragonal | $4, \overline{4}, 4 / \mathrm{m}, 422,4 \mathrm{~mm}, \overline{4} 2 \mathrm{~m}, 4 / \mathrm{mmm}$ |
| Orthorhombic | 222, 2 mm , mmm |
| Trigonal | $3, \overline{3}, 32,3 \mathrm{~m}, \overline{3} \mathrm{~m}$ |
| Hexagonal | $6, \overline{6}, 6 / \mathrm{m}, 622,6 \mathrm{~mm}, \overline{6} \mathrm{~m} 2,6 / \mathrm{mmm}$ |
| Monoclinic | 2, m, 2/m |
| Triclinic | 1, $\overline{1}$ |

## The building blocks of 3D

## 7 Crystal systems

= geometrical bodies (unit cells) that can stack in 3D
\& 14 Bravais lattices
= basic atomic arrangements within a unit cell
\& 32 Point groups
= geometrical symmetry operation systems within the unit cell
$\rightarrow 230$ Space groups
= possible combinations of lattices and symmetry elements (impossible systems and doublets excluded)

## $\rightarrow$ SPACE GROUPS (230)

## Triclinic

(For the enlarged unit cells, click here)

| 1. $\underline{P 1}$ | 2. $\underline{P-1}$ |  |  |  |
| :---: | :---: | :---: | :---: | :---: |
|  | Monoclinic |  |  |  |
| (For a fuller list with alternative unique axes, origins, or enlarged unit cells click here) |  |  |  |  |
| 3. P121 | 4. P12 $1_{1} \underline{1}$ | 5. C121 | 6. $\underline{P 1 m 1}$ | 7. P1c1 |
| 8. C 1 ml | 9. $C 1 c 1$ | 10. $P 12 / \mathrm{ml}$ | 11. $\mathrm{P}_{12} 1_{1 / \mathrm{m} 1}$ | 12. $\mathrm{C} 12 / \mathrm{ml}$ |
| 13. P12/c1 | 14. $\underline{12}_{1} / \mathrm{Cc}$ | 15. C12/c1 |  |  |
| Orthorhombic |  |  |  |  |
| (For a fuller list with alternative axes and origins click here) |  |  |  |  |
| 16. $\underline{P 222}$ | 17. $\underline{P 2221}_{1}$ |  | 19. $\underline{2}_{1}^{1} \underline{2}_{1} \underline{2}_{1}$ | 20. $\underline{C 2221}$ |
| 21.C222 | 22.F222 | 23. $\underline{1222}$ | 24. $\underline{I 2}_{1} \underline{2}_{1} \underline{2}_{1}$ | 25. Pmm 2 |
| 26. $P_{m c 2}{ }_{1}$ | 27. $P \subset c 2$ | 28. $P$ ma2 | 29. ${\underline{C B a} 2_{1}}^{1}$ | 30. $\underline{P n c 2}$ |
| 31. $P_{m}$ | 32. $\underline{P b a 2}$ | 33. ${\underline{P n a} 2_{1}}^{1}$ | 34. $P$ nn2 | 35. Cmm 2 |
| 36. $\mathrm{Cmc2}{ }_{1}$ | 37. Ccc 2 | 38. Amm 2 | 39. Abm 2 | 40. Ama 2 |
| 41. $A$ ba 2 | 42. Fmm 2 | 43. Fdd 2 | 44. Imm 2 | 45. Iba 2 |
| 46. $\mathrm{Ima2}$ | 47. Pmmm | 48. $P \underline{n n n}$ | 49. Pccm | 50. Pban |
| 51. Pmma | 52. Pnna $^{\text {a }}$ | 53. $\mathrm{Pmna}^{\text {a }}$ | 54. Pcca | 55. Pbam |
| 56. $\underline{P c c n}$ | 57. Pbcm | 58. $\mathrm{Pnnm}^{\text {a }}$ | 59. Pmmn | $60 . \mathrm{Pbcn}$ |
| 61. Pbca | 62. Pnma | 63. Cmcm | 64. Cmca | 65. Cm m m |
| 66. Cccm | 67. Cmma | 68. Ccca | 69. Fmmm | 70. Fddd |
| 71. 1 mmm | 72. Ibam | 73. Ibca | 74. Imma |  |

Tetragonal
(For the enlarged $C$ - and $F$-centred unit cells, click here)


All space group diagrams and tables online:
http://img.chem.ucl.ac.uk/sgp/large/sgp.htm

## BIBLE OF CRYSTALLOGRAPHY

- Space groups (and the characteristic information in 2 pages) are listed in International Tables for Crystallography
- Next slide: Space Group P4/mmm as an example



Origia at centre (4/mmet
Asymmetric unit $\quad 0 \leq x \leq t ; 0 \leq y \leq t ; 0 \leq z \leq 1 ; x \leq y$

## Symmetry operations



## Maximal nee-isomorphic subgroups

```
I [2]P422
    [2]P4/暒11(P4/m)
    [2]P4mm
    [2]P42m
    [2]P4m2
    [2]P2/m2/m| (PN\piN)
        1;2:3;4;5;6;7;8
        1;2;3;4;9;10;11;12
    \ 1;2;5;6;11;12;15;16
    (2)
(1/ल (Cmmm) 1:2;7;8;9;10;15;16
```





## Maximal isomorphic subgroups of lowest index

Ile $\quad\left[2 \mid P 4 / m m m\left(c^{\prime}=2 c\right) ;[2] C 4 /\right.$ ww w $\left(a^{\prime}=2 a, b^{\prime}=2 b\right)(P 4 / \mathrm{mmm})$

## Minimal non-isomorphic supergroups

I $\quad[3] P_{w}{ }^{3}=x$
II $[2] \mathrm{J} / \mathrm{mmm}$

Generators selected (1); $t(1,0,0) ; \quad 1(0,1,0) ; \quad 1(0,0,1) ; \quad$ (2); (3); (5); (9)

## Positions

| Nat | Coordinates | Reflection conditions |
| :---: | :---: | :---: |

Sik ymemy.

|  |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |
|  |  |  |  |  |
|  |  |  |  |  |

General
no conditions

Special:
no extri conditions
no extra canditions
no extra conditions
to catra conditions
ho extra conditions
no extri conditions
no extri conditions
no extra conditions
50 entra conditions
Do extra condition:
no extra conditions
$k k 1: h+k=2 n$
no extra conditions
no extra cooditions
At : $: 4+k=2 \pi$
hat : $h+k=2 n$
no extra conditions
no extra conditions
ne extra conditions
no extra conditions

## Symmetry of special projections

| Along [001] $\mathrm{p}^{4 m \ldots 1}$ | Along [100] $\quad \mathrm{p} 2 \mathrm{~mm}$ | Along [110] p 2 mw |
| :---: | :---: | :---: |
| $a^{\prime}=a \quad b^{\prime}=b$ | $\mathbf{a}^{\prime}=\boldsymbol{b} \quad \boldsymbol{b}^{\prime}=\boldsymbol{c}$ | $a^{\prime}=:(-a+b) \quad b^{\prime}=c$ |
| Origin at $0,0, z$ | Origin $x$ a $x, 0,0$ | Orizin at $x, x, 0$ |

$\boldsymbol{a}^{\prime}=\boldsymbol{b} \quad \boldsymbol{b}^{\prime}=\boldsymbol{c}$
Origin $x a, 0,0$

Along [ILO] p2ww
$a=:(-a+b) \quad b^{\prime}=c$

## EXAMPLES of INFORMATION

- Space group number: 123
- Name: P4/mmm
- Complete name: $P 4 / m 2 / m 2 / m$; showing the symmetry elements (4fold rotation axis, mirror planes)
- Crystal system: tetragonal
- Lattice type: P (primitive)
- Site symmetry of the highest-symmetry site: $\mathrm{D}_{4 \mathrm{~h}}$
- Asymmetric unit (basis): smallest closed part of space the entire space is filled by applying all symmetry operations


## On the second page:

- List of possible sites for the atoms

These are indicated/named by: multiplicity, Wyckoff letter \& site symmetry

Not all sites are actually occupied by an atom

On the top: general site (16u)
At the bottom: the highest symmetry site (1a)

Multiplicity: number of identical sites
Generators selected (1); t(1,0,0); t(0,1,0); t(0,0,1); (2); (3); (5); (9)

| Positions |  |  |  |  |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Multiplicity. Wyckoff letter, Site symmetry |  |  |  | Coordinates |  |  |  | Reflection conditions <br> General: <br> no conditions |
| 16 | $u$ | $1 \begin{array}{rr}(1) \\ & (5 \\ & (9) \\ & (13)\end{array}$ |  | (2) $\bar{x}, \bar{y}, z$ <br> (6) $x, \bar{y}, z$ <br> (10) $x, y, z$ <br> (14) $\bar{x}, y, z$ | (3) $\bar{y}, x, z$ <br> (7) $y, x, z$ <br> (11) $y, \bar{x}, \bar{z}$ <br> (15) $\bar{y}, \bar{x}, z$ |  | (4) $y, \bar{x}, z$ <br> (8) $\bar{y}, \bar{x}, \bar{z}$ <br> (12) $\bar{y}, x, \bar{z}$ <br> (16) $y, x, z$ |  |
|  |  |  | $y, \bar{z}$ |  |  |  |  |  |
|  |  |  | $\bar{y}, \bar{z}$ |  |  |  |  |  |
|  |  |  |  |  |  |  |  |  |
|  |  |  |  |  |  |  |  | Special: |
| 8 | $t$ | . $m$. | $x, \frac{1}{2}, z$ | $\overline{\mathbf{x}}, \frac{1}{2}, z$ | $\frac{1}{2}, x, z$ | $\frac{1}{2}, \bar{x}, z$ |  | no extra conditions |
|  |  |  | $\bar{x}, \frac{1}{2}, \bar{z}$ | $x, \frac{1}{2}, \bar{z}$ | $\frac{1}{2}, x, \bar{z}$ | $\frac{1}{\frac{1}{2}, \bar{x}, \bar{z}}$ |  |  |
| 8 | $s$ | . $m$. | $x, 0, z$ | $\bar{x}, 0, z$ | 0,x,z | $0, \bar{x}, z$ |  | no extra conditions |
|  |  |  | $\bar{x}, 0, \bar{z}$ | $\boldsymbol{x}, 0, \bar{z}$ | $0, x, z$ | $0, \bar{x}, \bar{z}$ |  |  |
| 8 | $r$ |  | $x, x, z$ | $\bar{x}, \bar{x}, z$ | $\bar{x}, x, z$ | $x, \bar{x}, z$ |  | no extra conditions |
|  |  |  | $\bar{x}, x, \bar{z}$ | $x, \bar{x}, \bar{z}$ | $x, x, \bar{z}$ | $\bar{x}, \bar{x}, \bar{z}$ |  |  |
| 8 | $q$ | $m$. | $x, y, \frac{1}{2}$ | $\bar{x}, \bar{y}, \frac{1}{2}$ | $\bar{y}, x, \frac{1}{2}$ | $\bar{y}, \bar{x}, \frac{1}{2}$ |  | no extra conditions |
|  |  |  | $\bar{x}, y, \frac{1}{2}$ | $x, \bar{y}, \frac{1}{2}$ | $y, x, \frac{1}{2}$ | $\bar{y}, \bar{x}, \frac{1}{2}$ |  |  |
| 8 | $p$ | $m$. | $x, y, 0$ | $\bar{x}, \bar{y}, 0$ | $\bar{y}, x, 0$ | $y, \bar{x}, 0$ |  | no extra conditions |
|  |  |  | $\bar{x}, y, 0$ | $x, \bar{y}, 0$ | $y, x, 0$ | $\bar{y}, \bar{x}, 0$ |  |  |
| 4 | $o$ | $m 2 m$. | $x, \frac{1}{2}, \frac{1}{2}$ | $\bar{x}, \frac{1}{2}, \frac{1}{2}$ | $\frac{1}{2}, x, \frac{1}{2}$ | $\frac{1}{2}, \bar{x}, \frac{1}{2}$ |  | no extra conditions |
| 4 | $n$ | $m 2 m$ | $x, \frac{1}{2}, 0$ | $\bar{x}, \frac{1}{2}, 0$ | $\frac{1}{2}, x, 0$ | $\frac{1}{2}, \bar{x}, 0$ |  | no extra conditions |
| 4 | m | $m 2 m$. | $x, 0, \frac{1}{2}$ | $\bar{x}, 0, \frac{1}{2}$ | $0, x, \frac{1}{2}$ | $0, \bar{x}, \frac{1}{2}$ |  | no extra conditions |
| 4 | 1 | $m 2 m$. | $x, 0,0$ | $\boldsymbol{\chi}, 0,0$ | $0, x, 0$ | $0, \bar{x}, 0$ |  | no extra conditions |
| 4 | $k$ | m. 2 m | $x, x, \frac{1}{2}$ | $\bar{x}, \bar{x}, \frac{1}{2}$ | $\bar{x}, \boldsymbol{x}, \frac{1}{2}$ | $x, \bar{x}, \frac{1}{2}$ |  | no extra conditions |
| 4 | $j$ | $m .2 m$ | $\boldsymbol{x}, \boldsymbol{x}, 0$ | $\bar{X}, \bar{X}, 0$ | $\bar{x}, x, 0$ | $x, \bar{x}, 0$ |  | no extra conditions |
| 4 | $i$ | 2 mm . | 0, $\frac{1}{2}, z$ | $\frac{1}{2}, 0, z$ | 0, $\frac{1}{2}, z$ | $\frac{1}{2}, 0, \bar{z}$ |  | $h k l: h+k=2 n$ |
| 2 | $h$ | 4 mm | $\frac{1}{2}, \frac{1}{2}, z$ | $\frac{1}{2}, \frac{1}{2}, \bar{z}$ |  |  |  | no extra conditions |
| 2 | $g$ | 4 mm | 0,0,z | 0,0, $\bar{z}$ |  |  |  | no extra conditions |
| 2 | $f$ | $m m m$. | 0, 1,0 | $\frac{1}{2}, 0,0$ |  |  |  | $h k l: h+k=2 n$ |
| 2 | $e$ | $m m m$. | 0, ${ }^{\frac{1}{2}, \frac{1}{2}}$ | $\frac{1}{2}, 0, \frac{1}{2}$ |  |  |  | $h k l: h+k=2 n$ |
|  | $d$ | $4 / \mathrm{mmm}$ | $\frac{1}{2}, \frac{1}{2}, \frac{1}{2}$ |  |  |  |  | no extra conditions |
|  | $c$ | $4 / \mathrm{mmm}$ | $\frac{1}{2}, \frac{1}{2}, 0$ |  |  |  |  | no extra conditions |
|  | $b$ | $4 / \mathrm{mmm}$ | 0,0, $\frac{1}{2}$ |  |  |  |  | no extra conditions |
|  | $a$ | 4/mmm | 0,0,0 |  |  |  |  | no extra conditions |

Symmetry of special projections

Along [110] p2mm $\mathbf{a}^{\prime}=\frac{1}{2}(-a+b)$

## EXAMPLE: Potassium tetrachloroplatinate(II): $\mathrm{K}_{2} \mathrm{PtCl}_{4}$

Space group: $P 4 / m m m$ (No. 123)
Lattice parameters: $a=b=7.023 \AA, c=4.1486 \AA$
Atomic positions: Pt 1a: $0,0,0$
K $2 e: 0,1 / 2,1 / 2$
Cl 4j: $x, x, 0 ; x=0.23247$
(a) Draw the unit cell with the atoms.
(b) Draw the projection of the unit cell in $c$-axis direction.
(c) Theoretical density is $3.37 \mathrm{~g} / \mathrm{cm}^{3}$. Calculate Z ?
( $\mathrm{N}_{\mathrm{A}}=6.022 \times 10^{23}$; atomic weights: K 39.098 ; Pt 195.22; CI 35.453)
(d) Calculate the distances: $\mathrm{Pt}-\mathrm{Pt}, \mathrm{Pt}-\mathrm{K}, \mathrm{Pt}-\mathrm{Cl}$.
(e) What is the coordination number of platinum?
(f) What is the site symmetry of platinum?

## $P 4 / \mathrm{mmm} \quad D_{4 h}^{1}$

4／m m m
Tetragonal

No． 123
P4／m $2 / m 2 / m$
Patterson symmetry $P 4 / m m m$

Origia at centre（4／mmet
Asymmetric unit $\quad 0 \leq x \leq 1 ; 0 \leq y \leq t ; 0 \leq z \leq 1 ; x \leq y$

## Symmetry operations

| （1） 1 <br> （5） $20,9,0$ <br> （9） $10,0,0$ <br> （13） 地 $x, 0,2$ | （2） $20,0, z$ <br> （6） $2 x, 0,0$ <br> （10）m $x, y, 0$ <br> （14） m $0 . y, z$ | （3） $4^{*} 0,0,2$ <br> （7） $2, x, x, 0$ <br> （11） $4 \cdot 0,0, z: 0,0.0$ <br> （15）$m x, x, z$ | （4） $4^{-} \quad 0,0, z$ <br> （8） $2 x, \pi, 0$ <br> （12） $4-0.0 . z: 0,0.0$ <br> （16）$m ~ x, x, z$ |
| :---: | :---: | :---: | :---: |
|  | $P t$ | $1 a: 0,$ |  |
|  | $K$ | $\text { 2e: } 0,1$ | $1 / 2$ |
|  | $\Theta 1$ | $4 j: \quad X, X$ | $X=0$ |

## Maximal noe－isomorphic subgroups

I | $[2] P 422$ | $1 ; 2 ; 3 ; 4 ; 5 ; 6 ; 7 ; 8$ |
| :--- | :--- |
|  | $[2] P 4 / m 11(P 4 / m)$ |
|  | $1 ; 2 ; 3 ; 4 ; 9 ; 10 ; 11 ; 12$ |
| $[2] P 4 m m$ | $1 ; 2 ; 3 ; 4 ; 13 ; 14 ; 15 ; 16$ |
| $[2] P 42 m$ | $1 ; 2 ; 5 ; 6 ; 11 ; 12 ; 15 ; 16$ |
|  | $[2] P 4 / m 2$ |

IIa none
 $\left(21 C 4 / m \pi d\left(a^{\prime}=2 a \cdot b^{\prime}=2 b\right)(P 4 /\right.$ 皮 $b$ m $) ;\left(2 \mid C 4 / a\right.$ ww $\left(a^{\prime}=2 a, b^{\prime}=2 b\right)(P 4 / \pi m m) ;$
［2］F4／www $\left(a^{\prime}=2 a, b^{\prime}=2 b, c^{\prime}=2 c\right)(I 4 /$ womm $):[2] F 4 / m m c\left(a^{\prime}=2 a, b^{\prime}=2 b, c^{\prime}=2 c\right)(14 / w c$ w $)$

## Maximal isomorphic subgroups of lowest index

Ile $\quad[2] P 4 / m m m\left(c^{\prime}=2 \boldsymbol{c}\right) ;[2] C 4 / \mathrm{wtw}$ w $\left(\boldsymbol{a}^{\prime}=2 \boldsymbol{a}, \boldsymbol{b}^{\prime}=2 b\right)(P 4 / \mathrm{mmm})$
Minimal nen－isomorphic supergroups
I $\quad[3 \mid P \omega 1 / \pi$
II［2］ $4 / \mathrm{m} / \mathrm{mm}$



Generators selected（1）；$\quad 1(1,0,0) ; \quad 1(0,1,0) ; \quad 1(0,0,1) ;(2) ; \quad(1) ; \quad(5) ; \quad(9)$
Positions

| Maliplicty， <br> Wrchet！lestr． <br>  |  |  | Coordinates |  |  |  |  | Reflection conditions |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 16 | $\checkmark$ | $\begin{array}{r} (1) \\ (5) \\ (9) \\ \text { (13) } \end{array}$ | $y, 2$ $y, 2$ 9,2 $y, z$ | （2）$A_{1}, 5$ <br> （6）$x, 5$ <br> （10）$x, y$ <br> （14）$x, y$ |  | （3）$y \times x$ <br> （7）$y, x, z$ <br> （11）$y, f, t$ <br> （15） 5.8 .2 | （4）$y, x, z$ <br> （8） $9,8,5$ <br> （12）f．x．： <br> （16）$y_{-x}$ ．z | no conditions |
|  |  |  |  |  |  |  |  | Special |
| 8 | I | ．m | $\begin{aligned} & x, f, 2 \\ & x, t, i \end{aligned}$ | $\begin{aligned} & 8,+, z \\ & x, t, q \end{aligned}$ | $\begin{aligned} & i, x, z \\ & i, x, z \end{aligned}$ | $\begin{aligned} & i, R, 2 \\ & i, R, 8 \end{aligned}$ |  | no extri conditions |
| 8 | $s$ | m | $\begin{aligned} & x .0, z \\ & i, 0, Z \end{aligned}$ | $\begin{aligned} & 8,0-2 \\ & x, 0 \% \end{aligned}$ | $\begin{aligned} & 0, x, z \\ & 0, x, z \end{aligned}$ | $\begin{aligned} & 0, \boldsymbol{R}, z \\ & 0, \boldsymbol{R}, ? \end{aligned}$ |  | no extra conditions |
| 8 | $r$ | ．． 1 | $\begin{aligned} & x, x, z \\ & i, x, z \end{aligned}$ | $\begin{aligned} & x, x, z \\ & x, X, E \end{aligned}$ | $\begin{aligned} & f, x, z \\ & x, x, f \end{aligned}$ | $\begin{aligned} & x, R, Z \\ & f, f, \mathcal{L} \end{aligned}$ |  | no extra conditions |
| 8 | 9 | m ． | $\begin{aligned} & x, y, f \\ & f, y, f \end{aligned}$ | $\begin{aligned} & 8,9,1 \\ & x, 9,1 \end{aligned}$ | $\begin{aligned} & \text { 9.x.1 } \\ & y . x, 1 \end{aligned}$ | $\begin{aligned} & y, x, t \\ & y, x, t \end{aligned}$ |  | to catra conditions |
| 8 | $p$ | m．． | $\begin{aligned} & x, y, 0 \\ & f, y, 0 \end{aligned}$ | $\begin{aligned} & 8,5,0 \\ & x, 5,0 \end{aligned}$ | $\begin{aligned} & 9, x, 0 \\ & y, x, 0 \end{aligned}$ | $\begin{aligned} & Y . R .0 \\ & Y . R .0 \end{aligned}$ |  | （t）extra conditions |
| 4 | 0 | ＊ $2 \pi$ | x．1． 1 | s．t．t | t．x，${ }^{\text {¢ }}$ | t．8．t |  | no extry conditions |
| 4 | 月 | ＊2m | $x, 1,0$ | S． 1.0 | 1．8．0 | t．e． 0 |  | no extra conditions |
| 4 | $\cdots$ | $w 2 m$ 。 | $x, 0, \ddagger$ | 8，0，1 | 0．x．i | 0.2 .4 |  | no extra eceditions |
| 4 | 1 | m 2m | $x, 0,0$ | 1． 0,0 | 0，x，0 | 0， 8.0 |  | no eltra conditions |
| 4 | $k$ | m 2 m | $x, x, 1$ | R，R，$\dagger$ | 2，$\times$ ．+ | 8，Q， |  | no eitra conditions |
| 4 | ） | W1．2w | $x, x, 0$ | 8． 8.0 | 1，x， 0 | $x, 8,0$ |  | no extra conditions |
| 4 | 1 | $20 \%$ | 0． 1.2 | 4，0，z | 0，2．？ | 1．0．8 |  | kki ：$\lambda+k=2 n$ |
| 2 | 万 | 480\％ | t， 5.2 | 4， 8.2 |  |  |  | no extra conditions |
| 2 | $g$ | 4 Nm | 0，0， 2 | 0，0，t |  |  |  | no extra cooditions |
| 2 | I | MNM | 0，4，0 | 1，0，0 |  |  |  | hel ：$A+k=2 N$ |
| 2 | $e$ | mmm | 0，t， 1 | 4，0， |  |  |  | hikl ：$h+\hat{k}=2 n$ |
| 1 | $d$ | 4／womm | 1．1．i |  |  |  |  | no extra conditions |
| 1 | $c$ | 4／080＊ | 1．1．0 |  |  |  |  | no extra conditions |
| 1 | $b$ | 4／ヵли | 0，0，$\dagger$ |  |  |  |  | ne extre conditions |
| 1 | $a$ | 4／mmm | 0，0，0 |  |  |  |  | no extra conditions |

[^0]（Continned on preceding page）
no extra conditions

Along［110］p2ww
$a^{\prime}=:(-a+b) \quad b^{\prime}=c$ Orizin at $x, x, 0$


Pt: 1 atom in unit cell
K: 2 atoms in unit cell
CI: 4 atoms in unit cell

Bond lengths:
Pt-Pt: $(1-0)^{2} \cdot 4.15 \AA ̊$
Pt-K: $\sqrt{ }\left[(0.5-0)^{2} \cdot 7.023 \AA ̊+(0.5-0)^{2} \cdot 4.149 \AA\right]=4.08 \AA$
Pt-Cl: $\sqrt{ }\left[(0.232-0)^{2} \cdot 7.023 \AA+(0.232-0)^{2} \cdot 7.023 \AA ̊\right]=2.30 \AA$


Site symmetry of Pt : $\mathrm{D}_{4 \mathrm{~h}}$
$a b$-projection (seen from c-direction)


## $\mathrm{K}_{2} \mathrm{PtCl}_{4}$

- $\rho=3.37 \times 10^{6} \mathrm{~g} / \mathrm{m}^{3}$
- $V=7.023 \AA \times 7.023 \AA \times 4.1486 \AA=204.62 \times 10^{-30} \mathrm{~m}^{3}$
- $M=(2 \times 39.098+195.22+4 \times 35.453) \mathrm{g} / \mathrm{mol}=415.228 \mathrm{~g} / \mathrm{mol}$
- $Z=\left(\mathrm{V} \times \rho \times \mathrm{N}_{\mathrm{A}}\right) / \mathrm{M}=1$
- Distances:Pt-Pt: $4.15 \AA$

Pt-K: $4.08 \AA$
Pt-Cl: $2.31 \AA(\rightarrow$ chemical bond $)$

- $\mathrm{CN}(\mathrm{Pt})=4$
- PI site symmetry: $D_{4 h}$


## WHAT WE LIKE TO KNOW ABOUT THE CRYSTAL STRUCTURE

## CRYSTALLOGRAPHY

- symmetry
- unit cell
- lattice parameters
- number of formula units in unit cell
- space group
- etc.


## CRYSTAL CHEMISTRY

- coordination numbers
- coordination polyhedra
- bond lengths/angles
- occupation factors
- etc.

Discussed in Lecture 2!

$\mathrm{HgBa}_{2} \mathrm{Ca}_{2} \mathrm{Cu}_{3} \mathrm{O}_{9-\delta}$

## EXAMPLE: Chromium oxychloride CrOCI

Space group Pmmn (No. 59)
Lattice parameters: $a=3.88 \AA, b=3.20 \AA, c=7.72 \AA(Z=2)$
Atomic positions:
$\begin{array}{lll}C r & 2 a & z=0.109 \\ \text { Cl } & 2 b & z=0.327 \\ 0 & 2 b & z=0.960\end{array}$
(a) Draw the unit cell.
(b) Give for chromium:

- bond lengths
- coordination numbers
- site symmetry
(c) Calculate BVS for chromium. [ $R^{0}$ values: $\mathrm{Cr}^{\text {III }}-\mathrm{O}^{-1 \mathrm{II}}: 1.724, \mathrm{CrIII}^{\text {II }} \mathrm{Cl}^{-1}: 2.08$ ]


## Pmmn $\quad D_{2 h}^{13}$ <br> $\mathrm{mmm} \quad$ Orthorhombic

No, 59
$P 2_{1} / m 21 / m 2 / n$
Patterson symmetry $\boldsymbol{P}_{\text {mmm }}$
ORIGIN CHOICE I


Origin si wiw $2 / n, 3 t \ddagger, \ldots, 0$ from
Asymmetrie unit 0SxSt; OSySt; 0Szst

## Symmetry optratioas

(1)
(2) $20,0, z$
(3) $20,2,0,0, t, y, 0$
(4) $2( \pm, 0,0), \pi, 1,0$
(5) $] \quad 1,1,0$
(6) $\mathrm{n}(5,2,0) \quad x, y, 1)$
(7) $\pi \times, 0, t$
(B) $w \quad 0, y, z$

Generators selected (1) $\quad \mid(1,0,0) ; \quad\{(0,1,0) ; \quad r(0,0,1) ; \quad(2) ;(3) ;(5)$

## Pasitions

Matigaing

Mychat kakd,
fir motion


Maximal non-isomorphic subgroups
I 12$] P 2,2 ; 2 \quad 1 ; 2 ; 3 ; 4$
[2]P112/K(P2/c) $\quad 1 ; 2 ; 5 ; 6$
(2)P12, $/ m \mid\left(P_{2}, / m\right) \quad 1 ; 3 ; 5 ; 7$
 [2]Pmen 2

1;4;5;8

[2]P2 1 mn $\left(P m \in 2_{1}\right) \quad 1 ; 4 ; 6 ; 7$
IIa nose

Maximal Isomorphic subgroeps of lowest index
Ifc [3]Pminn $\left(a^{\prime}=3 a\right.$ or $\left.b^{\prime}=3 b\right) ;[2]$ Pwiwn $\left(c^{\prime}=2 c\right)$

## Minimal non-soanorphic superzroups

1 [2]P4/nmm; [2]P4, /nwe
 [2] Р届畕 $a\left(2 b^{\prime}=b\right)$


## Chromium bonding

$2 \times \mathrm{Cr}-\mathrm{Cl}: \sqrt{ }\{(0.891-0.673) \times 7.72 \AA\}^{2}+\{0.5 \times 3.20 \AA\}^{2}=2.3222 \AA$ $2 \times \mathrm{Cr}-\mathrm{O}: \sqrt{ }\{(0.960-0.891) \times 7.72 \AA\}^{2}+\{0.5 \times 3.88 \AA\}^{2}=2.0118 \AA$ $2 \times \mathrm{Cr}-\mathrm{O}: \sqrt{ }\{[(1-0.891)+0.04] \times 7.72 \AA\}^{2}+\{0.5 \times 3.20 \AA\}^{2}=1.9706 \AA$
$\mathrm{CN}(\mathrm{Cr})=6$
Cr site symmetry: $\mathrm{C}_{2 \mathrm{v}}$
BVS(Cr): +2.985

## CrOCl : simulated XRD pattern based on the structure data



## Your EXERCISE question

White balls are Ti atoms, red balls are oxygen atoms.
Unit cell parameters: $\mathrm{a}=\mathrm{b}=4.5937 \AA, \mathrm{c}=2.9587 \AA$; all angles $90^{\circ}$.
(a) What is the crystal system?
(b) What is the formula of the compound?
(c) Please calculate the density.


## Some extra slides...

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## Supporting material

Mathematical descriptions of 2D point groups:
https://www.cryst.ehu.es/plane/get point genpos.html https://en.wikipedia.org/wiki/Point groups in two dimensions

3D models with symmetry element visualizations
... For molecules
https://symotter.org/gallery
... For crystals
https://crystals.symotter.org/viztools/

## CLASSIFICATIONS

- "Macroscopic shape of the crystal"
$\rightarrow$ "Point group for the lattice" $\rightarrow 7$ CRYSTAL SYSTEMS (can fill the space without holes; no information of the lattice points/atoms)


## CRYSTAL SYSTEM \& LATTICE POINTS (historical importance)

- Combination of crystal system and lattice type $\rightarrow 14$ Bravais lattices
- Lattice type: positions of lattice points ( $\neq$ atoms) within the cell considered: primitive (P), body-centered (I), face-centered (F), base-centered (A/B/C), rhombohedral (R)


## CRYSTAL SYSTEM \& SYMMETRY (most important in crystallography)

- Possible combinations of point symmetry elements (which leave one point fixed)
$\rightarrow 32$ (geometric) cystal classes
$\rightarrow$ describe completely the symmetry of macroscopic crystals
There are an infinite number of three-dimensional point groups, but the crystallographic restriction results in there being only 32 crystallographic point groups.
- Considering also the translational symmetry operations in 3D $\rightarrow 230$ space groups

| Crystal System | \# of Point Groups | \# of Crystal Lattices |
| :--- | :---: | :---: |
| Cubic | 5 | 3 |
| Tetragonal | 7 | 2 |
| Orthorhombic | 3 | 4 |
| Monoclinic | 3 | 2 |
| Triclinic | 2 | 1 |
| Hexagonal | 7 | 1 |
| Trigonal | 5 | 1 |
|  | 32 | 14 |


[^0]:    Symmetry of special projections
    Along［OO1］$\rho 4$ man
    $a^{\prime}=a \quad b^{\prime}=b$
    Origin at $0,0,2$

