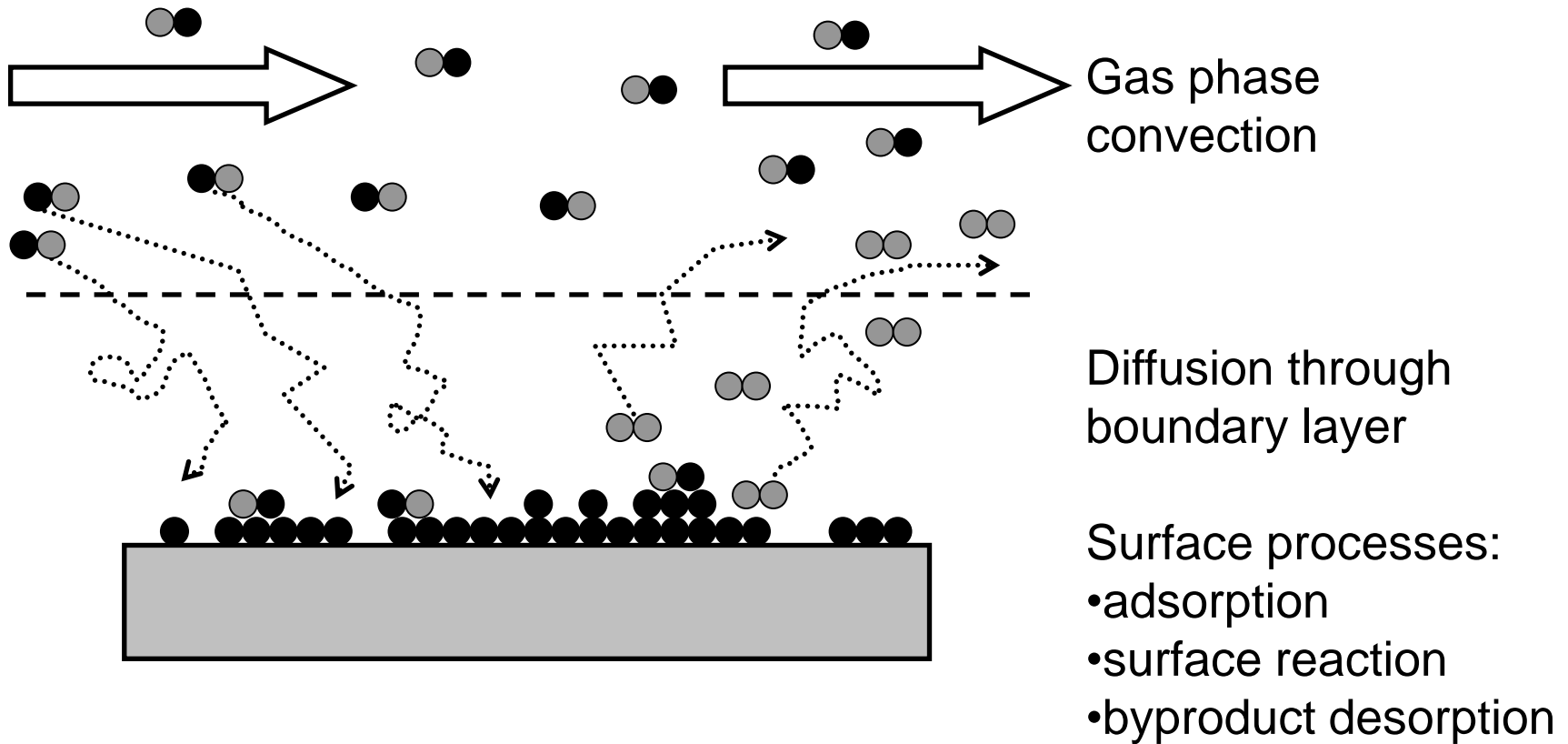


# CVD & ALD

(2 hour set for 2021 TFT-course)

sami.franssila@aalto.fi

# CVD schematically



# First lecture: CVD

Gas phase  
convection

How are gases introduced to a reactor  
and how they flow?

Diffusion through  
boundary layer

Basic physics of boundary layer;  
LPCVD (Low Pressure CVD);  
And how this affects reactor design

Surface processes:

- adsorption
- surface reaction
- byproduct desorption

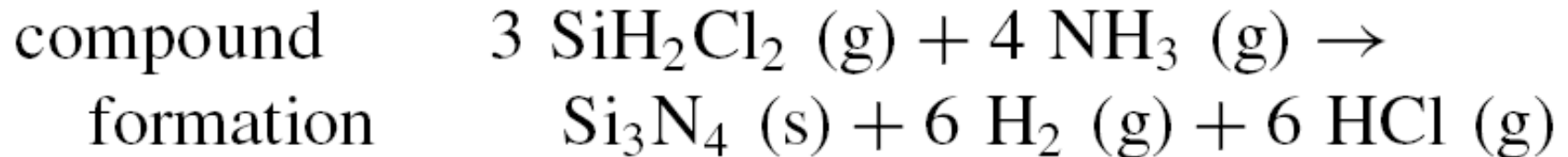
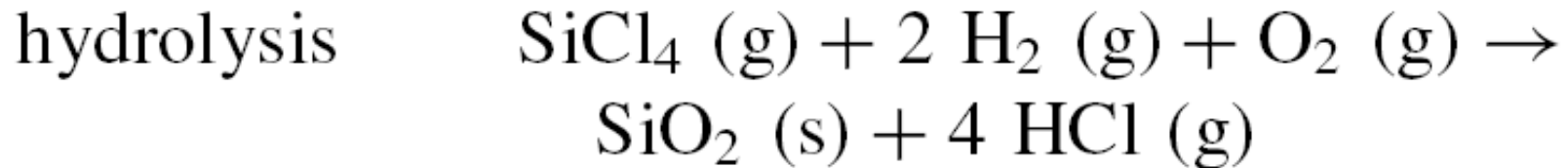
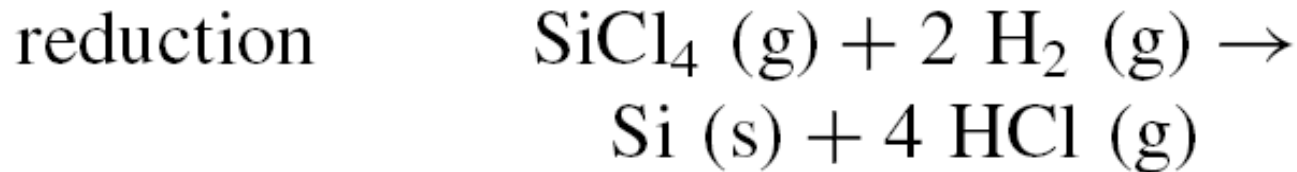
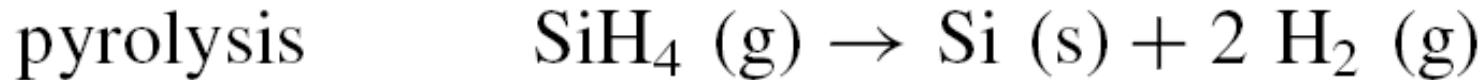
Not so much on surface processes  
(but a lot in case ALD)

The resulting film

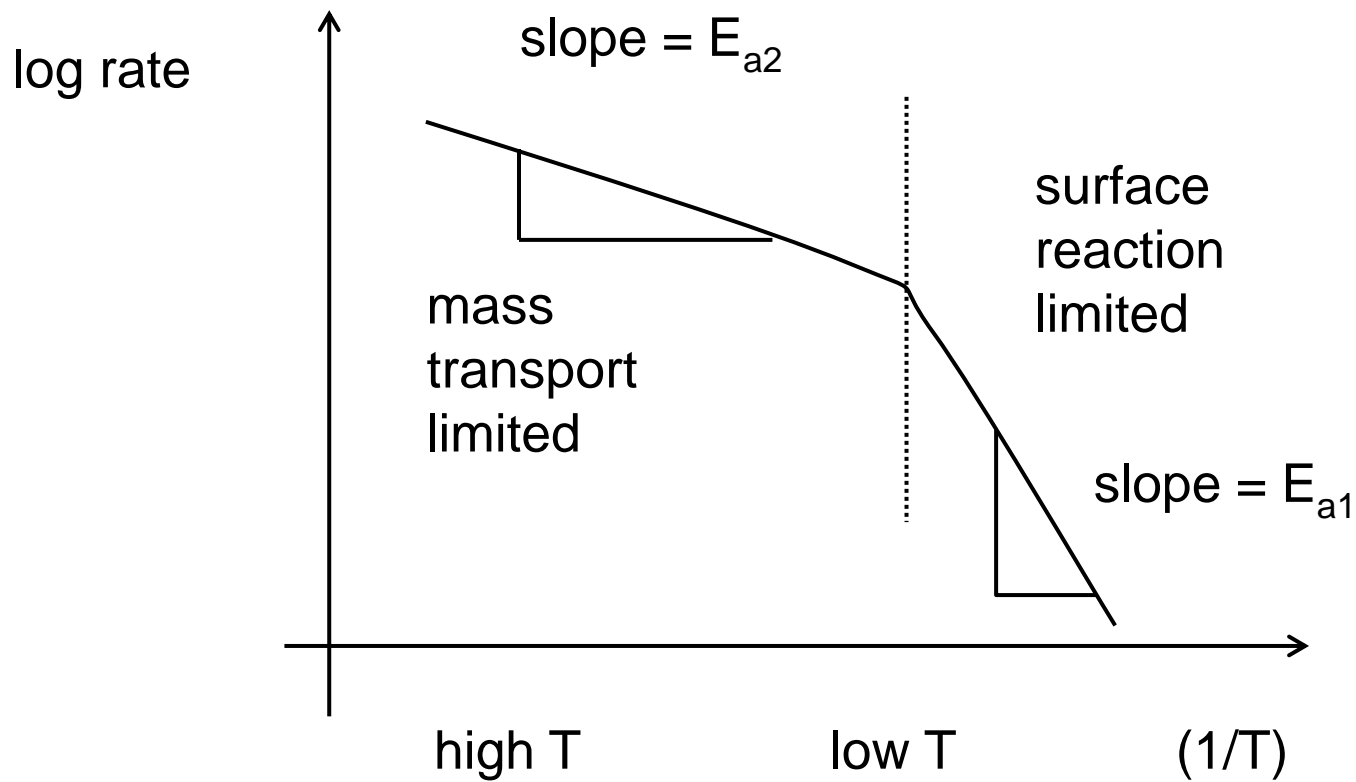
Factors affecting the film structure  
Measures of film quality

# Thermal CVD reactions

**Gaseous precursor + surface reaction**  
**→ solid film + gaseous byproducts**



# Surface limited vs. mass transport limited reactions



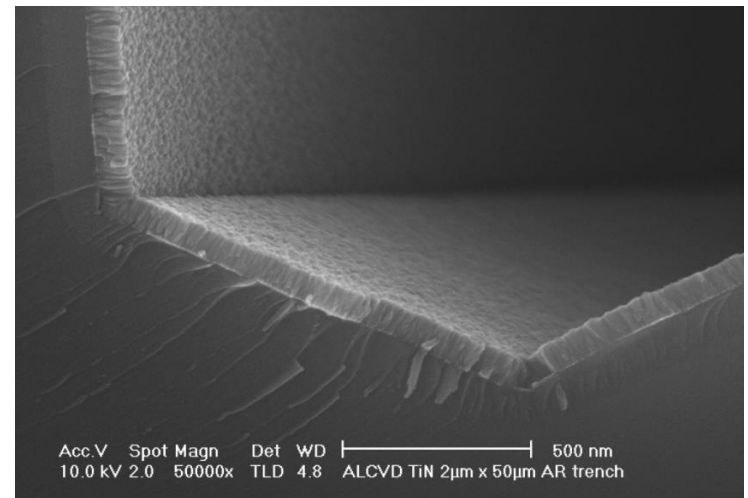
# Surface controlled reaction

Slow reaction rate (e.g. due to low temperature). Lots of gas available, and only a fraction of it has chance to react.

Because all surfaces are at same temperature, same deposition rate everywhere. Because surface controlled → good step coverage.

ALD is a prime surface controlled reaction, excellent conformality. Thermal CVD (especially LPCVD) also excellent conformality.

ALD TiN at the bottom of high aspect ratio groove.

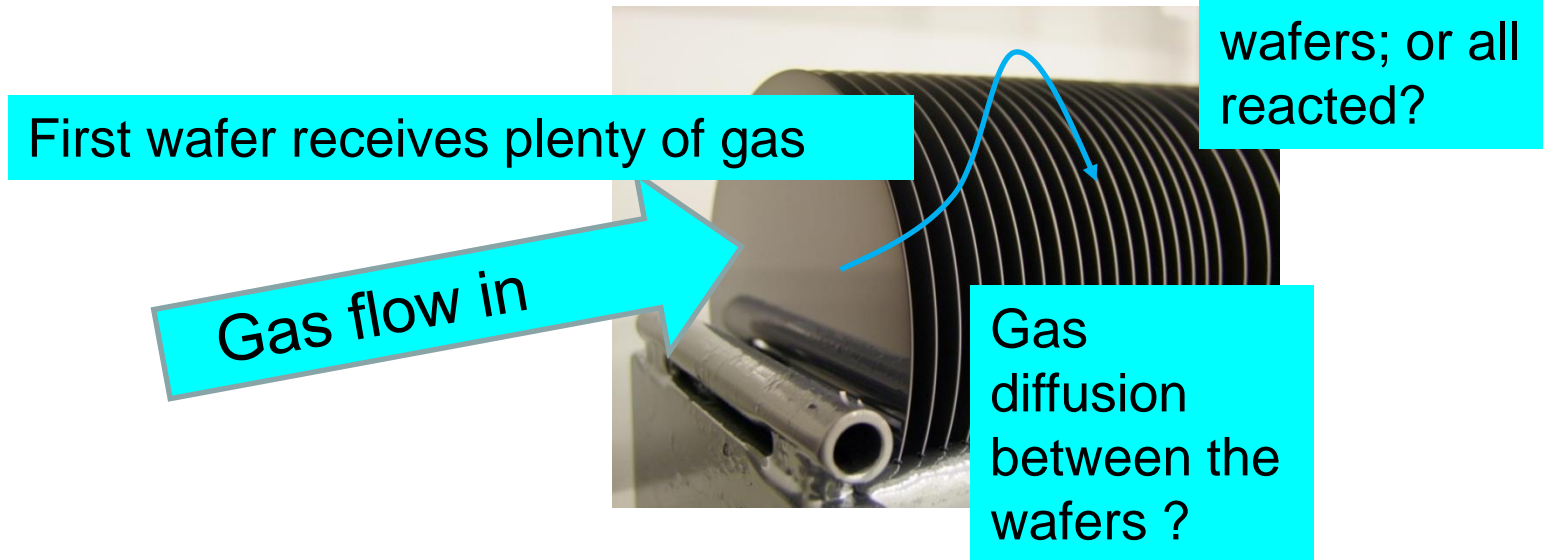


# Mass transport limited reaction

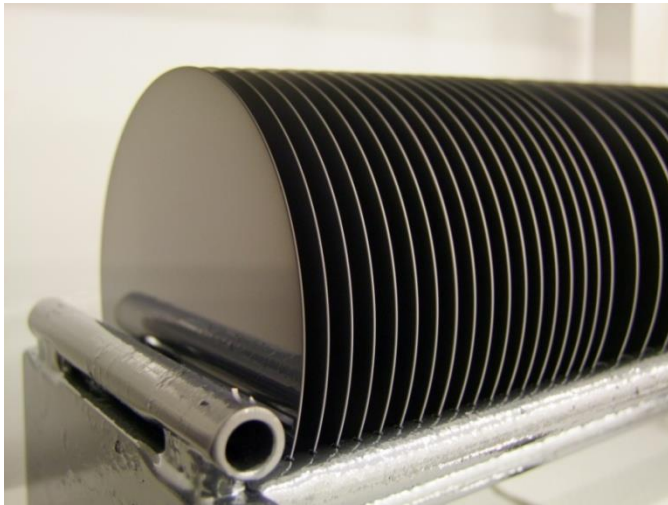
Reaction rate is very fast at high temperatures  
(Arrhenius: rate is exponentially temperature-dependent).

All arriving gases react immediately → need to ensure that gases arrive equally to all parts of reactor. If not, position dependent depo rate.

Reaction is in mass transport limited mode.



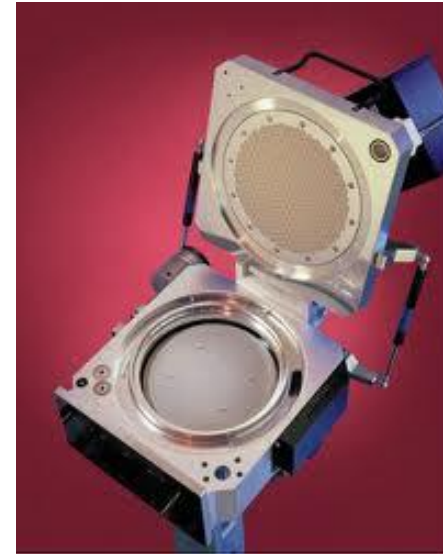
# Surface limited vs. mass transport limited reactions



Surface reaction limited mode:

- slow reaction rate
- time for gases to diffuse
- always extra gas available
- can pack wafers tightly

**Batch reactors**



A mass transport limited reactor:

- all arriving gases react at once
- therefore all wafers need to experience the same gas flow
- easier to design uniform flow for

**single wafer reactors**



# Rate modeling

$$J_{\text{gas-to-surface}} = -\frac{D}{\delta} C_{\text{gas}}$$

Diffusion of precursor gas from main flow to the surface.

$$\delta = \sqrt{\frac{\eta L}{v\rho}}$$

Boundary layer thickness  $\delta$ .

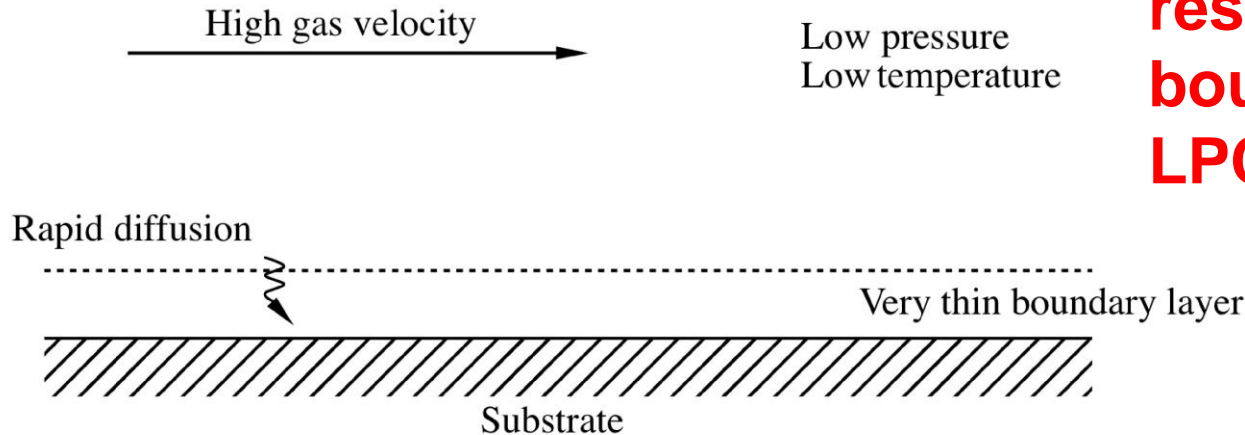
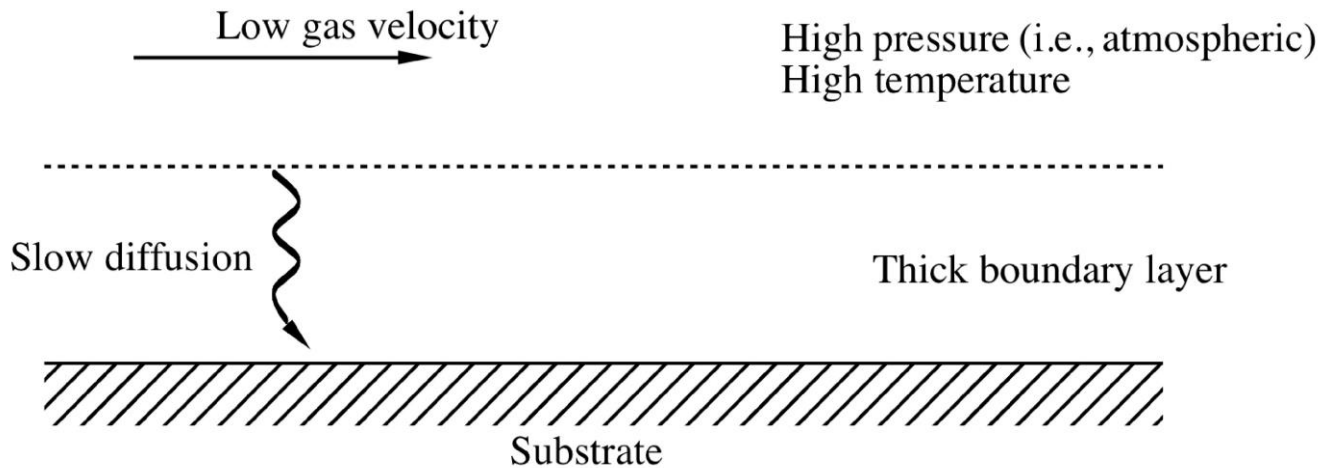
Raising temperature  $\rightarrow$  small changes in density, velocity and viscosity

$$D \propto \frac{T^{\frac{3}{2}}}{P}$$

But lowering pressure, e.g. by 1000X  
 $\rightarrow$  Diffusivity  $D$  increases 1000-fold.

$\rightarrow$   $J_{\text{gas-to-surface}}$  increases dramatically

# Boundary layer thickness



**These variables result in thin boundary layer → LPCVD reactor**

# CVD reactors are classified by their operating pressure range:

- atmospheric pressure, APCVD
- sub-atmospheric, SACVD 10-100 Torr
- low-pressure, LPCVD at ~Torr
- ultra-high vacuum, UHV-CVD,  $10^{-6}$  Torr base pressure
- In UHV reactors the actual process pressures are 1-10 mTorr when gases are flowing, very much like magnetron sputtering systems. In both cases good base vacuum (of  $10^{-6}$  –  $10^{-9}$  Torr level) is mandatory for removal of residual gases from the chamber.

# How are CVD processes developed ?

What reactor parameters to vary ?

What film quality targets ?

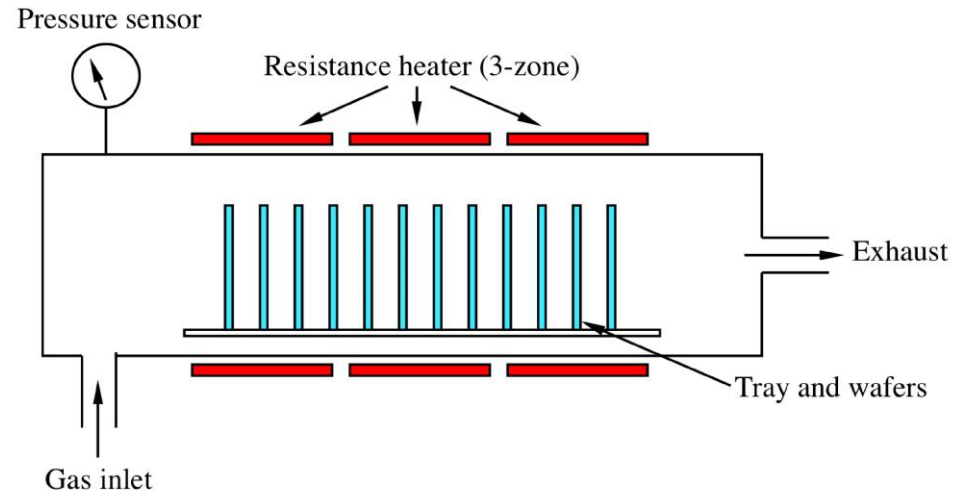
Film quality measurements ?

What productivity targets?

# Case: LPCVD polysilicon



- Pressure
- Flow rate
- Temperature
- Source gas purity
- Reactor vacuum (residual gases, leaks)
- Reactor cleanliness (particles)



# Targets

## Film:

- Crystalline structure (e.g. grain size, orientation)
- Atoms & bonds (e.g. Si-Si bonds vs. Si-H; dangling bonds Si-\*)
- Key parameters ( $t$ ,  $R_s$ ,  $n_f$ ,  $\epsilon$ , ...)
- Uniformity (of thickness & other parameters)
- Impurities and defects

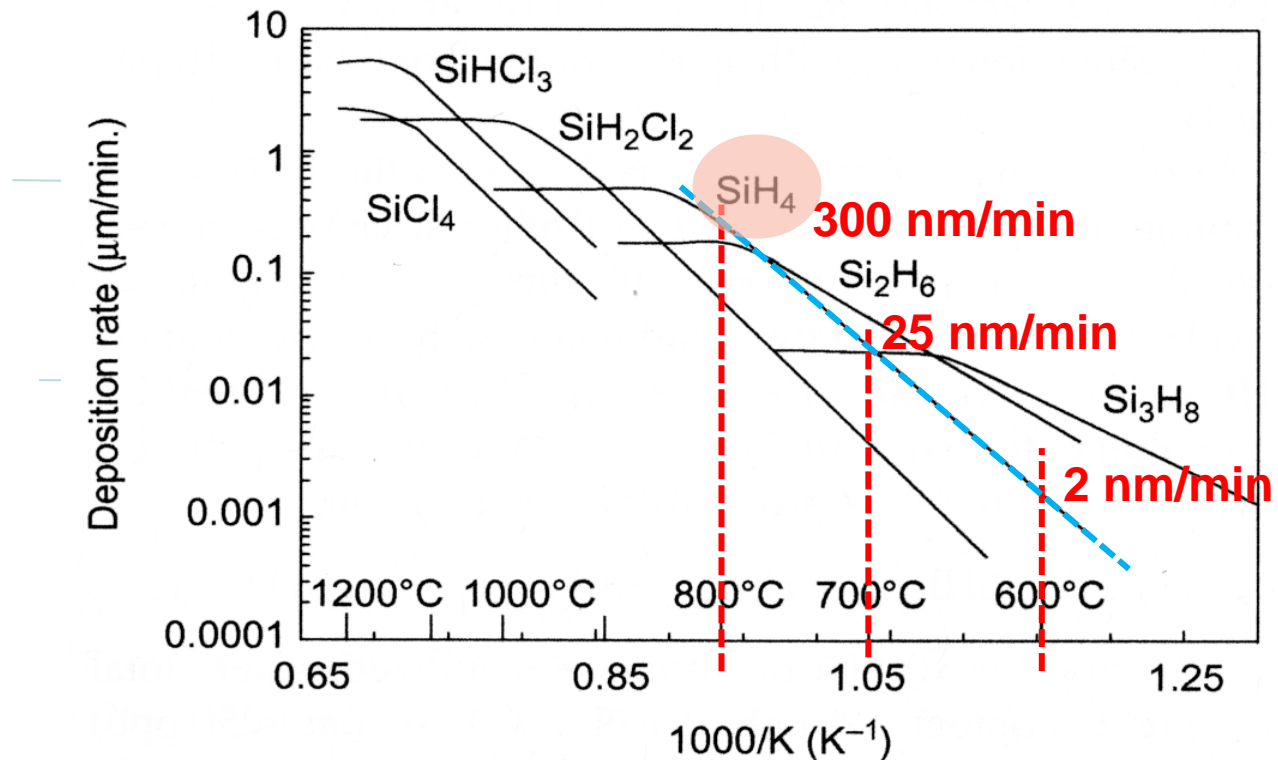
## Productivity:

- Deposition rate (in units nm/min)
- Thruput (in units wafers/hour = WPH)
- Reproducibility (run-to-run; day-to-day)
- Yield (particles, out-of-specification thickness...)

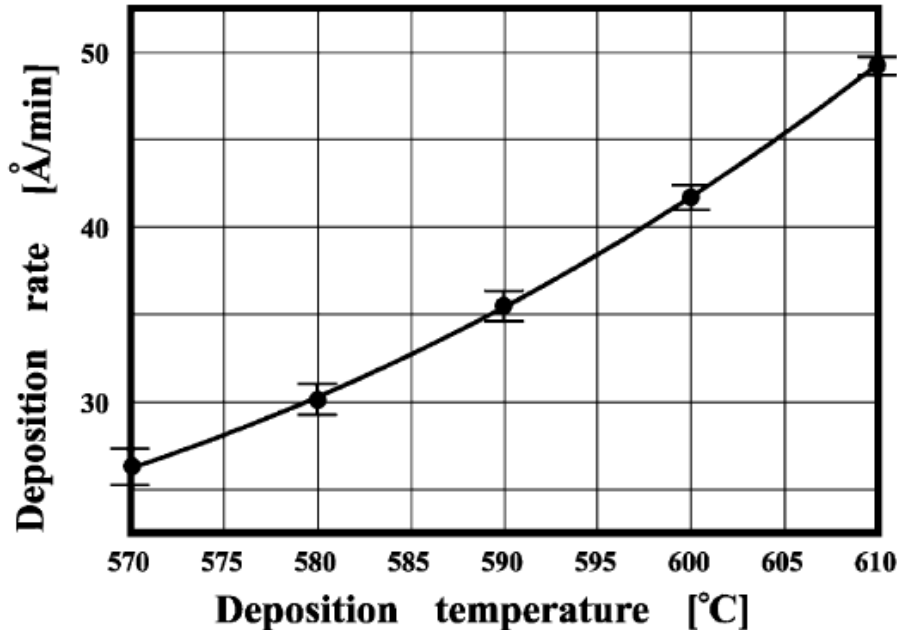
# Deposition rate

Arrhenius behavior.

Below 800°C surface reaction controlled.



# Deposition rate (2)



Undoped polysilicon

$\text{PH}_3$  dopant ( $\rightarrow$  PSG, phosphorous doped silica glass)

Phosphorous retards deposition rate via high rate adsorption of phosphorous which blocks adsorption sites for silane

$\text{BCl}_3$  dopant: (BSG)

Boron enhances deposition rate via unknown mechanism.

Also: PBSG



# Structure and Properties of LPCVD Silicon Films

T. I. Kamins\*

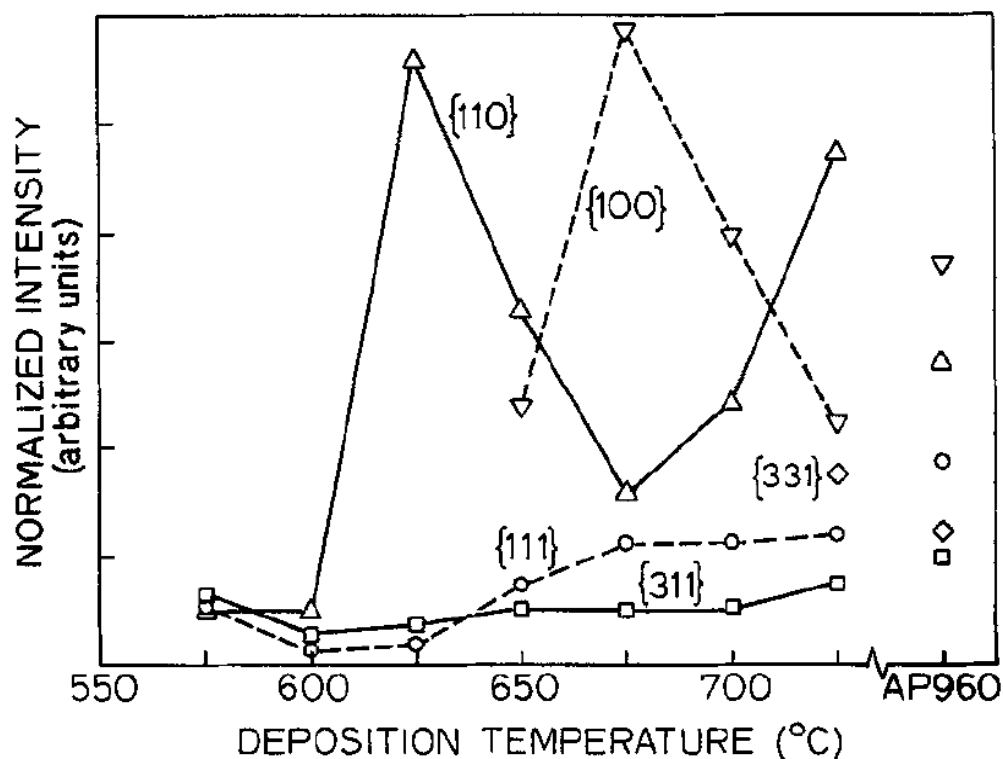
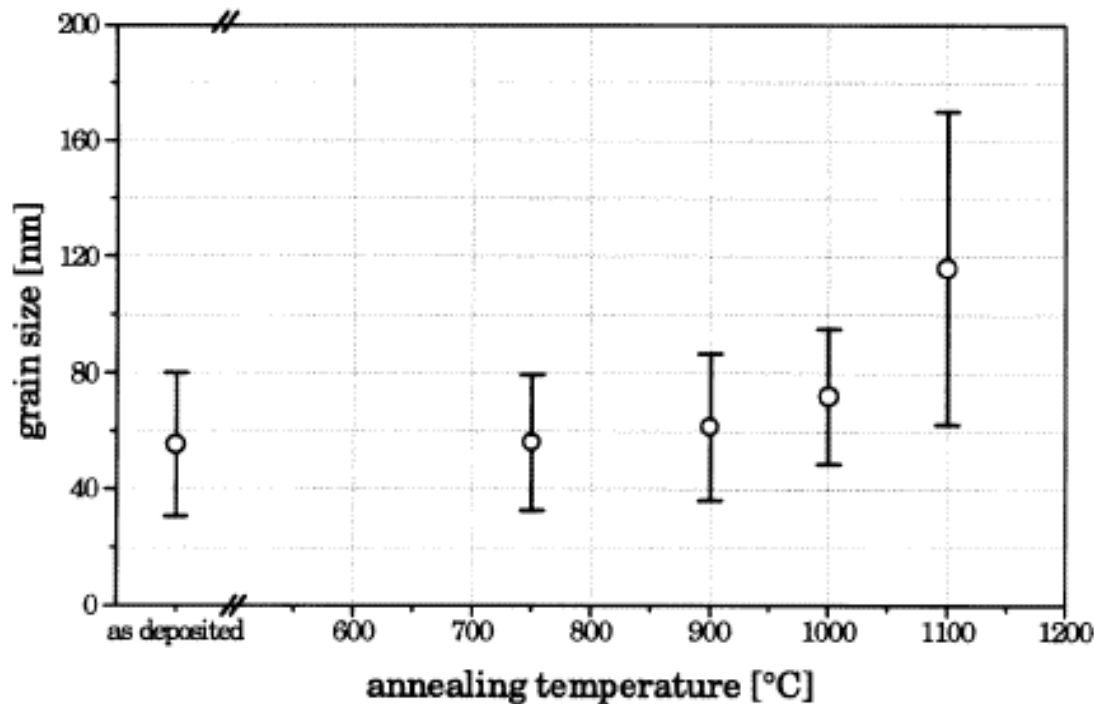


Fig. 1. X-ray texture as a function of deposition temperature for LPCVD silicon films and for an atmospheric pressure film.

**Table I. Average grain size as a function of deposition temperature**

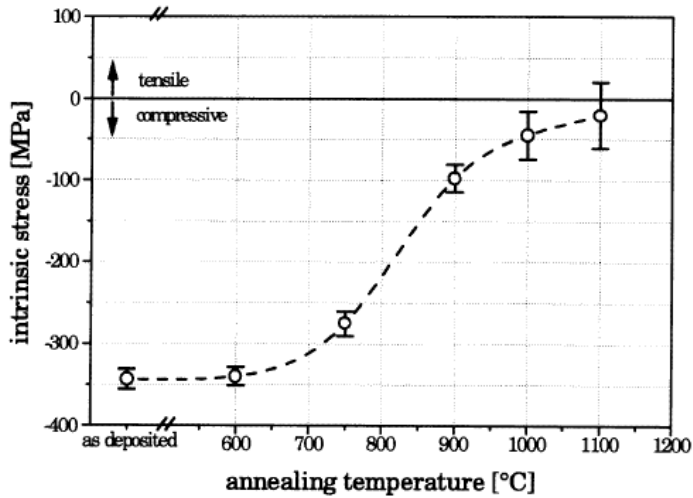
Temperature (°C)	Grain size (nm)
600	55
625	87
650	72
675	74
700	73
725	86

Deposition temperature affects grain size

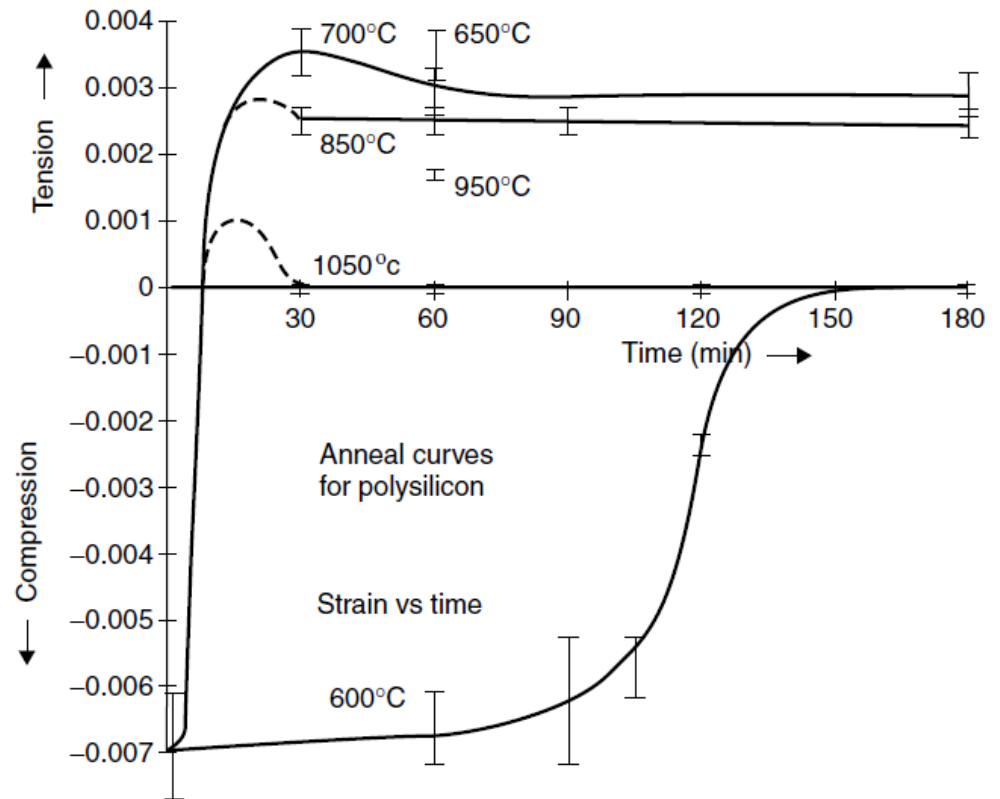


Post-deposition anneal affects grain size

# Stress relief anneal



Stress reduction is anneal temperature and time dependent !



580°C deposition, i.e. amorphous initially

# a-Si by LPCVD

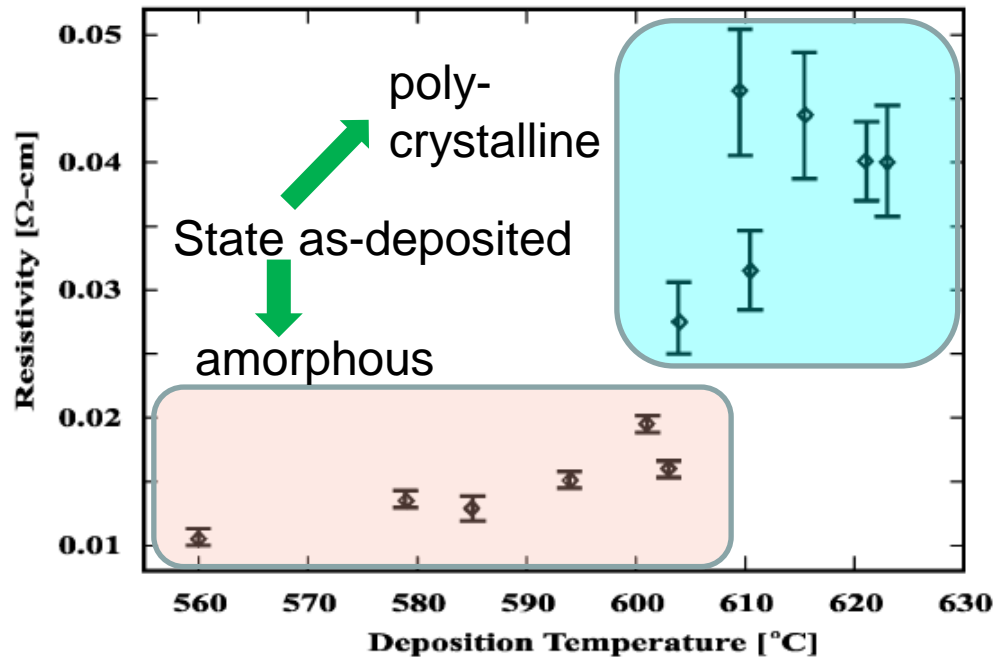
~600°C or lower deposition temperature the resulting film will be amorphous.

When amorphous film is annealed, it will behave differently from polycrystalline.

In polycrystalline film certain things are “frozen”, e.g. crystal directions.

Amorphous film can crystallize in whatever way it wants...

# Doping of a-Si vs. poly

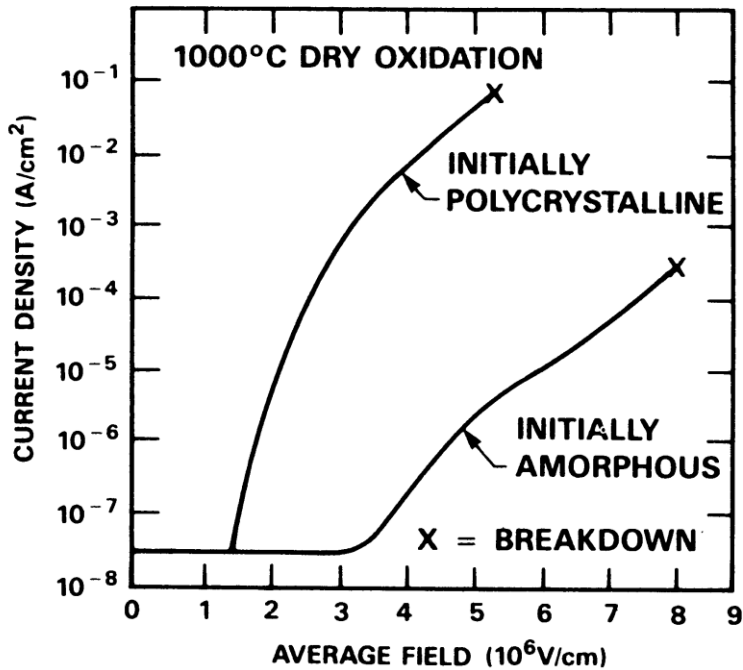


Resistivity after doping:

Ion implantation into film ( $1 \times 10^{15}$  ions/cm<sup>2</sup>) and subsequent anneal (1000 $^{\circ}\text{C}$ ).

Why the difference ?

Grain boundaries trap dopant atoms; more available in initially amorphous film.

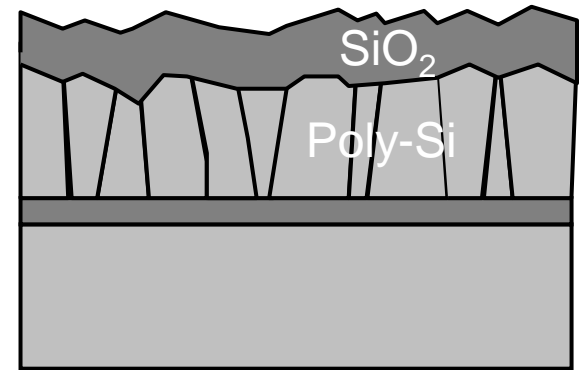
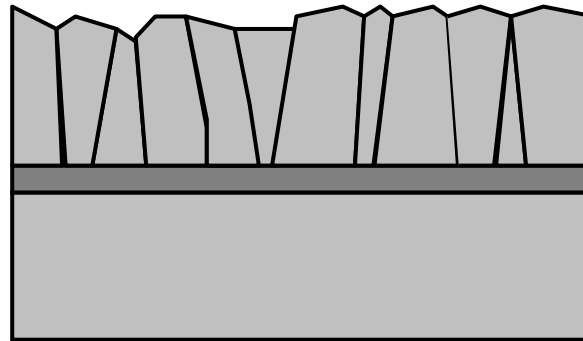
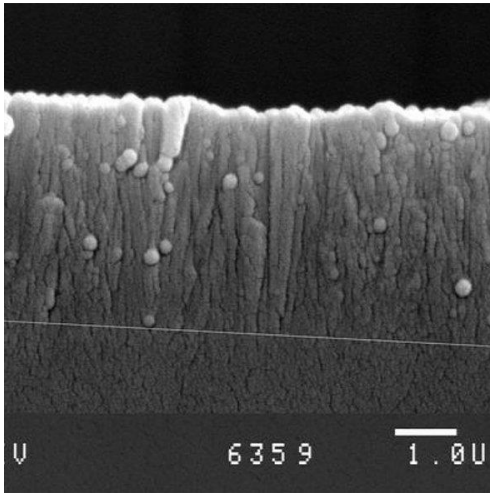


Breakdown field is a common measure of oxide film quality.

Why is initially amorphous film better in terms of oxide breakdown ?

Note: Oxide on monocrystalline silicon ca. 10 MV/cm

# Poly oxidation (2)



- Polycrystalline film is rougher than amorphous film
- oxidized polycrystalline film has variable thickness (because grain orientation affects oxidation rate)
- electric field concentration at tips and thin points
- easy breakdown

# Grain size & roughness

AFM:  
surface roughness

$S_q=40\text{nm}$

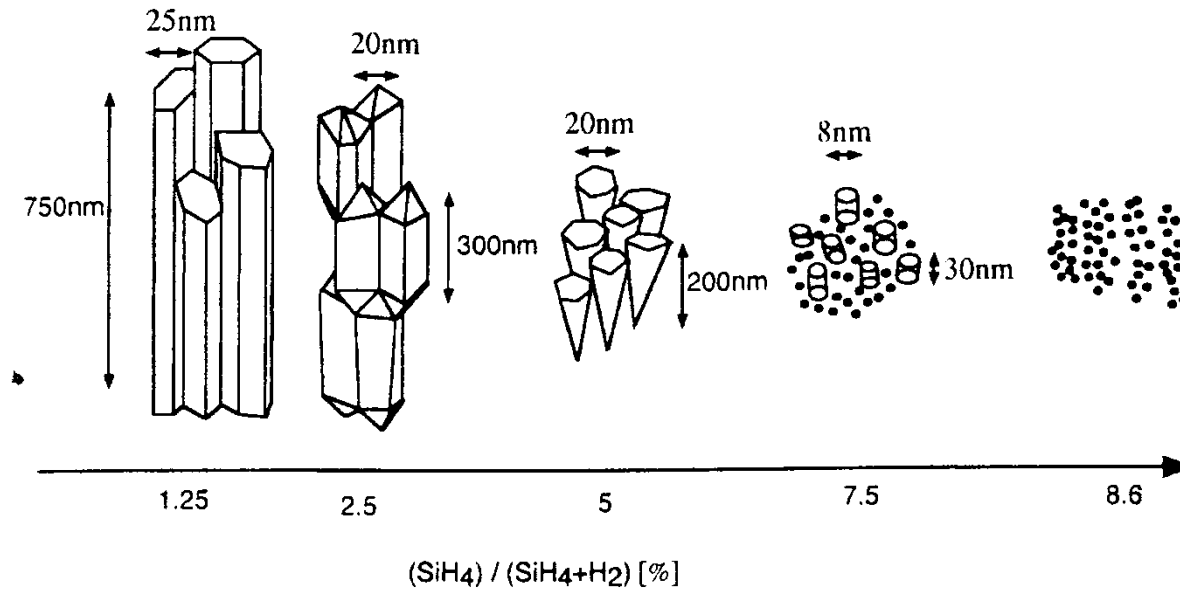
$S_q=18\text{nm}$

$S_q=17\text{nm}$

$S_q=16\text{nm}$

$S_q=4\text{nm}$

TEM:  
size and shape of the grains





# Polysilicon thin film vs. <Si>

Density: same 2.3 g/cm<sup>3</sup>

Young's modulus: same 170 GPa

CTE: same 2.5 ppm/K

Thermal conductivity:

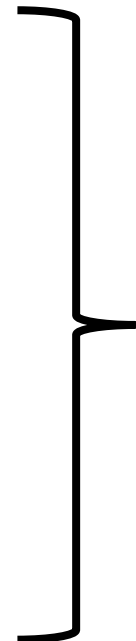
<Si> 156 W/K\*m (at room temp)

poly 32 W/K\*m

Carrier mobility:

<Si> 100 cm<sup>2</sup>/Vs

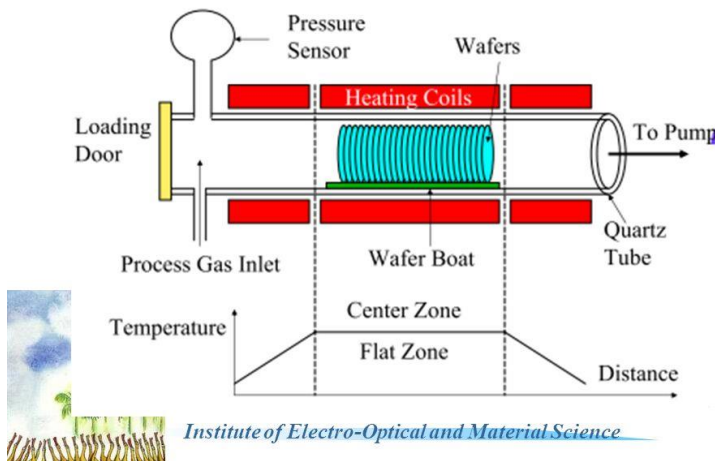
poly 10 cm<sup>2</sup>/Vs



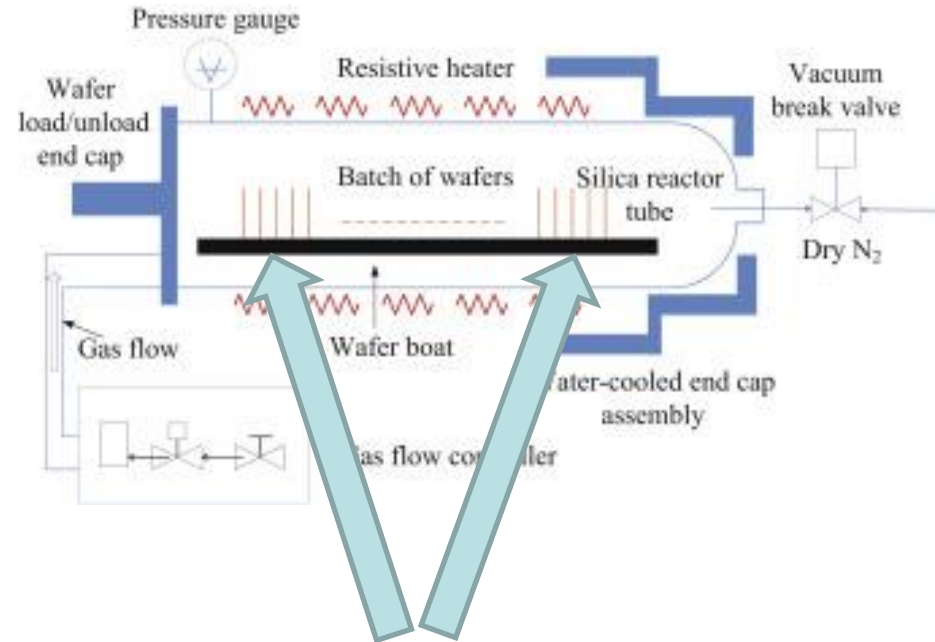
Affected by grain boundaries; electrons and phonons experience losses at boundaries

# Temperature control zones

## LPCVD System



19

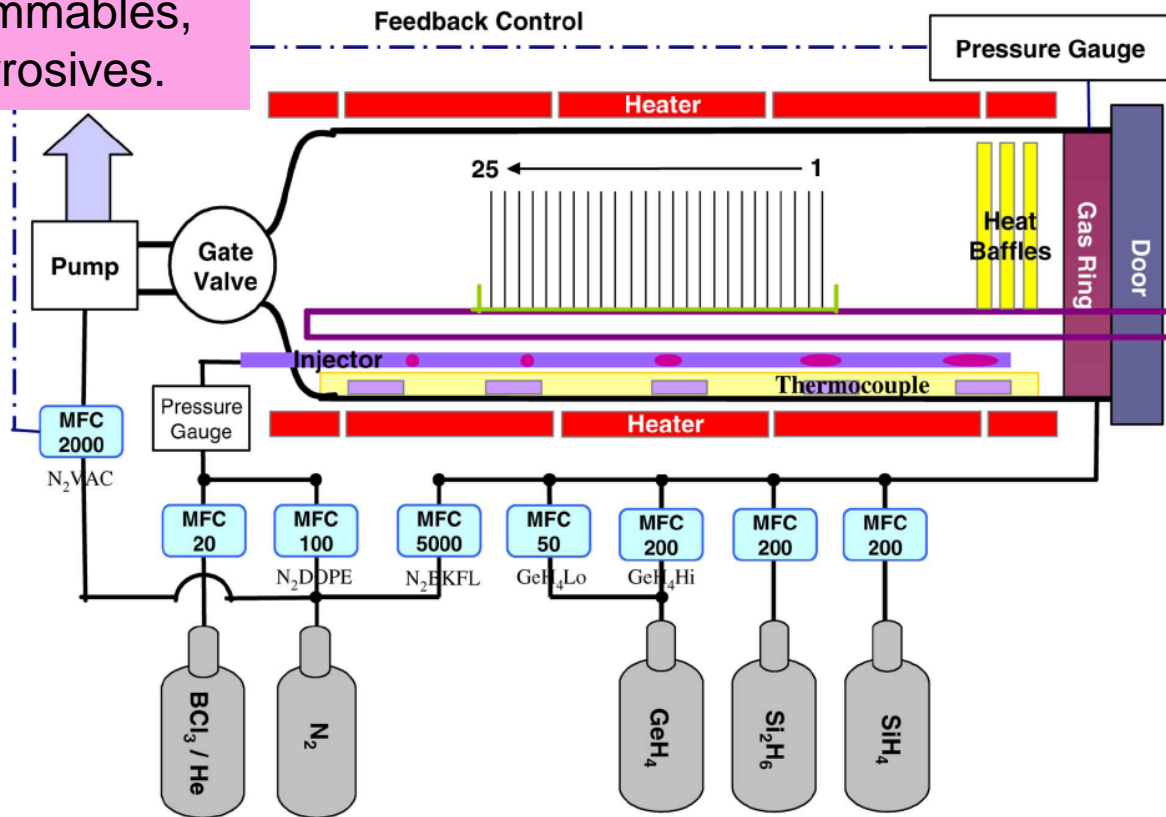


Multiple heater zones ensure flat temperature profile in the central section (maybe little higher in the ends to compensate).

Dummy wafers make sure end of tube thermal losses reduced.

# LPCVD reactor

Gas abatement:  
remove  
toxics,  
flammables,  
corrosives.

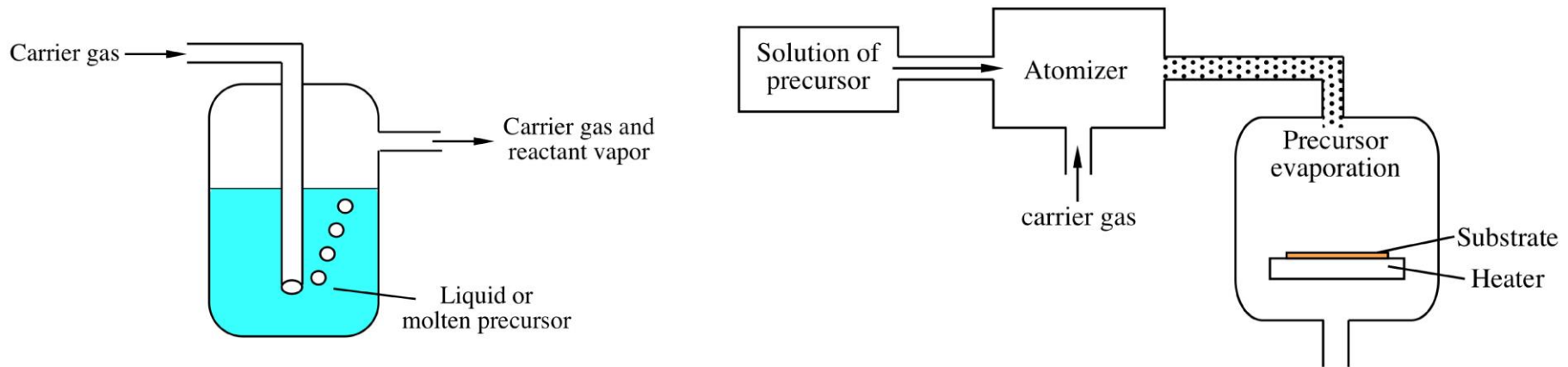
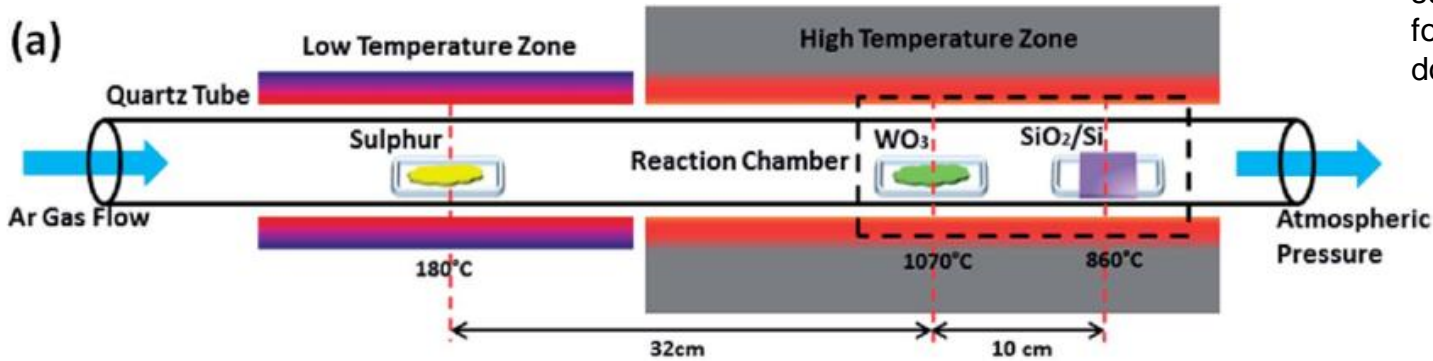


Gas injection:  
from below,  
from hundreds  
of small  
injector holes.

Mixing  
different  
processes in  
the same tube  
leads to cross-  
contamination  
(residues).

# Solid and liquid precursors

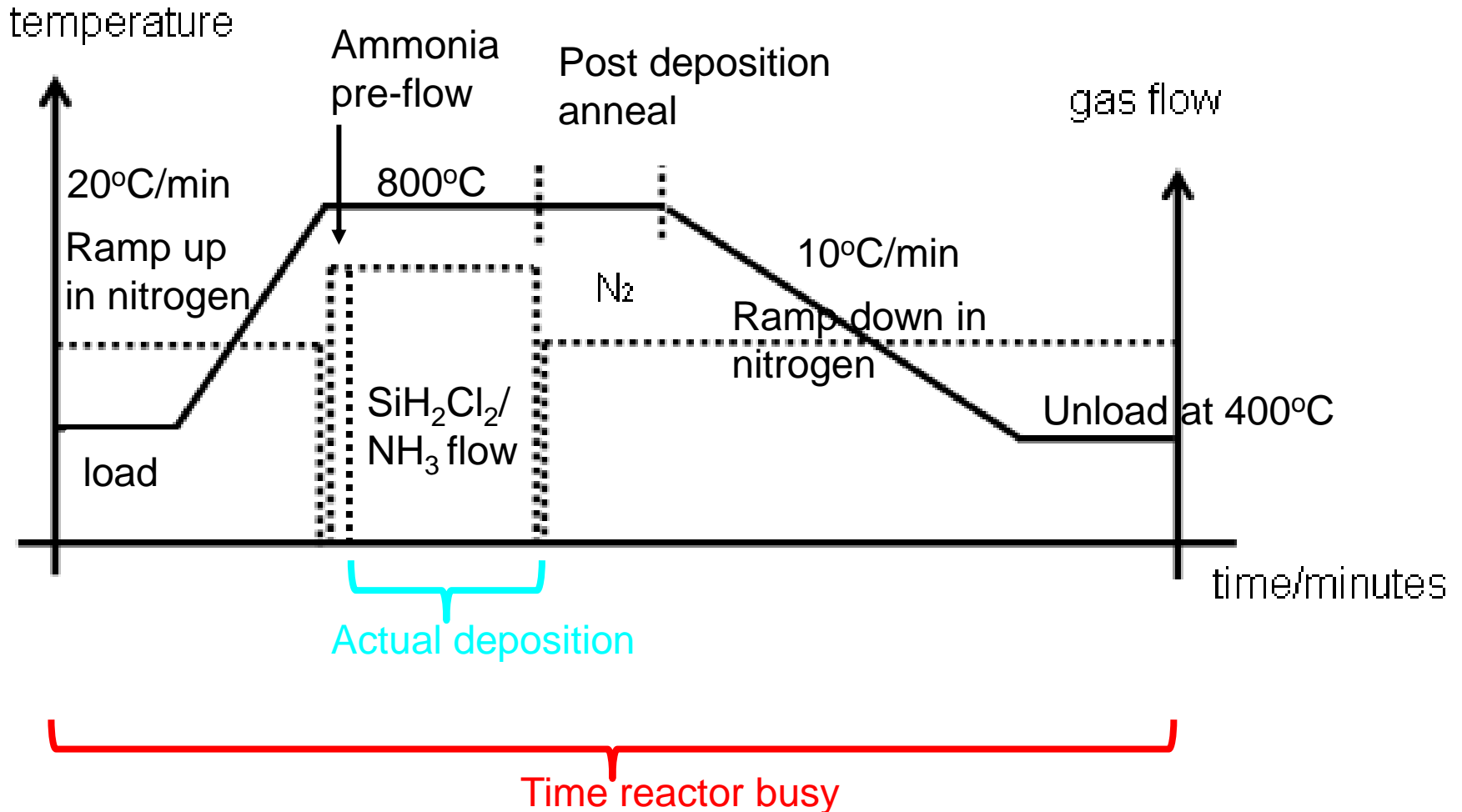
Rong et al: Controlling sulphur precursor addition for large single crystal domains of  $WS_2$ , 2014



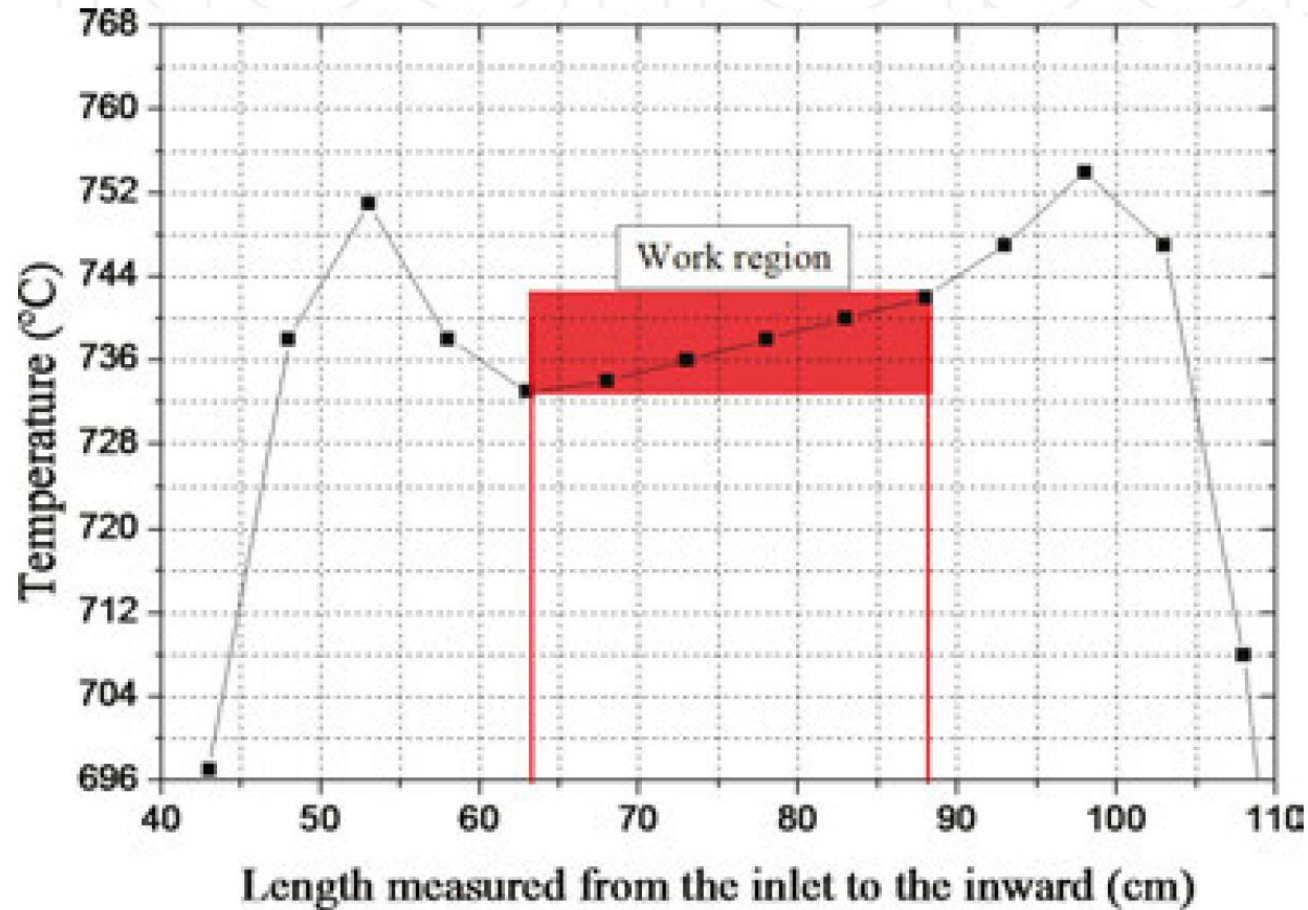
# LPCVD of silicon nitride ( $\text{Si}_3\text{N}_4$ )

- Load the boat, fill with dummy wafers to equalize load and flow patterns.
- Ramp temperature from  $500^\circ\text{C}$  to  $800^\circ\text{C}$  under nitrogen flow, 50 min ( $5^\circ\text{C}/\text{min}$ )
- Pump to vacuum and perform leak check, 2 min
- Introduce ammonia  $\text{NH}_3$ , stabilize flow at 30 sccm, for 1 min
- Introduce dichlorosilane  $\text{SiH}_2\text{Cl}_2$ , flow 120 sccm, deposition starts
- Deposit at 300 mtorr for 25 min (4 nm/min deposition rate)
- Cool down to  $700^\circ\text{C}$  (10 min)
- **Take boat out**
- Monitoring: film thickness and refractive index by ellipsometer

# Total cycle time: LPCVD nitride



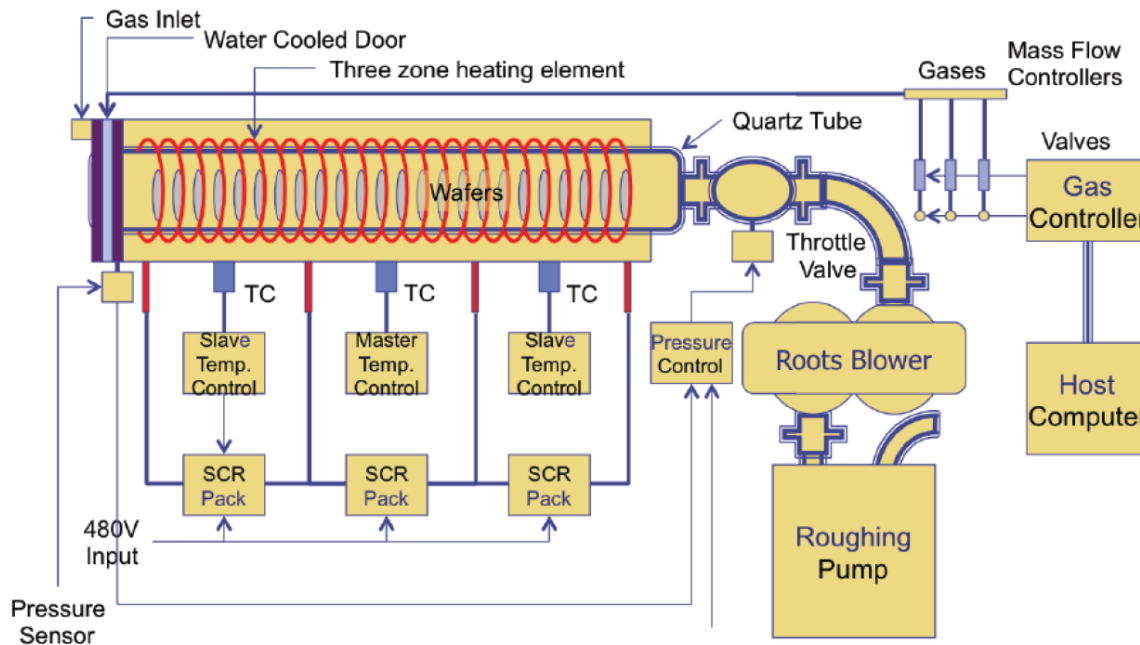
# Real temperature profile



Range 9  
degrees  
!

# Does 3-zone temperature work ?

Heating elements are arranged in three zones: e.g. T1: 747 °C, T2: 750 °C, T3: 753 °C for LPCVD silicon nitride. This temperature ramp compensates for the reactant depletion along the tube.



Amorphous films do not matter.  
Polycrystalline films show grain size gradient if  $\Delta T$ .



# Purity

Source gas issues:

- oxygen and water vapor oxidize silicon
- hydrogen remains in the film
- purity of source gases ? 99.99% vs. 99.99999%
- oxygen reactive and incorporated into film

Furnace issues:

- base pressure ? What is there to begin with?
- leaks: oxygen, water, backstreaming oils
- tube walls experience film deposition, too
- if these films flake off, they “rain” on wafers as particles

# CN-film: PECVD of $\text{NH}_3/\text{CH}_4/\text{Ar}$

XPS results: C, N, O content in films as-deposited and annealed at different  $T_A$

Sample no.	Sputter	C1s (at.%)	N1s (at.%)	O1s (at.%)	N/C ratio
As-deposited	×	66.35	17.80	15.85	0.268
	/	86.45	9.48	4.07	0.110
200 °C	×	73.93	16.10	9.97	0.218
	/	89.29	8.83	1.88	0.099
400 °C	×	77.07	12.01	10.92	0.156
	/	90.02	7.94	2.04	0.088
600 °C	×	82.28	7.42	10.30	0.090
	/	91.61	6.10	2.29	0.067
800 °C	×	85.00	2.80	12.20	0.033
	/	96.73	1.51	1.76	0.016

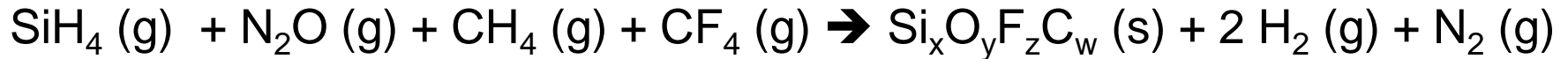
*Note.* Sign ‘×’ and ‘/’ mean without and with  $\text{Ar}^+$  sputter.

Not stoichiometric ( $\text{C}_3\text{N}_4$ , cf.  $\text{Si}_3\text{N}_4$ )

Oxygen is impurity

Argon ion bombardment during deposition seems to remove oxygen efficiently.

# $\text{Si}_x\text{O}_y\text{F}_z / \text{Si}_x\text{O}_y\text{C}_z$ films

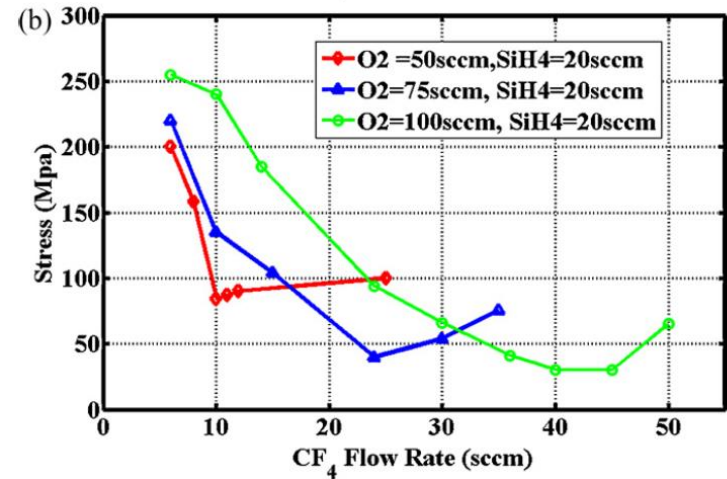
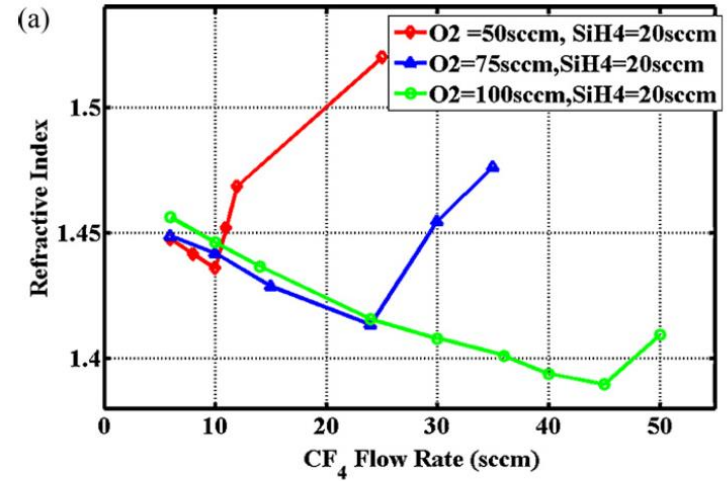
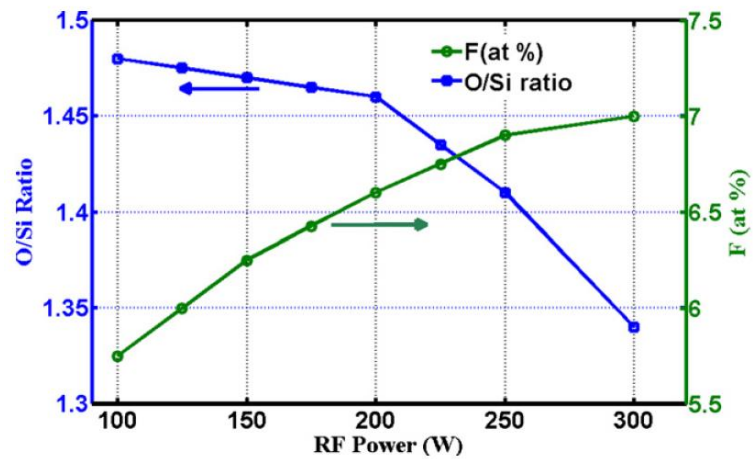
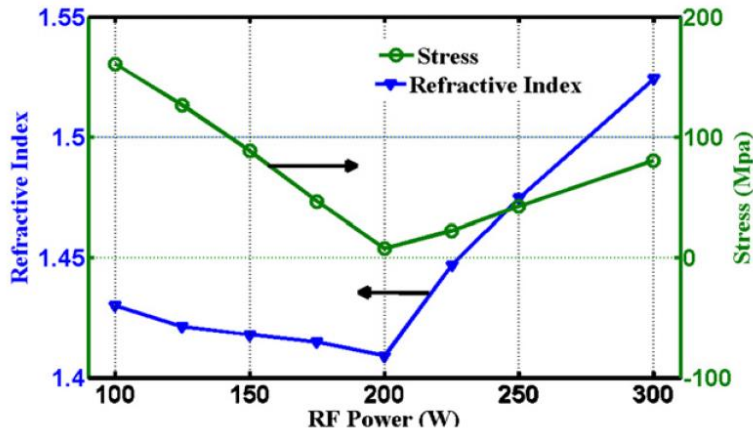


- Low dielectric constants  
(by chemistry and by porosity)
- Refractive index tailoring
- Composition tailoring
- Stress control

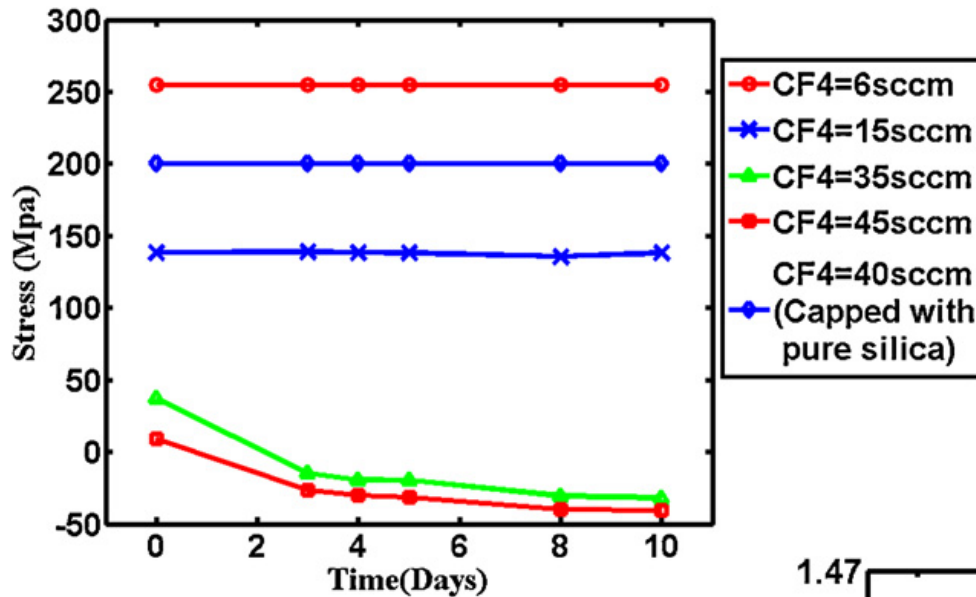
The more variables, the more things to control.

This must be achieved without sacrificing uniformity, reproducibility, stability...

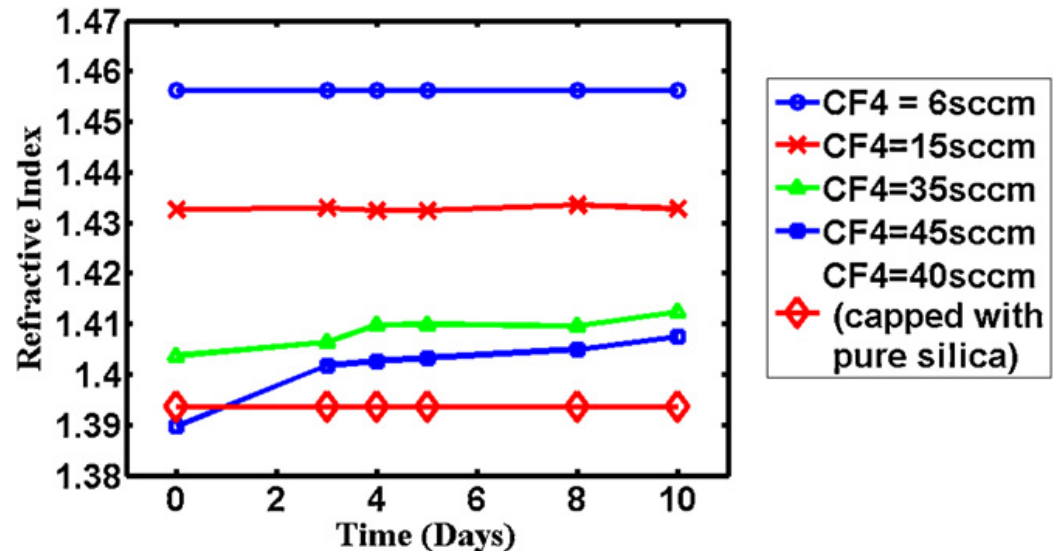
# Si<sub>x</sub>O<sub>y</sub>F<sub>z</sub>-film



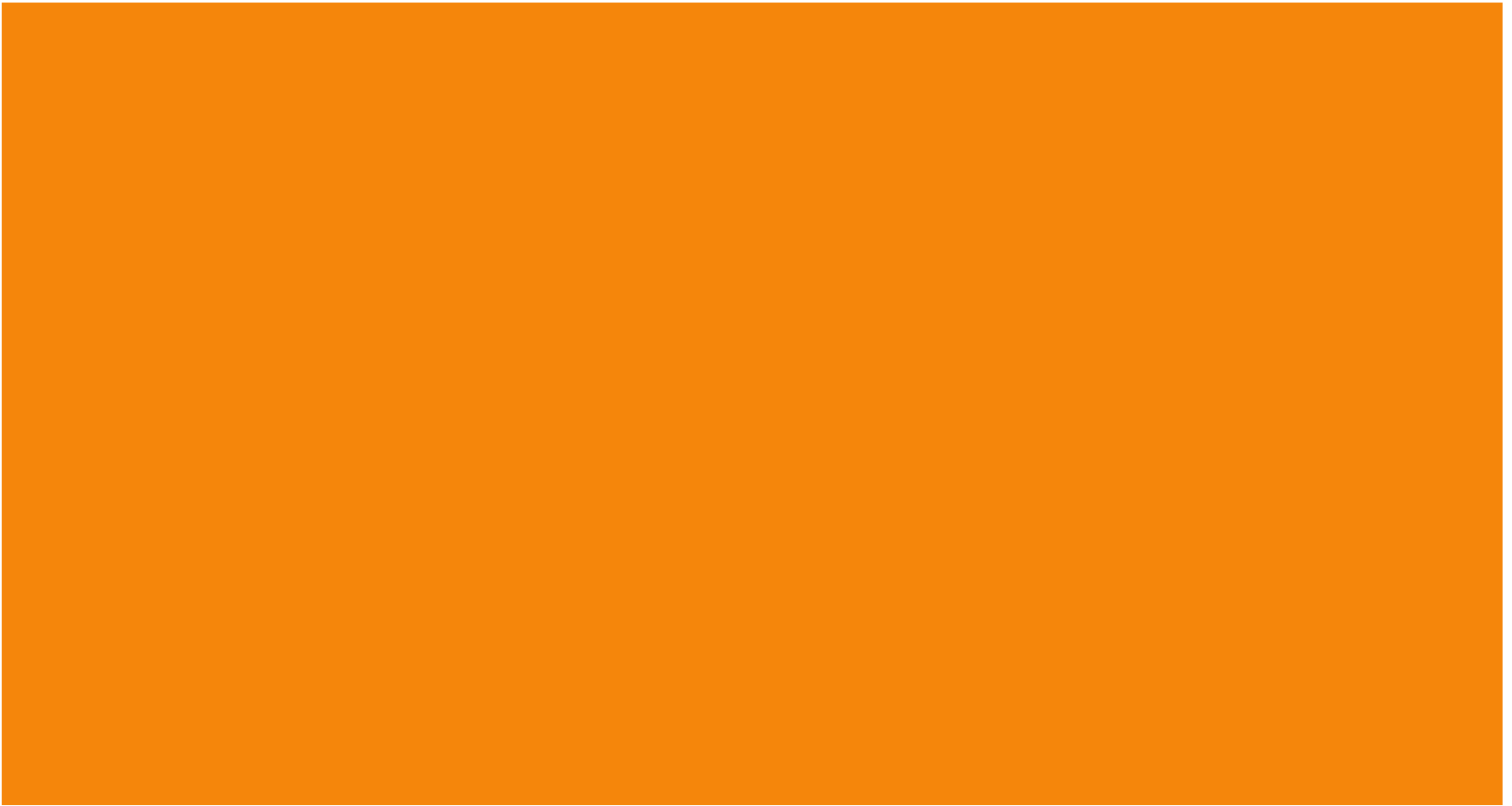
# Si<sub>x</sub>O<sub>y</sub>F<sub>z</sub> film (in)stability



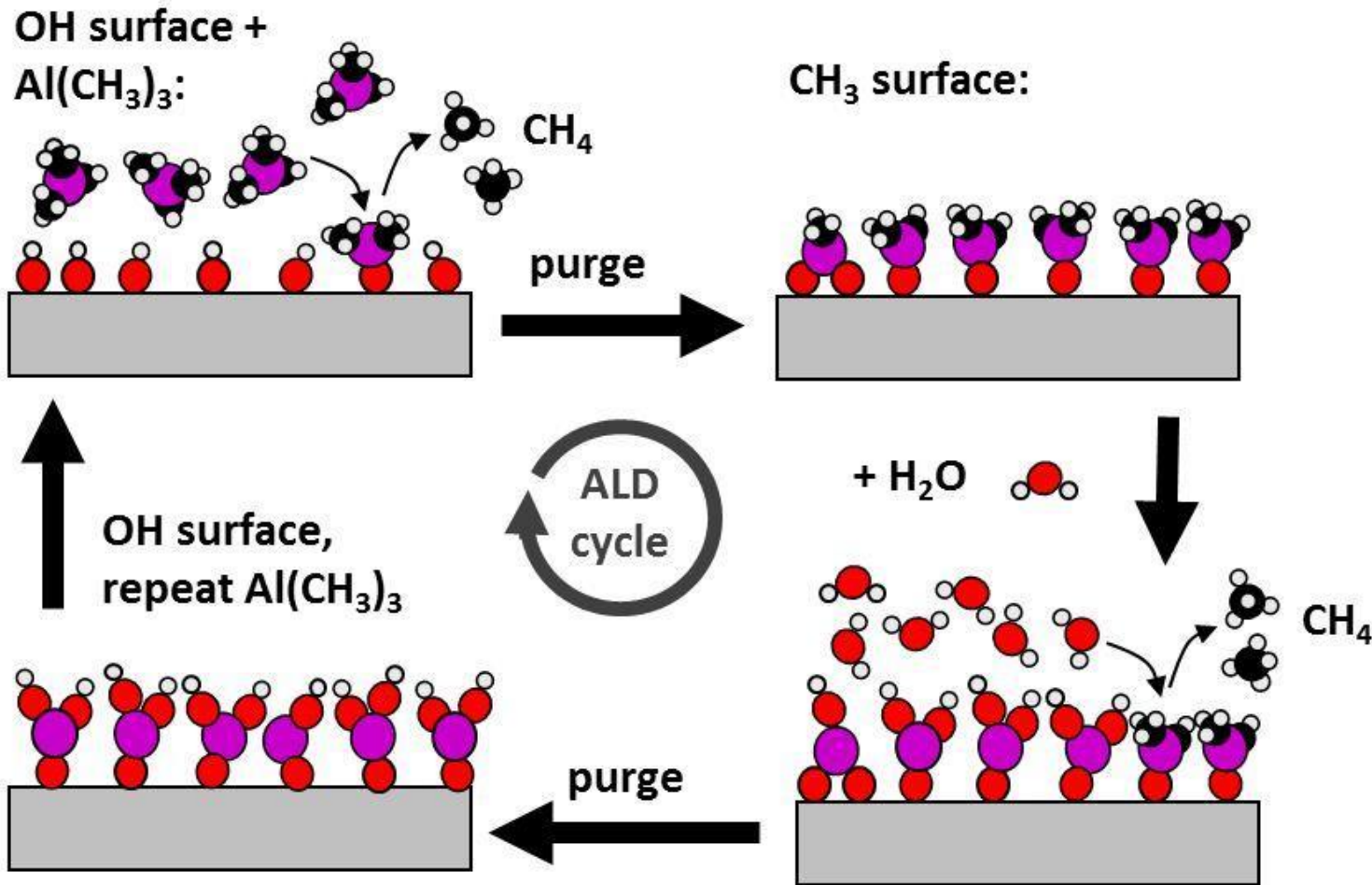
(i) surface effect (due to adsorbed water at the surface),  
 (ii) silanol effect (formation of Si-OH due to diffused water),  
 (iii) adsorbed water dipoles effect



# Half-time

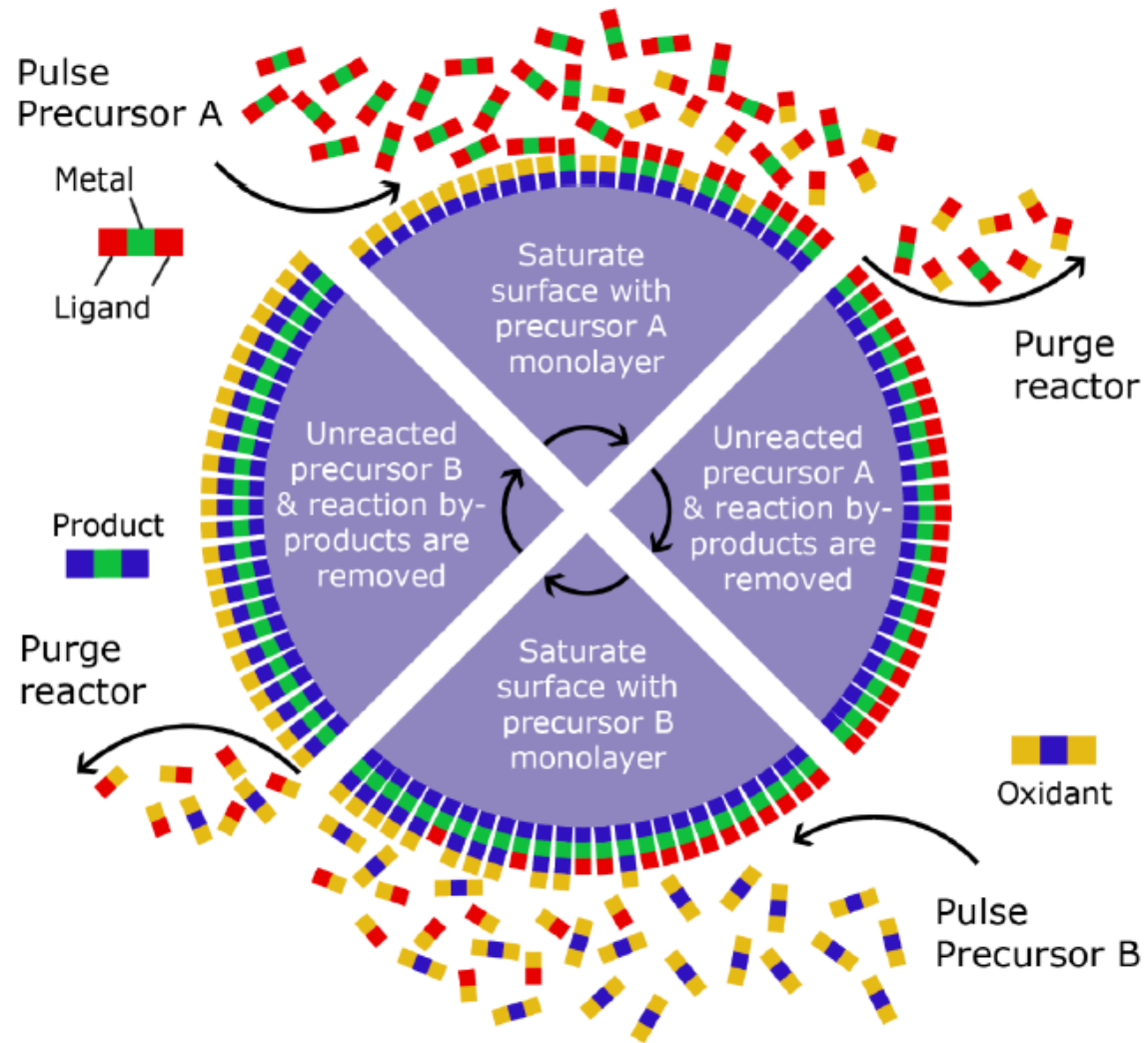


# ALD: Atomic Layer Deposition



Precursors introduced in pulses, with purging in-between

# ALD cycle





# ALD process based on:

## Chemisorption

- Suitable temperature for chemical bonding, no thermal decomposition
- Covalent bonding  $\Rightarrow$  excellent adhesion

## Saturation

- Sufficient dosing of precursor material
- Self-terminating reactions  $\Rightarrow$  extremely precise dosing not required

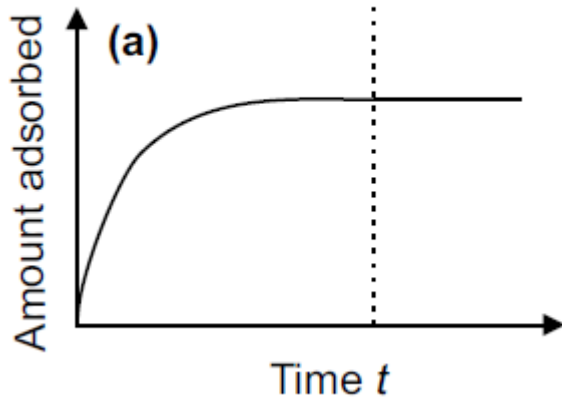
## Surface controlled reactions

- Film thickness is independent of substrate geometry  $\Rightarrow$  conformal film onto deep trenches and 3D structures

## Sequential

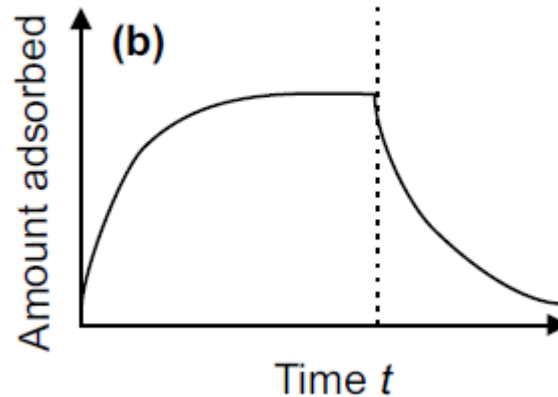
- Digital growth
- Sufficient purging needed between pulses
- Good flow dynamics required to ensure rapid gas changes

# Surface saturation



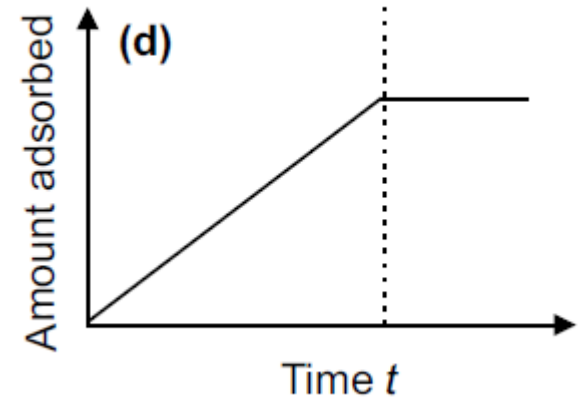
Irreversible saturation  
ALD reactions:

Surface saturates  
with a monolayer of  
precursor, strong  
chemisorption  
(=chemical bonds  
formed)



Reversible saturation:

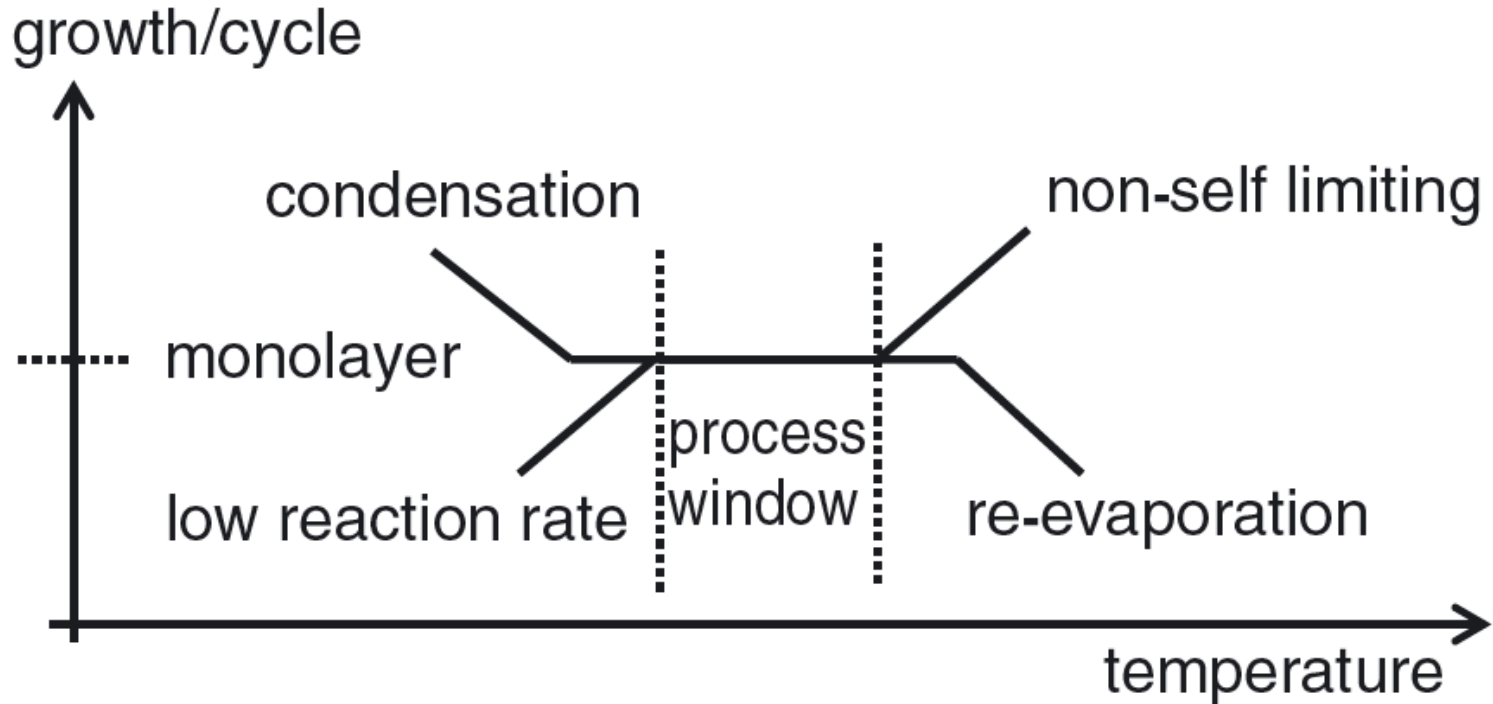
Physisorption only  
(weak bonds like van  
der Waals): once  
precursor flux is  
stopped, surface  
specie will desorb.



Irreversible non-  
saturating.

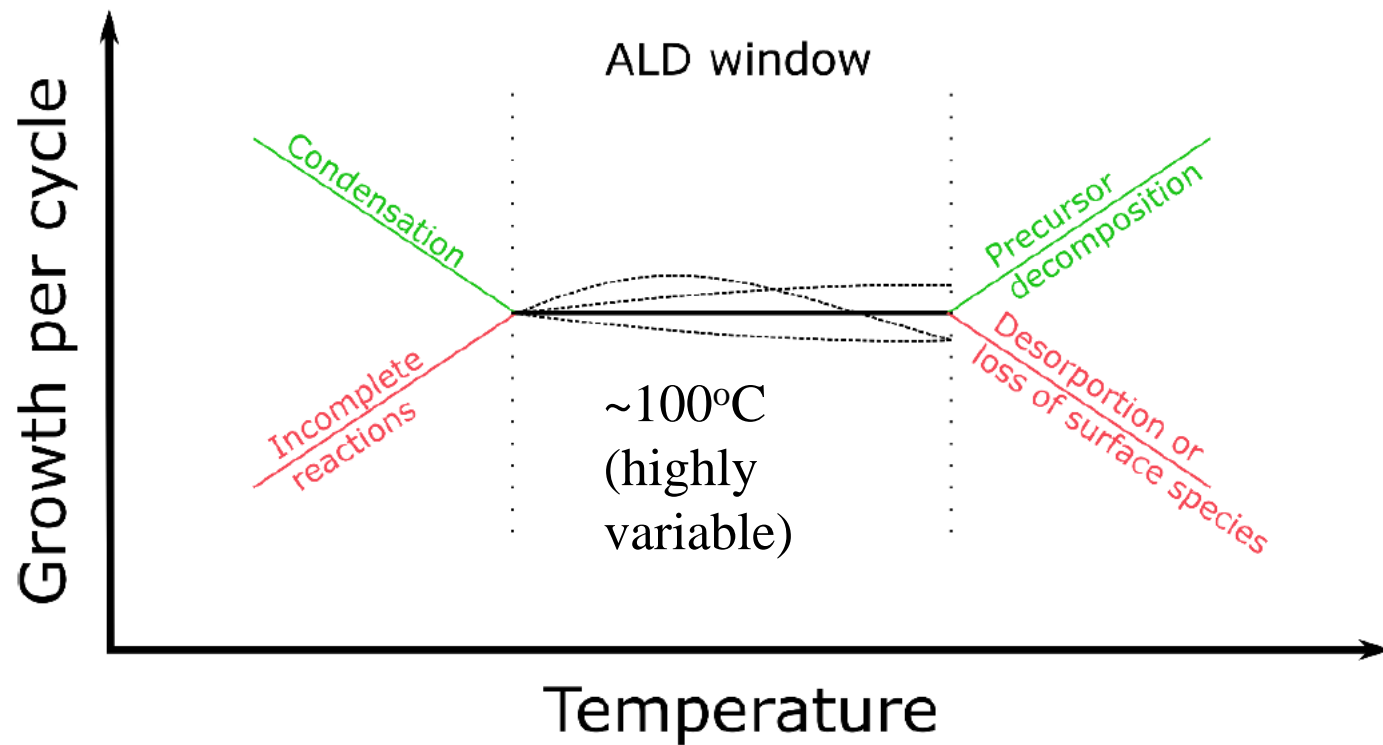
CVD regime:  
more reactants in,  
more film is  
deposited  
(continuosly)

# ALD window

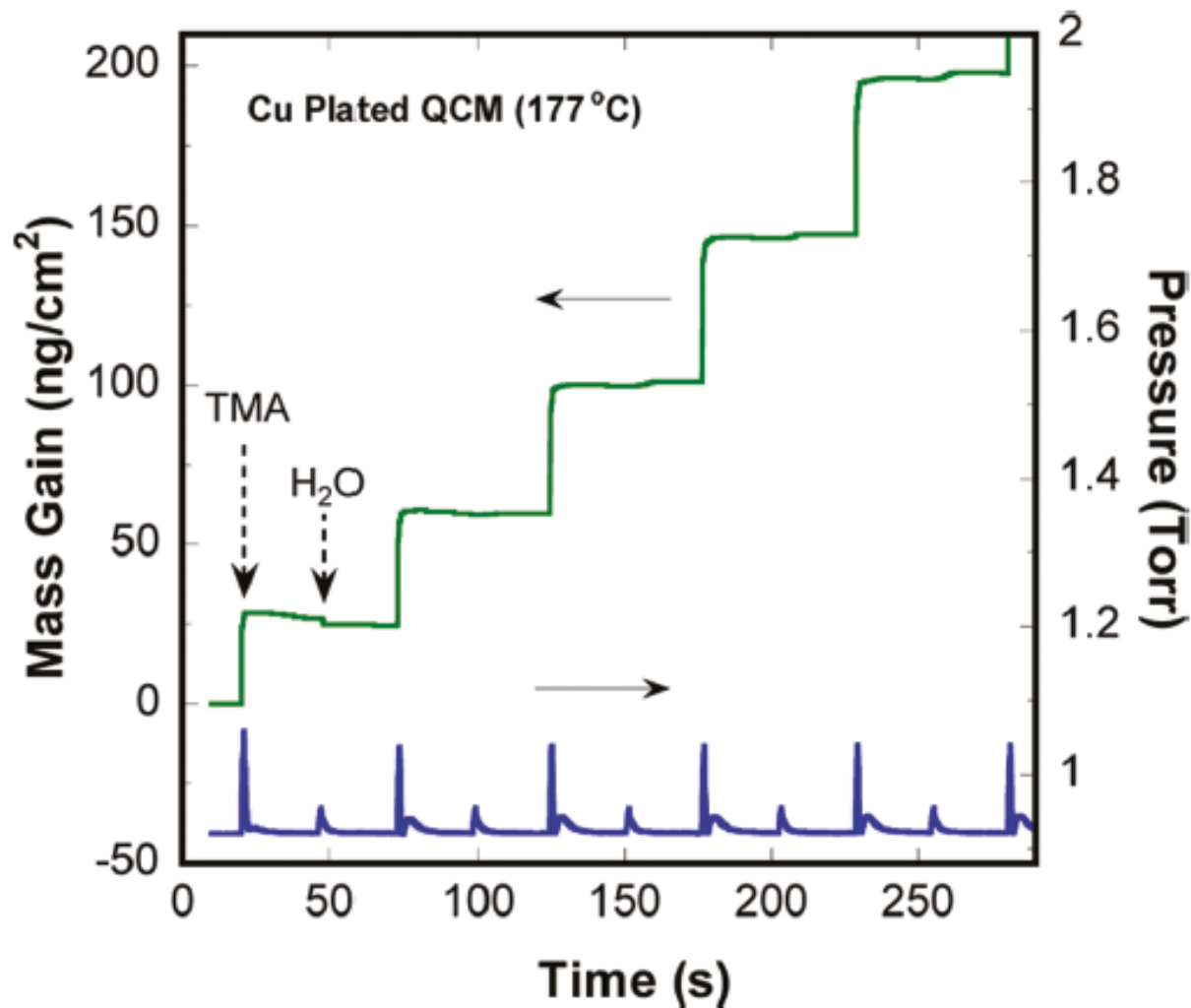


Compare with CVD: exponential (Arrhenius) temperature dependence.

# ALD window realistically



# Growth per cycle (GPC)



Abdulagatov & S.M. George:  
ACS Appl. Mater. Interfaces  
2011, 3, 4593–4601

# How are ALD processes developed ?

Precursor considerations

Finding ALD window (pulse parameters)

GPC and deposition rate

Characterizing films for

- Uniformity
- Continuity
- Conformality (step coverage)
- Impurities
- Crystallinity

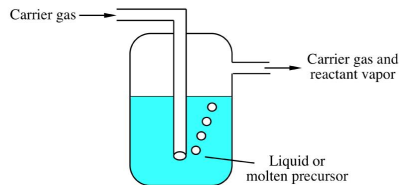
# Precursor considerations

Gas precursor dose/flow is easy to measure via mass flow controllers.

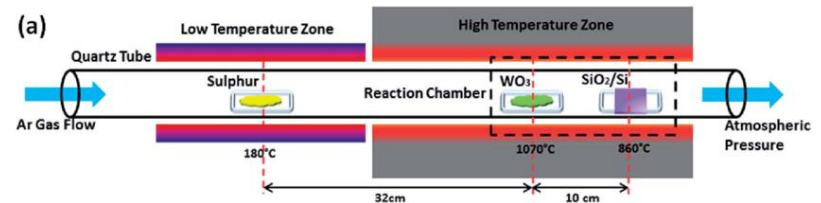
Liquid and solid precursors are also used: they are vaporized by bubbling and/or heating (similar to CVD).

Source temperature then becomes one reactor parameter to optimize.

It is important to all precursors that they do not prematurely dissociate.



Liquid bubbler  
Solid source

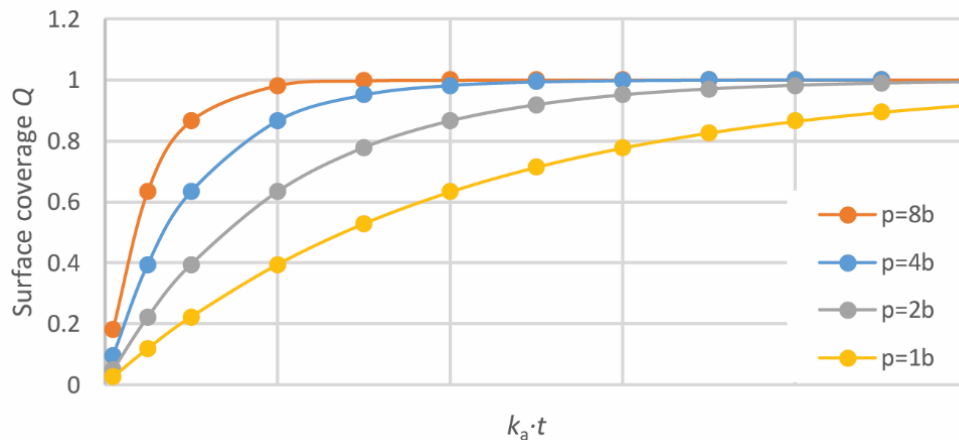


# Surface coverage

## Irreversible chemisorption case

$$Q = 1 - e^{-k_a p t}$$

$Q$  is chemisorption coverage,  $k_a$  reaction rate coefficient,  $p$  precursor partial pressure and  $t$  time.



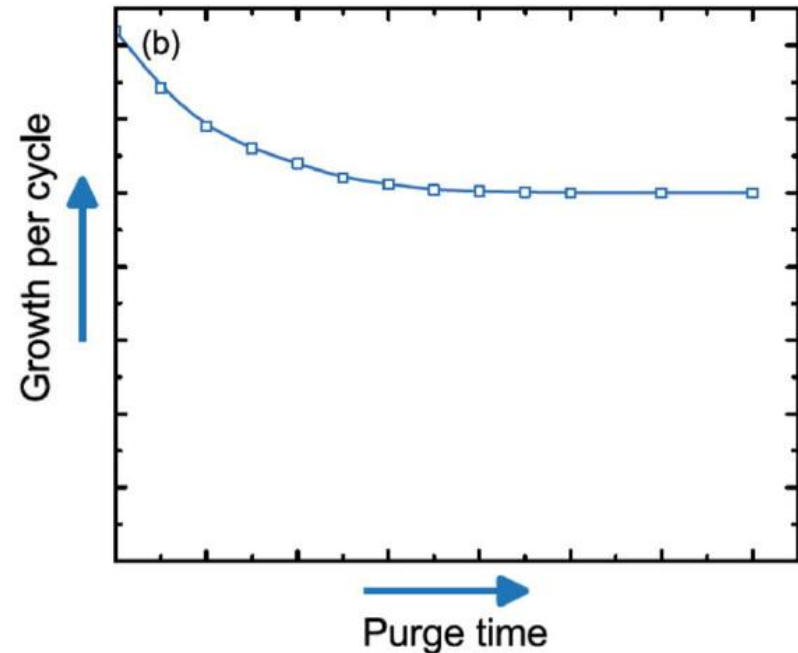
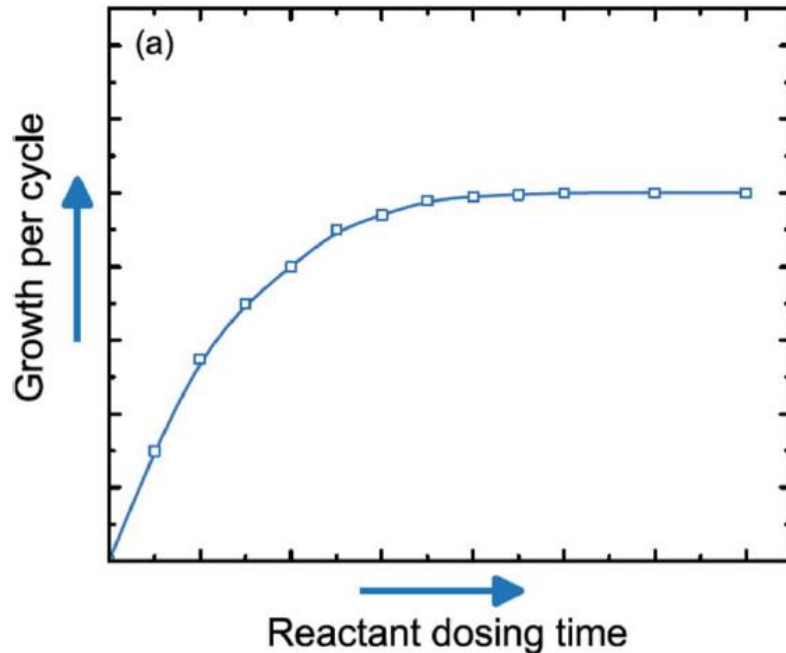
Increasing precursor vapour pressure decreases reaction time needed to reach saturation (but might increase purge pulse duration).



# Start with long pulses

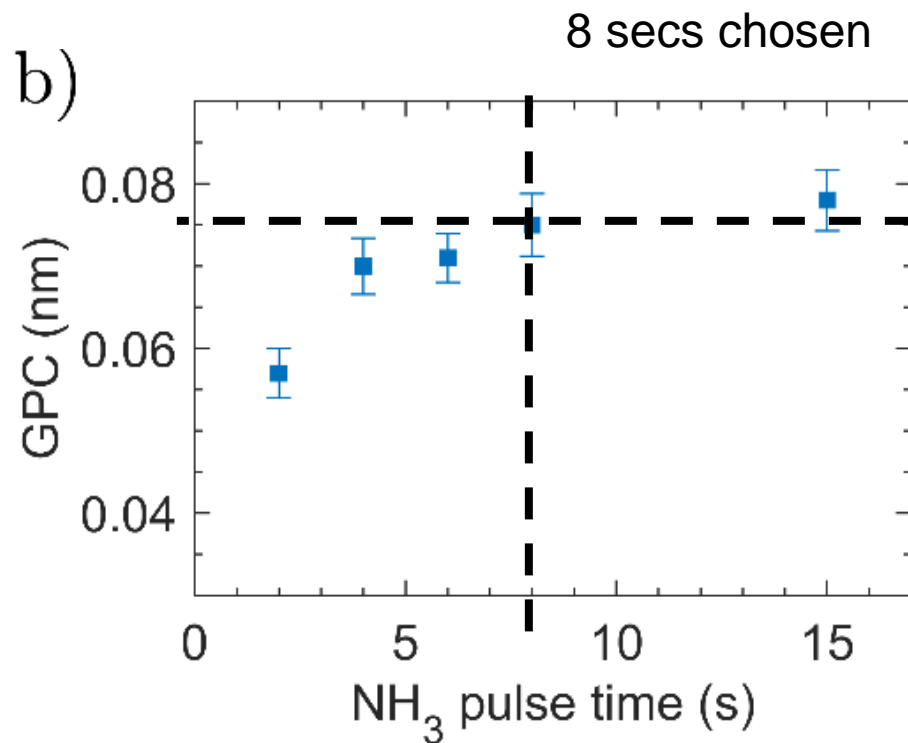
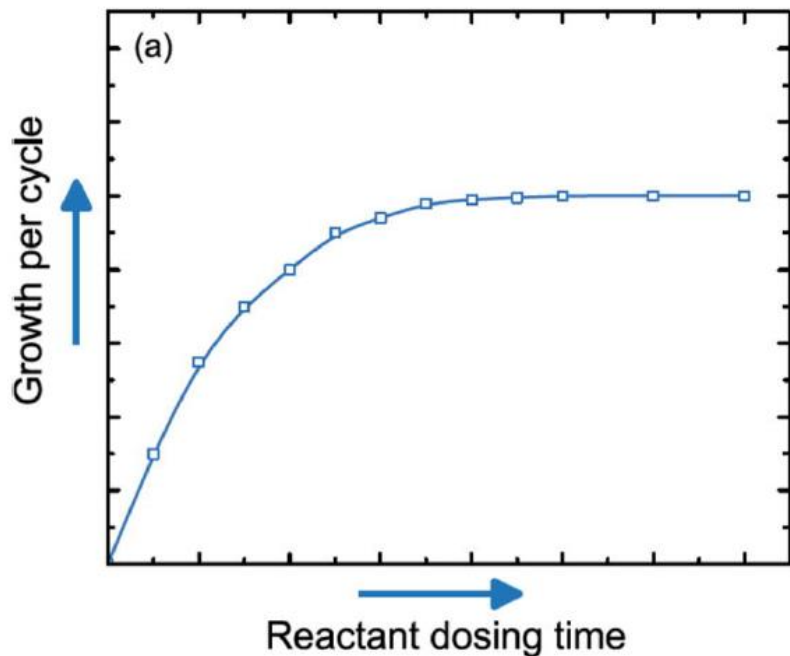
“Initial process development should be performed with long purges to ensure that they are sufficient. When the purge time is too short, precursor pulse A and B are no longer separated from each other and gas phase reactions may occur leading to CVD type deposition. This CVD type growth increases apparent GPC, but the film quality is not as good and the uniformity suffers, which can be detrimental for the application. It can also make it more difficult to optimize the process. Once the saturating pulsing parameters have been established, it is easy to decrease the purge times. The moment when the purge time becomes too short, manifests itself as an increase in GPC and a deteriorated uniformity.”

# Finding the pulse times



Find the plateau (← evidence of self-saturating reaction)

# Theory vs. practice



Growth saturation graphs  
for AlCl<sub>3</sub> and NH<sub>3</sub>.

# Purge pulse length

Must remove both unreacted precursor and reaction product

Becomes slower as temperature is reduced (volatility reduced)

Water vapor is difficult, esp.  $<100^{\circ}\text{C}$ ;  
 $\text{O}_2$  and  $\text{O}_3$  easier to pump away

If high aspect ratios or porous materials are coated, pulse lengths increase because of geometry restrictions.

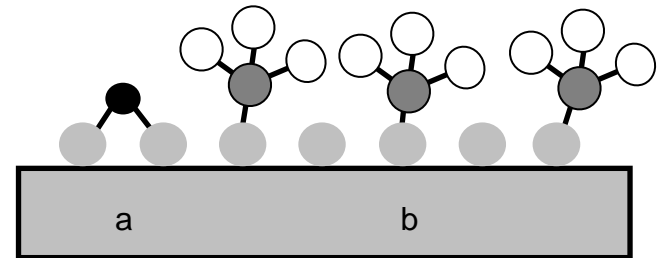
# Deposition rate

Basically one atomic layer per pulse

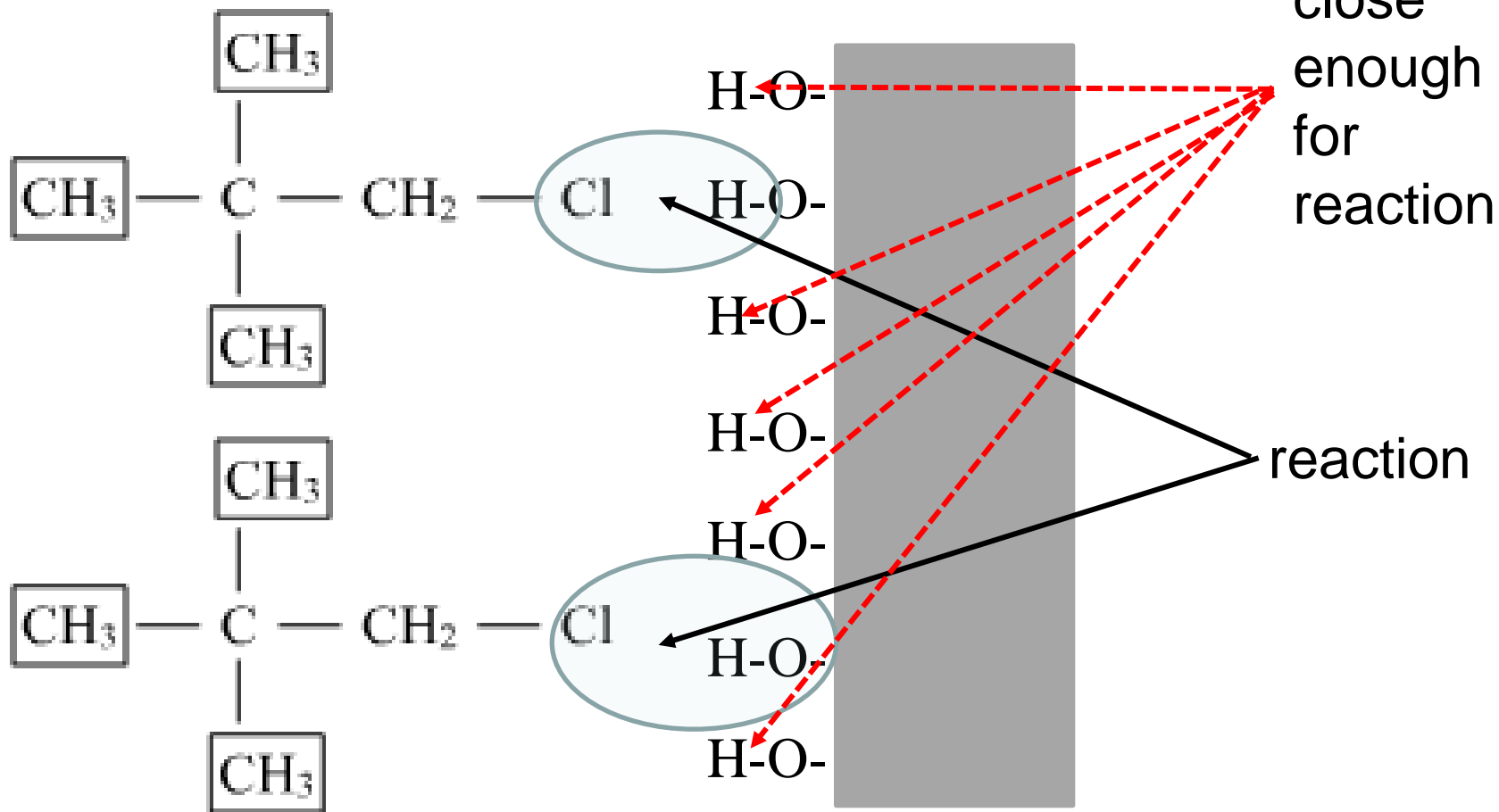
In practise less than an atomic layer because:

a) Inactive surface sites

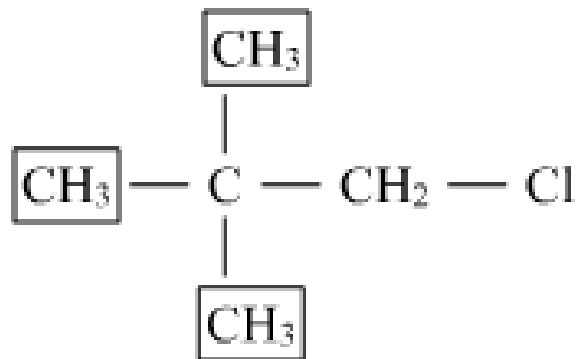
b) Steric hindrance: a large precursor molecule prevents another precursor molecule from approaching the reactive site



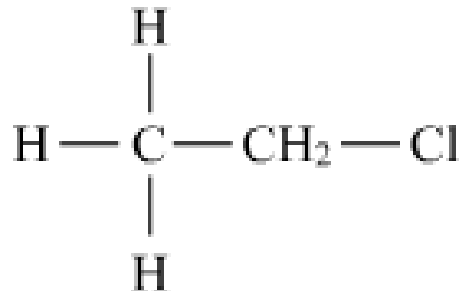
# Steric hindrance



# Precursor design



Large size,  
Steric hindrance



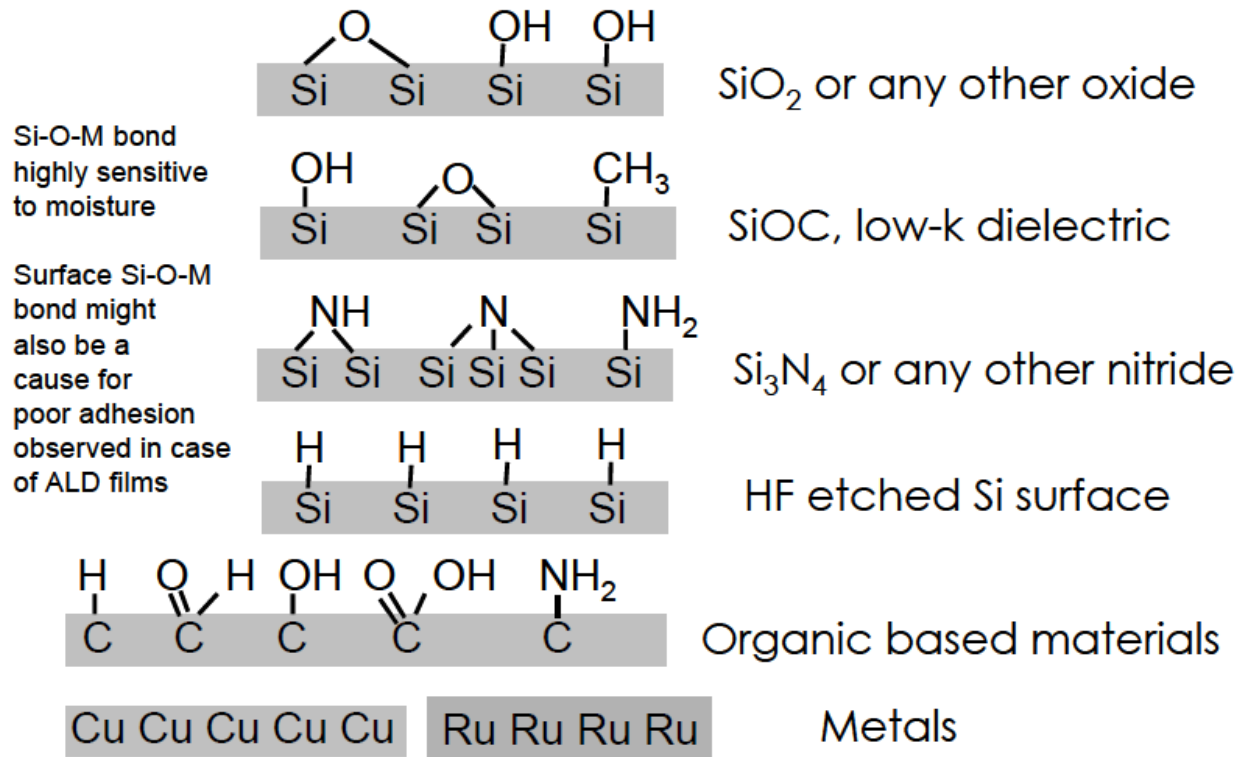
Small size,  
Less steric hindrance.

But you also need to consider:

- thermal stability
- vapour pressure
- toxicity
- price...

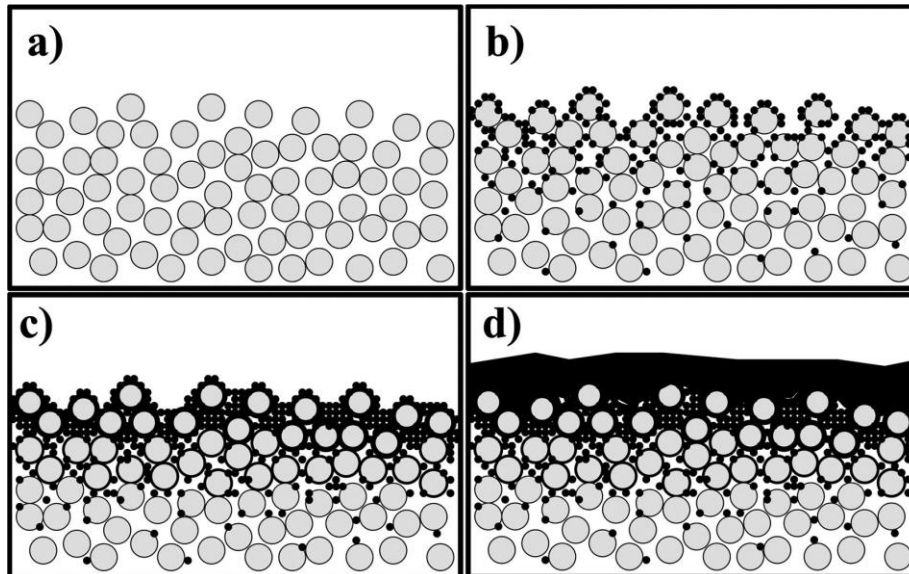
# Available growth sites

Surface preparation step must provide surface with chemically active sites that can react with precursors.





# ALD on polymer substrate



No suitable bonds available.

TMA precursor debris acts as a starting surface.

Time lag in initiation.

- (a) a cross section of the polymer chains at the surface of the polymer film,
- (b) Al<sub>2</sub>O<sub>3</sub> nucleation clusters formed from H<sub>2</sub>O reaction with TMA trapped in the near surface region,
- (c) coalescence of Al<sub>2</sub>O<sub>3</sub> clusters and closure of the space between the polymer chains
- (d) formation of a dense Al<sub>2</sub>O<sub>3</sub> film that grows on top of the polymer surface.

A misconception often encountered in the literature is that the growth of a full monolayer of  $MZ_x$  per cycle would correspond to “ideal ALD”, and growth of less than a monolayer per cycle would correspond to “non-ideal ALD”. It must be emphasized that, for growth from compound reactants, no requirement of a full monolayer growth can be deduced from the basic principles of ALD. In contrast, the theoretical application examples of this work and the real application examples reported elsewhere,<sup>[22]</sup> all done under the assumption of ideal ALD, show that achieving a growth per cycle of even half of a monolayer is rare.

# ALD $\neq$ one atomic layer/cycle

$\text{Al}_2\text{O}_3$       0.9-1.1 Å/cycle (TMA +  $\text{H}_2\text{O}$ )

$\text{TiO}_2$       0.5-0.9 Å/cycle ( $\text{TiCl}_4$  +  $\text{H}_2\text{O}$ )

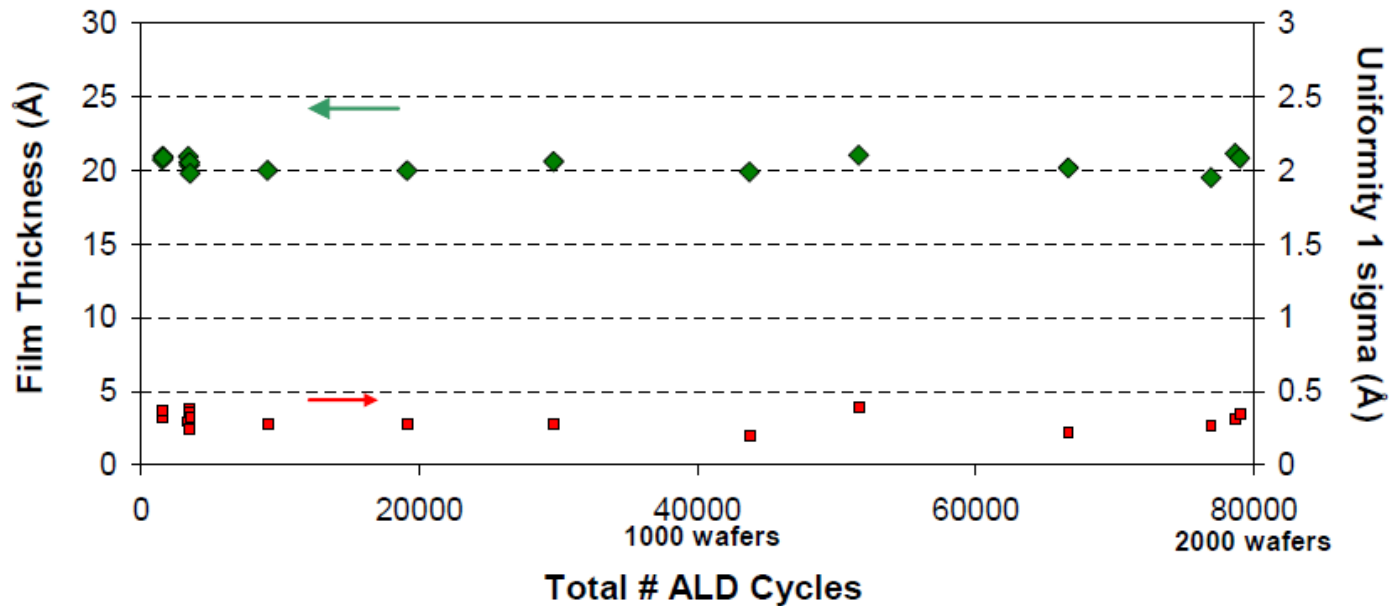
$\text{Y}_2\text{O}_3$       0.24 Å/cycle ( $\text{Y}(\text{thd})_3$  +  $\text{O}_3$ )

If pulses are one second  $\rightarrow$  15\*monolayer thickness/minute  $\sim$  2 nm/min

If 0.1 second pulses  $\rightarrow$  20 nm/min max.

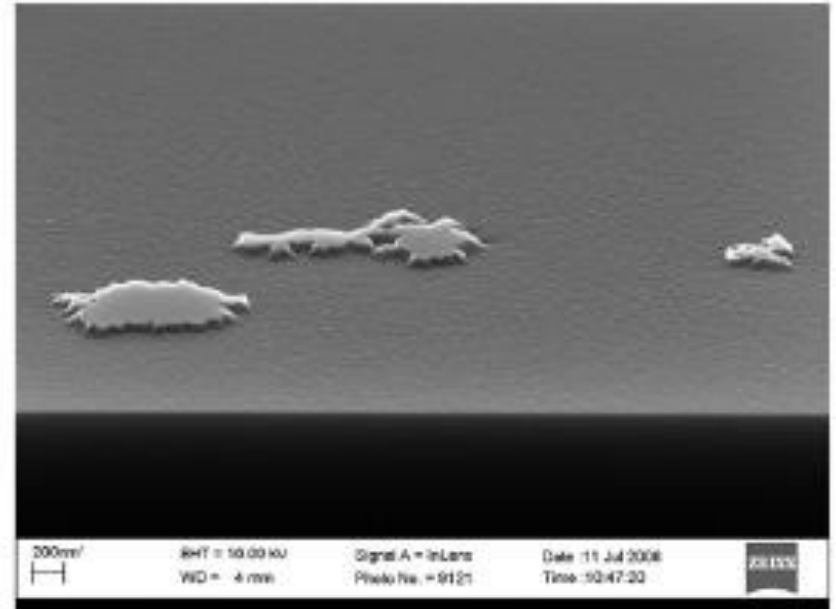
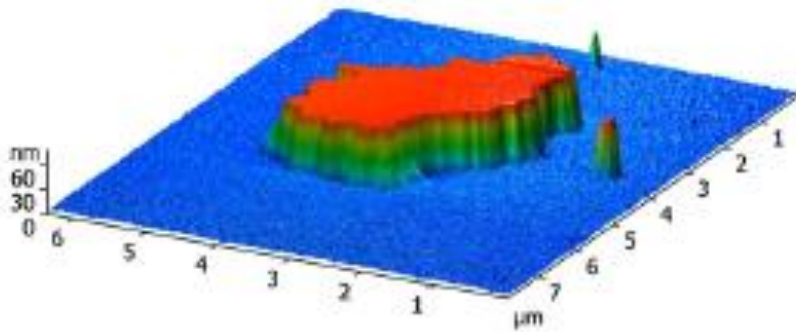
# ALD uniformity (=thickness across the wafer)

Good uniformity is a proof of surface controlled reaction.



HfO<sub>2</sub> dielectric marathon test: 2000 wafers

# Film continuity



$\text{Al}_2\text{O}_3$  of 5 ALD cycles and etched for 2 s cryo-DRIE:  
discontinuous film does not protect underlying silicon.

# Film continuity (2)

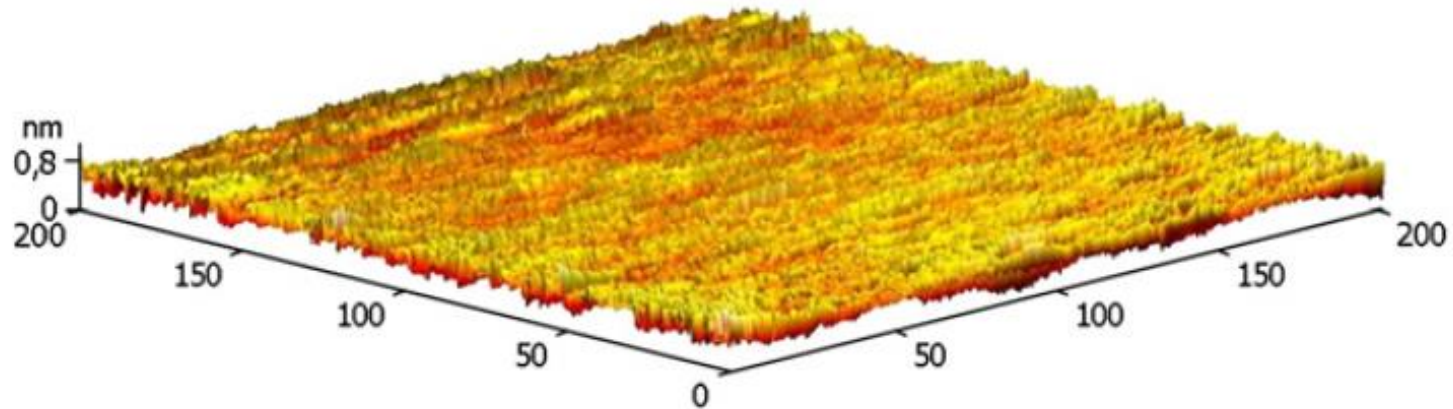
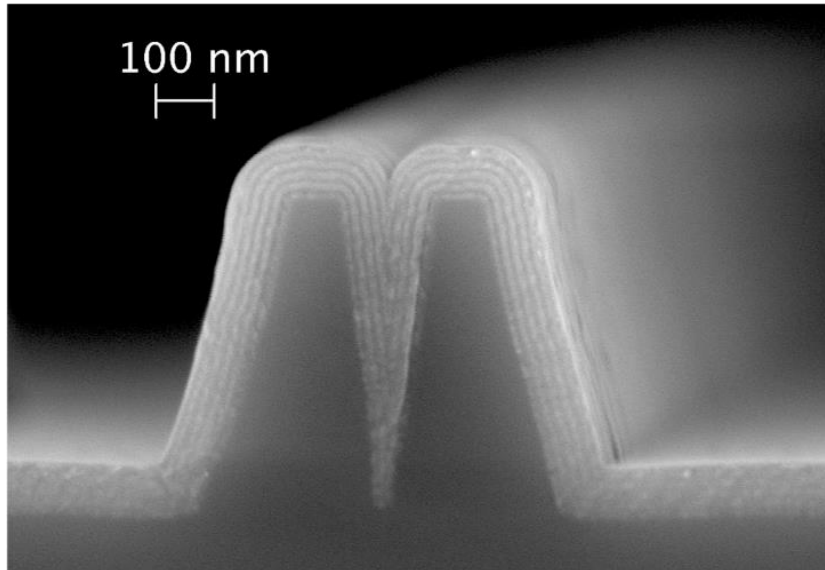


Fig. 4. AFM scan of the structure grown during 10 cycles and etched for 2 s in cryo-DRIE.

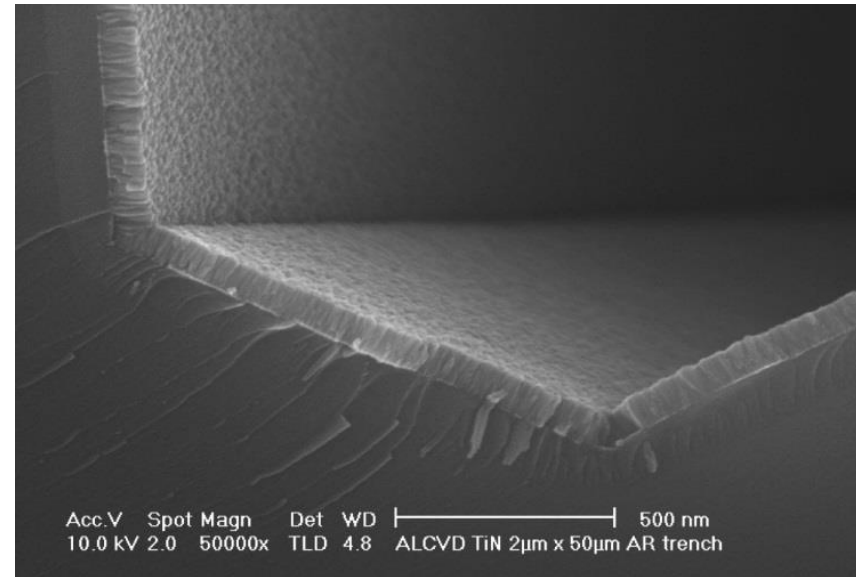
10 cycles ( $\approx 1$  nm) ALD film is continuous and serves as a protective coating.

# ALD conformality is excellent (=step coverage 100% in microstructures)

All surfaces coated by diffusing gaseous precursors in the surface reaction limited mode.

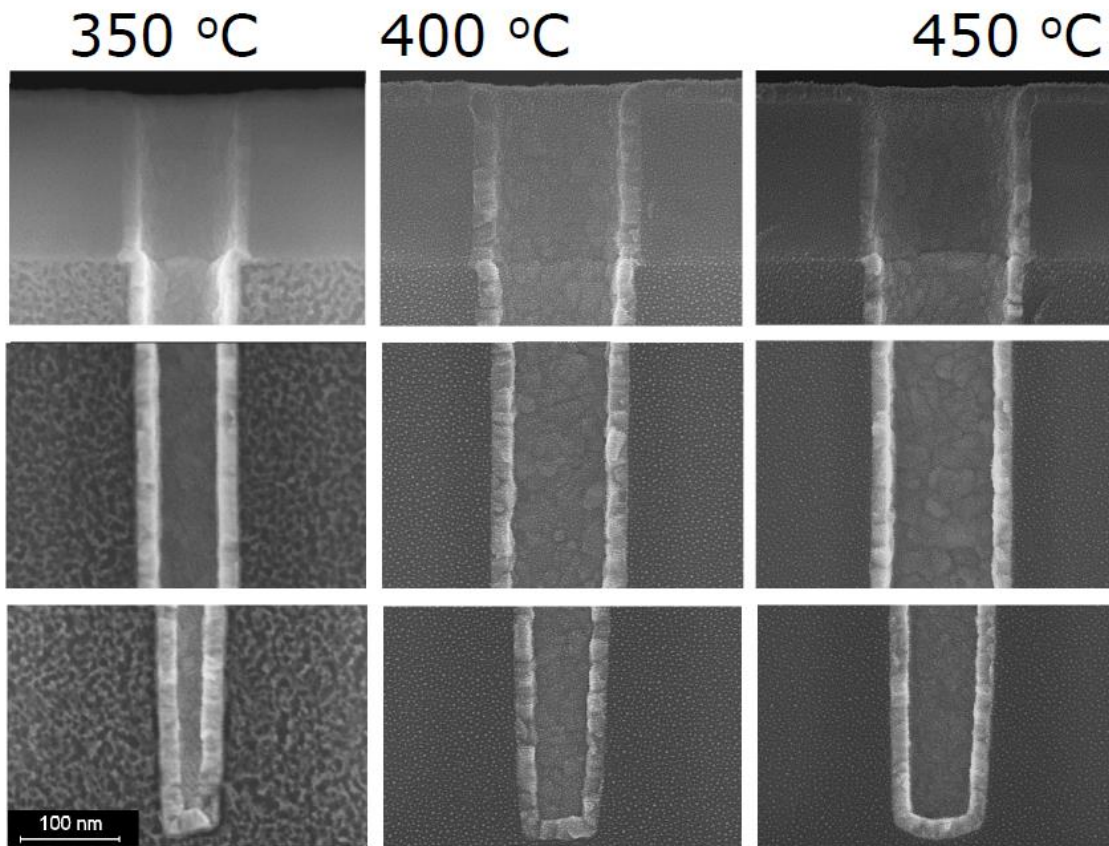


$\text{Al}_2\text{O}_3/\text{TiO}_2$   
nanolaminate



TiN barrier

# Step coverage (2)



Step coverage  
good also in high  
aspect ratio  
grooves,

BUT pulse lengths  
have to to be  
increased

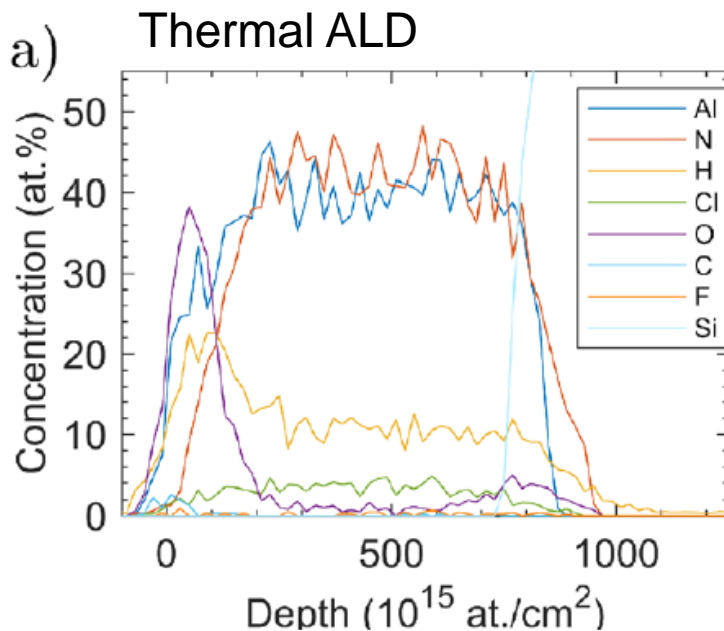
(in coating porous  
materials, pulses  
last for minutes !!).



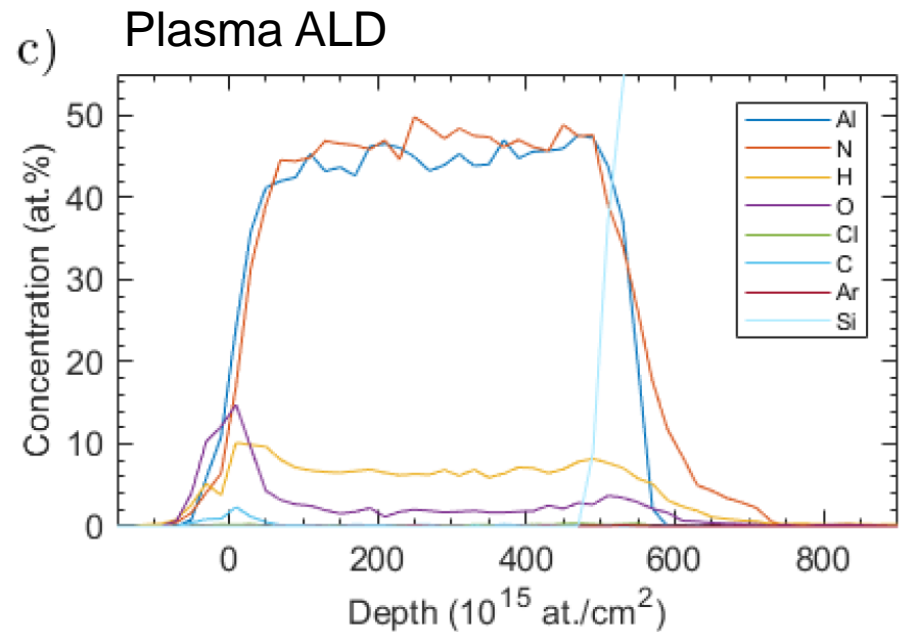
# AlN purity by ToF-ERDA

$\text{AlCl}_3 + \text{NH}_3$  process

Oxygen contamination on surface is post deposition

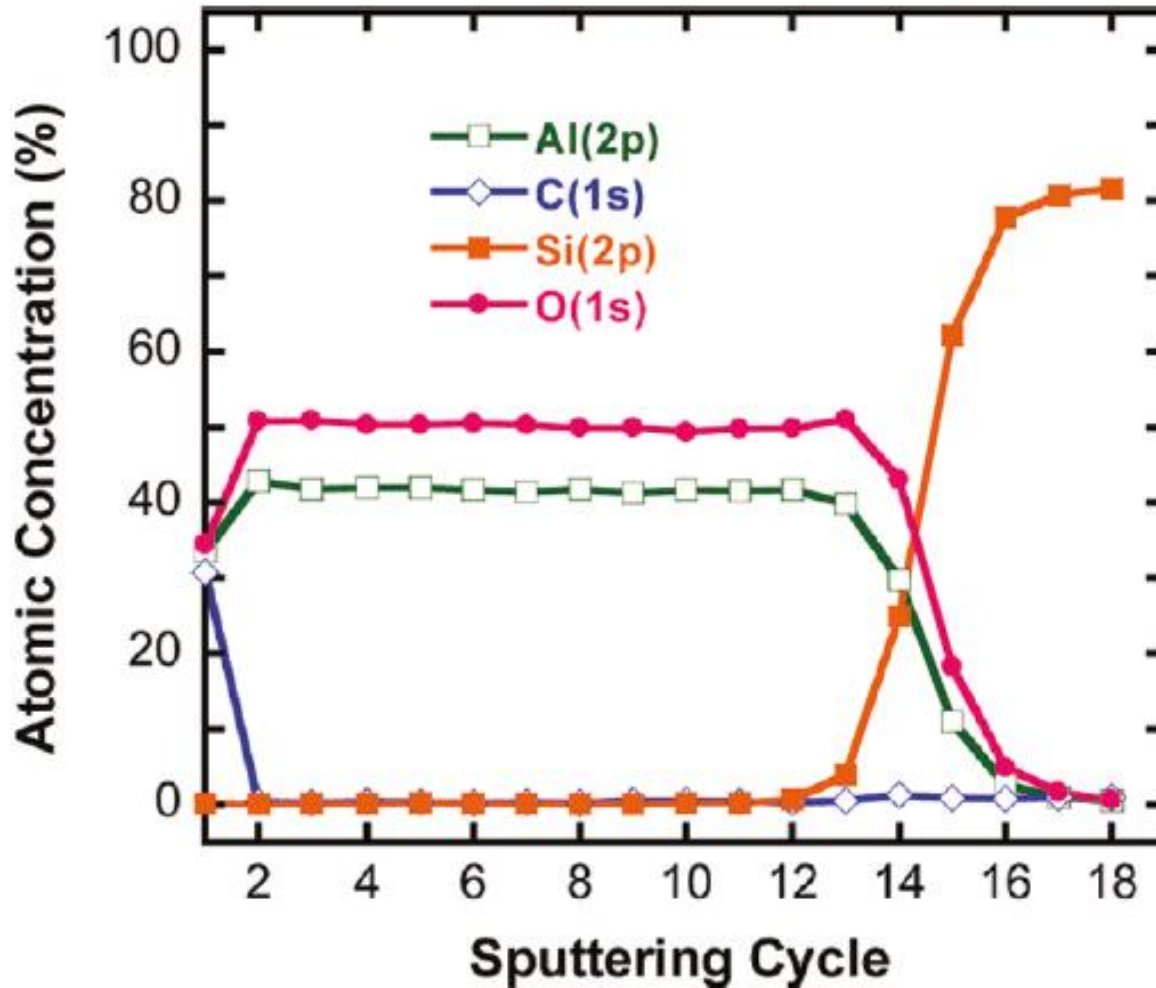


Thermal has much more hydrogen and chlorine.



Overall purity better in PEALD

# Al<sub>2</sub>O<sub>3</sub> purity by SIMS



Carbon on surface is most likely post deposition contamination.

# Crystalline structure: amorphous vs. polycrystalline ?

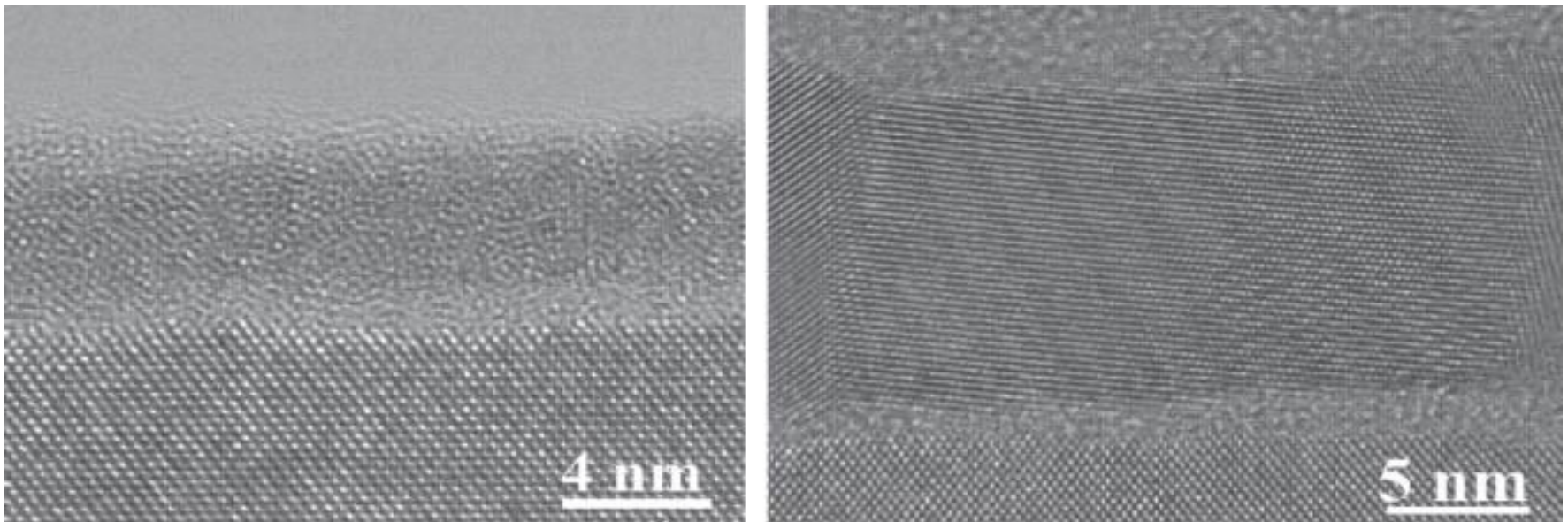
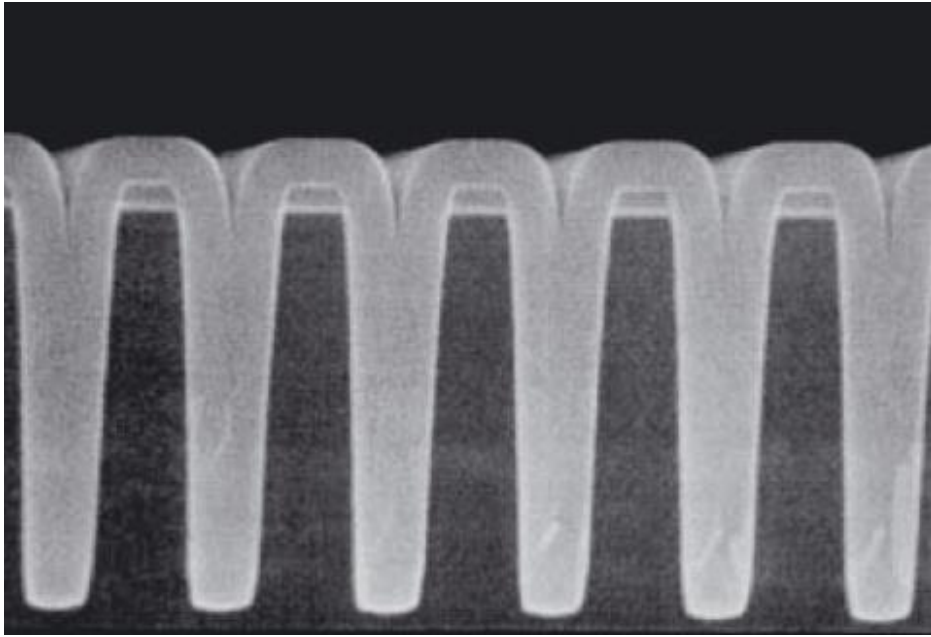


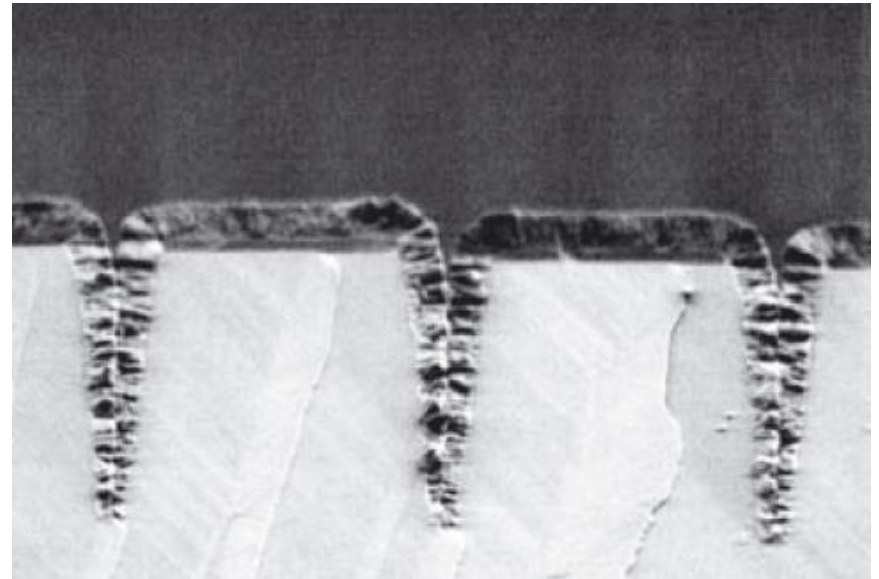
Figure 26-13

ALD  $\text{ZrO}_2$ : the 4 nm thick film is amorphous but the 12 nm thick film is polycrystalline. Reproduced from Kukli *et al.* (2007), copyright 2007, Elsevier.

# Crystallinity (2)



amorphous aluminum oxide



Polycrystalline strontium titanate

Vehkamäki *et al.* (2001)

# Crystallinity: precursors

Metalorganic precursors likely to leave carbon residues  
→ amorphous

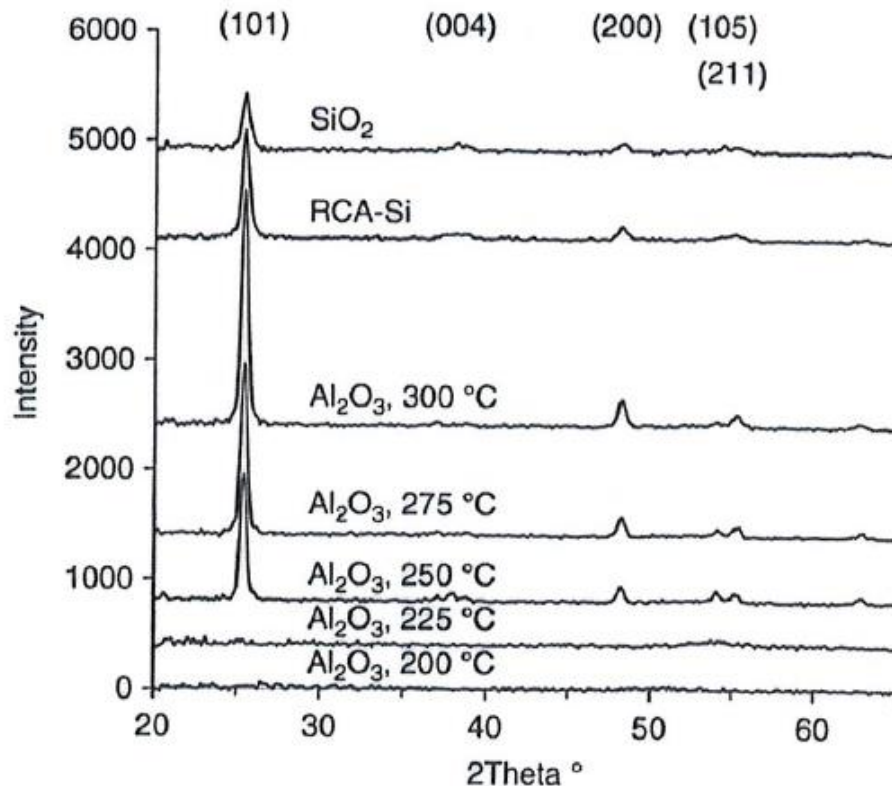
Halide-processes → crystallinity likely

$O_2$  or  $O_3$  more reactive than  $H_2O$  → ligand removed  
more efficiently → less contamination → crystalline

Plasma removes residues → crystallinity more likely

H and Cl are "benign" residues, carbon is "bad".

# Crystallinity & underlying substrate



TiO<sub>2</sub> ALD film on:

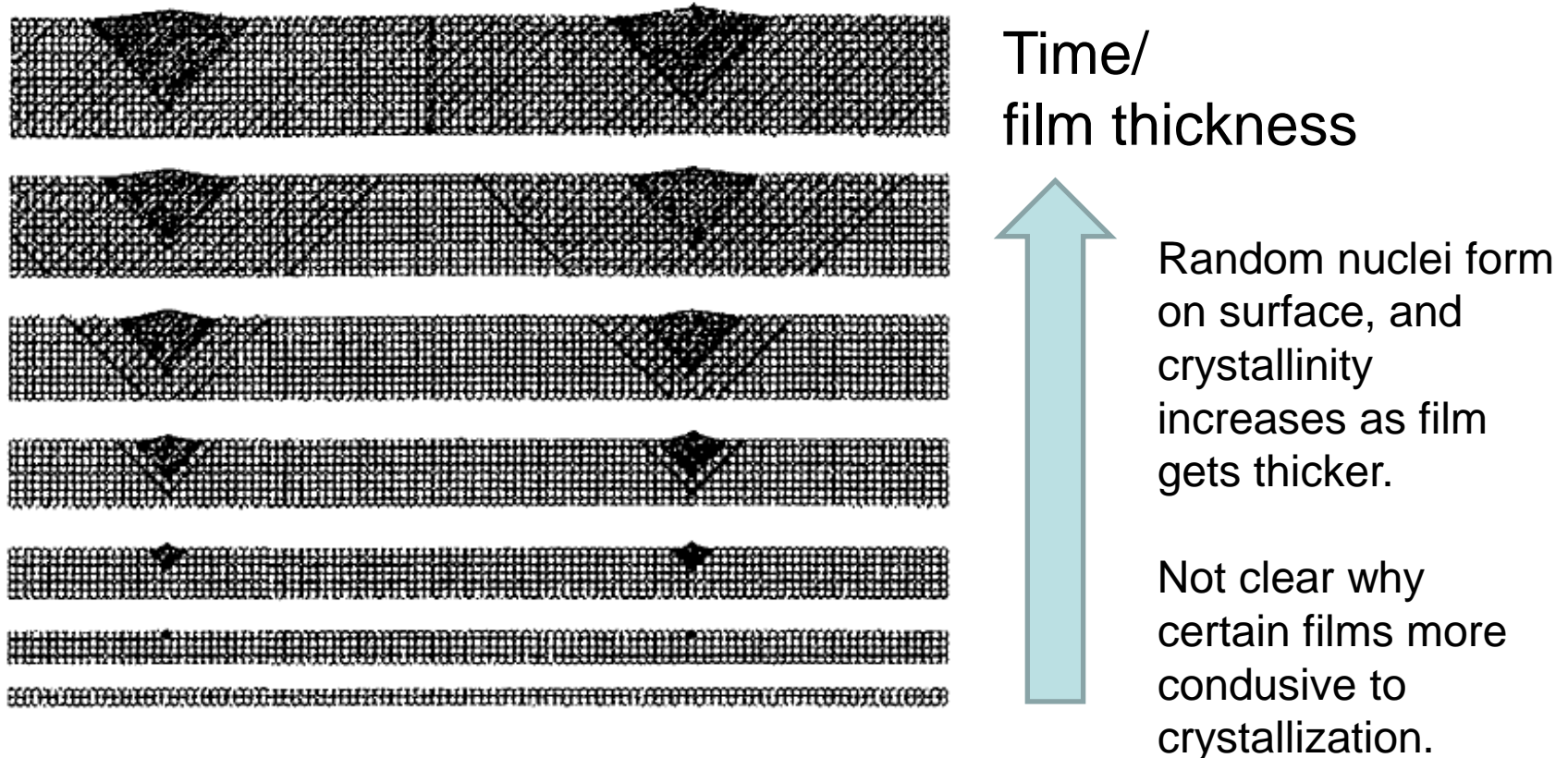
- Amorphous SiO<sub>2</sub>
- RCA cleaned <Si>
- Various Al<sub>2</sub>O<sub>3</sub> films

Crystalline TiO<sub>2</sub> on  
amorphous Al<sub>2</sub>O<sub>3</sub>

No crystalline TiO<sub>2</sub> on  
these amorphous Al<sub>2</sub>O<sub>3</sub>

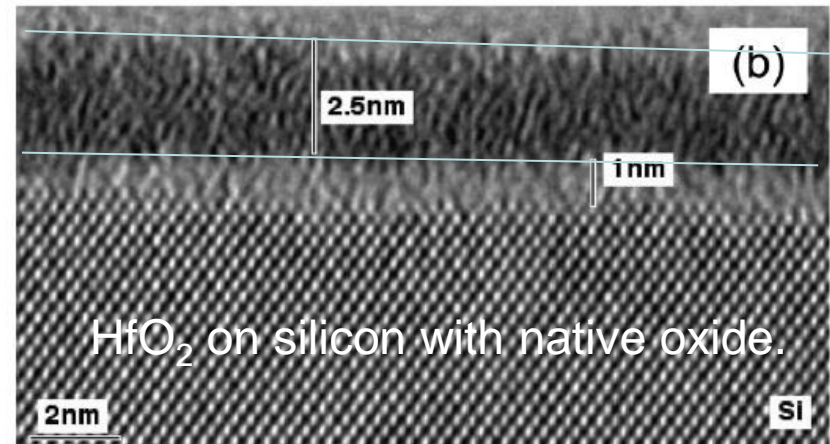
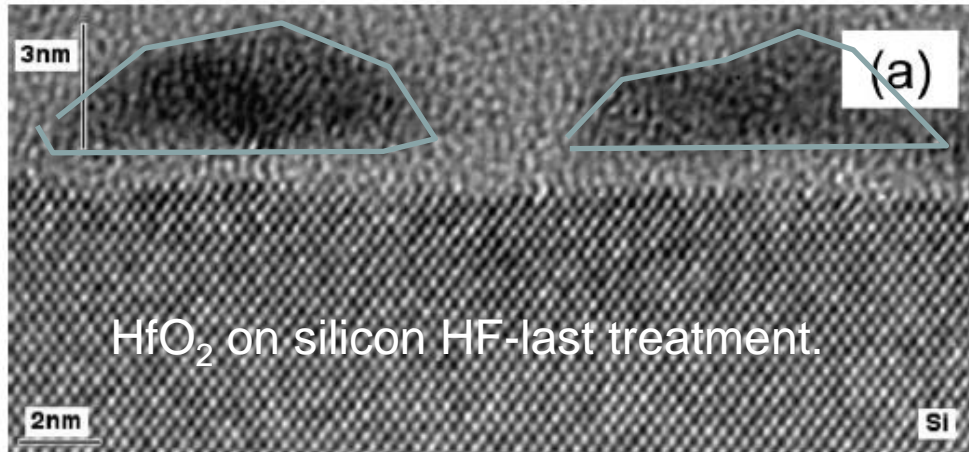
**Fig. 2.** GIXRD diffractograms for TiO<sub>2</sub> films deposited in 500 cycles on different substrates. Curves have been shifted vertically for clarity.

# Evolution of crystallinity



**Fig. 10.** Scheme for the initial growth of amorphous film (1st from bottom), the formation of crystalline  $\text{TiO}_2$  nuclei (2nd from bottom), growth of crystalline  $\text{TiO}_2$  grains (3rd–7th from bottom), and through-crystallization of the originally amorphous  $\text{TiO}_2$  film (4th–7th from bottom).

# Island growth vs. layer-by-layer growth

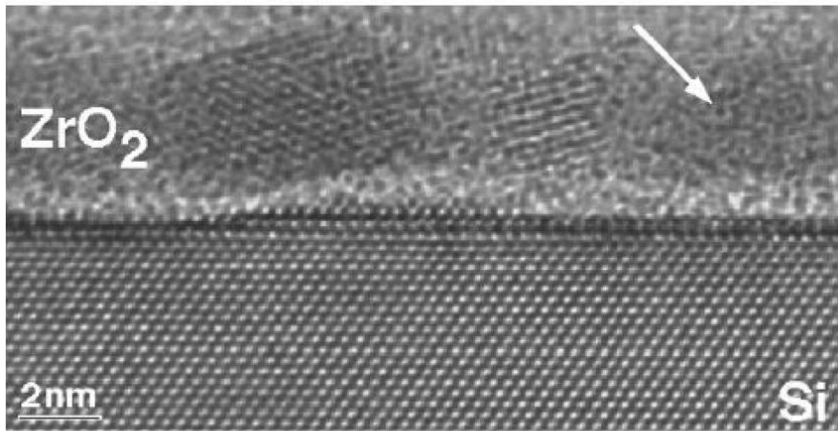


“One can observe a uniform, continuous film in the case of HfO<sub>2</sub>/SiO<sub>2</sub>/Si. For the film deposited directly on Si (after HF last pre-clean), the TEM image reveals the nucleation phenomenon, i.e. the deposited material forms clusters on the surfaces resulting in incomplete coverage of the surface by HfO<sub>2</sub>.”





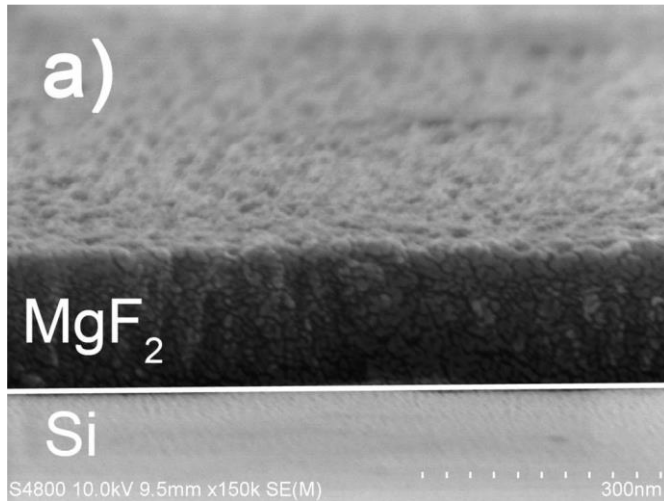
(a)



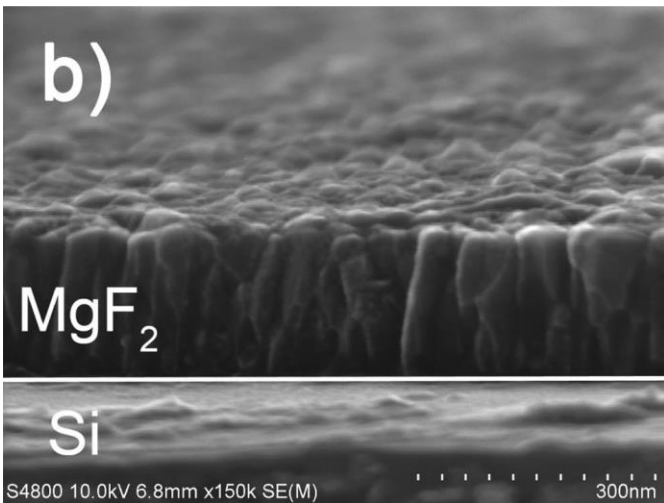
(b)

“On oxidized silicon, ZrO<sub>2</sub> layer is smooth yet clearly polycrystalline. On HF-etched silicon, the nucleation density has been lower and the film shows more or less separate crystallites and is much rougher, because the nuclei have had more time to grow independently before coalescing.”

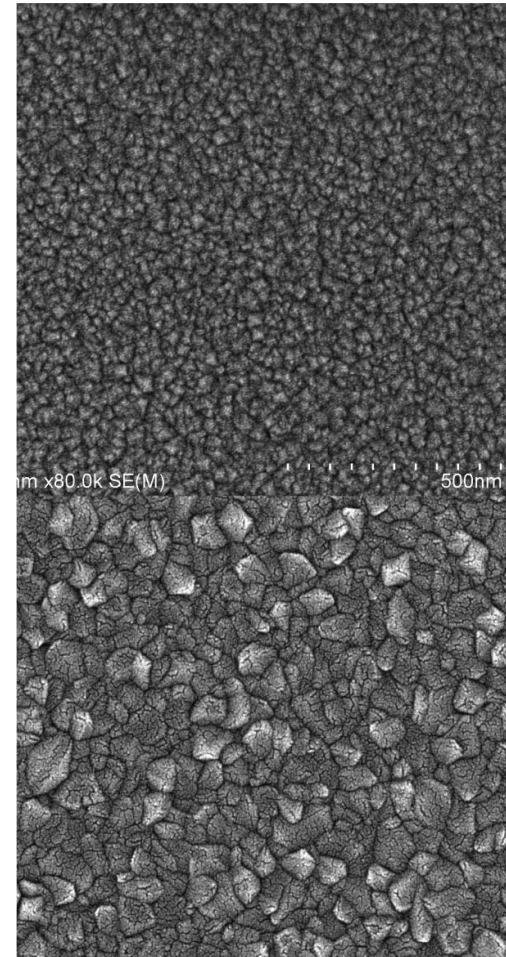
# Crystallinity & depo temperature



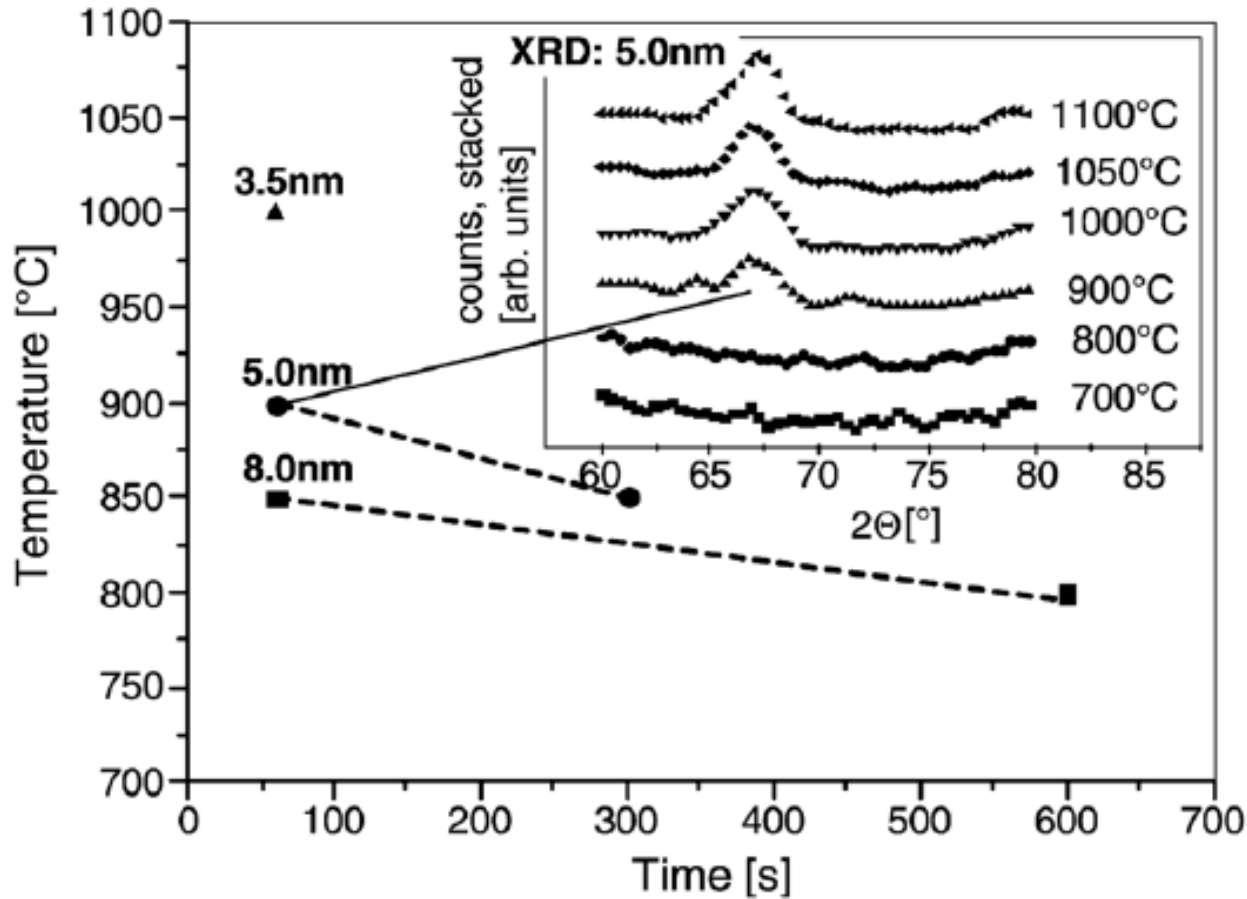
250°C



350°C  
Higher T  
favors  
crystallinity

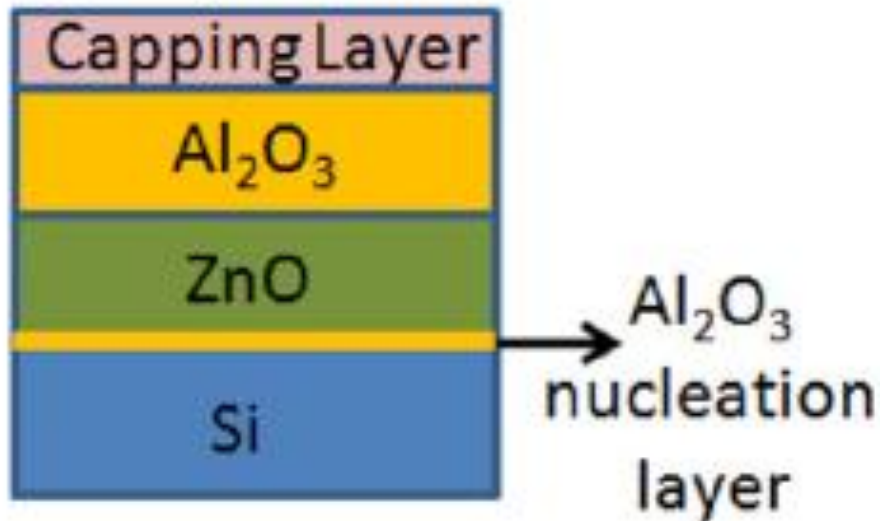


# Post deposition annealing



$\text{Al}_2\text{O}_3$   
turns  
crystalline  
at ca.  
900°C

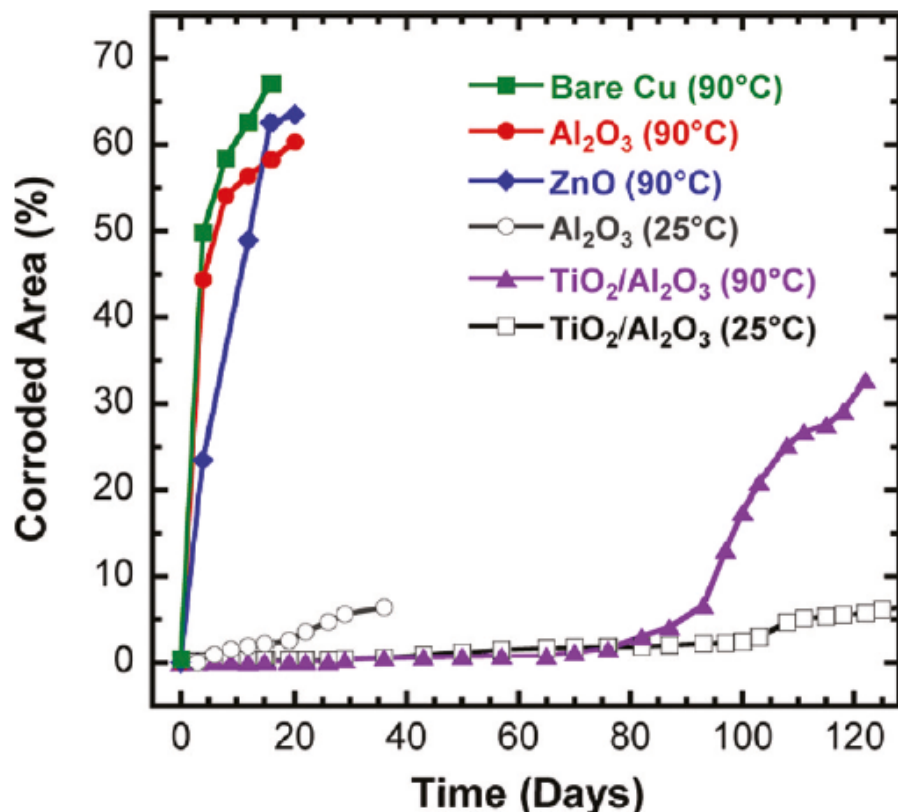
# Protective capping layers



## Capping Layers

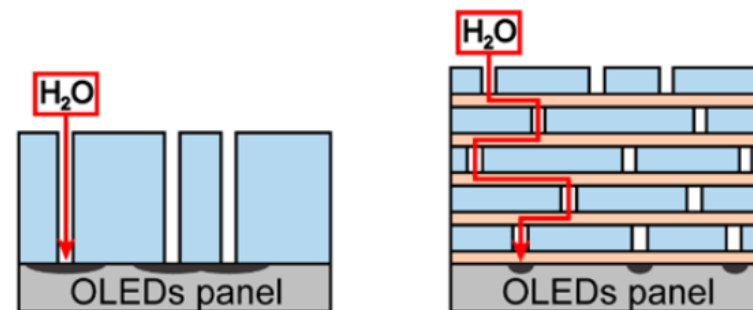
- (1) Ebeam NiO<sub>x</sub>
- (2) O<sub>2</sub> Plasma on NiO<sub>x</sub>
- (3) Ebeam TiO<sub>x</sub>
- (4) O<sub>2</sub> Plasma on TiO<sub>x</sub>
- (5) PEALD TiO<sub>2</sub>

# ALD for corrosion protection



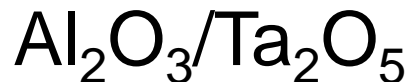
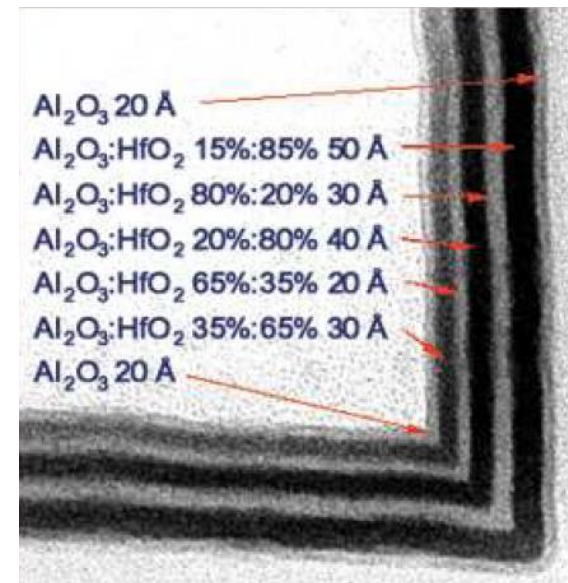
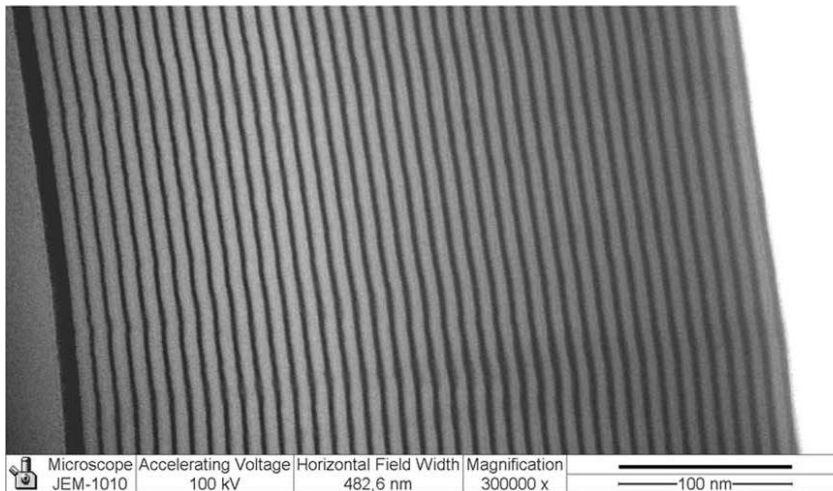
Usually higher deposition temperature results in higher quality film (denser film, less defects, e.g. holes, thinning).

In multilayers the defects are (in theory) non-correlated.



# Nanolaminates

- More layers, potentially more effective barrier.
- Lower deposition temperature but more layers maybe a good compromise.



Adriana Szeghalmi, Stephan Senz, Mario Bretschneider, Ulrich Gösele, and Mato Knez, APL 2009

# PVD

Atoms as source material

Solid source materials

Vacuum/high vacuum

Elemental films mostly

Room temperature

Alloy films easily (W:N)

One process, many materials

Al, Au, Cu, Pt, ... SiO<sub>2</sub>

# CVD & ALD

Molecules as source materials

Solid, liquid, gas precursors

Fluid dynamics important

Molecular/compound films mostly,  
Chemical bonds broken & formed

Needs elevated temperatures  
(or plasma activation)

Elements and compounds OK, alloys  
more difficult

Each process materials specific

SiO<sub>2</sub>, Si<sub>3</sub>N<sub>4</sub>, Al<sub>2</sub>O<sub>3</sub>, HfO<sub>2</sub>, ... Si, W

# CVD & ALD

Chemical Vapor Deposition, CVD  
Atomic Layer Deposition, ALD

Alternatives to PVD, but only partially.

Major uses:

- optical fiber fabrication
- films in microelectronics & MEMS
- optical coatings
- solar cells
- a-Si and poly-Si for flat panel displays



# Summary

- Thermal CVD: excellent film quality
  - PECVD: reasonable film quality at low T
  - ALD: excellent film quality at low T
- 
- Thermal CVD: high temperature needed
  - PECVD: very high rate possible
  - ALD: best for very thin films