Density-Functional Theory for Practitioners - Tutorial 3

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(Developed by Patrick Rinke and Milica Todorović)

Aalto University School of Science Department of Applied Physics



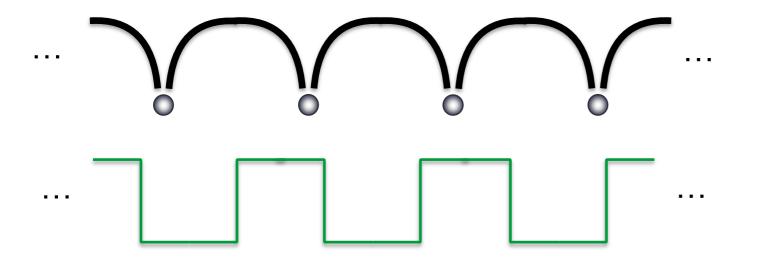
House Keeping - Bloch Theorem

periodic potential: (translational symmetry)

$$U(\mathbf{r} + \mathbf{R}) = U(\mathbf{r})$$

R is one of our lattice vectors:

$$\mathbf{R} = n_1 \mathbf{a}_1 + n_2 \mathbf{a}_2 + n_3 \mathbf{a}_3$$



Then you can find solution Schrödinger equation with that potential in this form:

Still 1 e- wavefunction

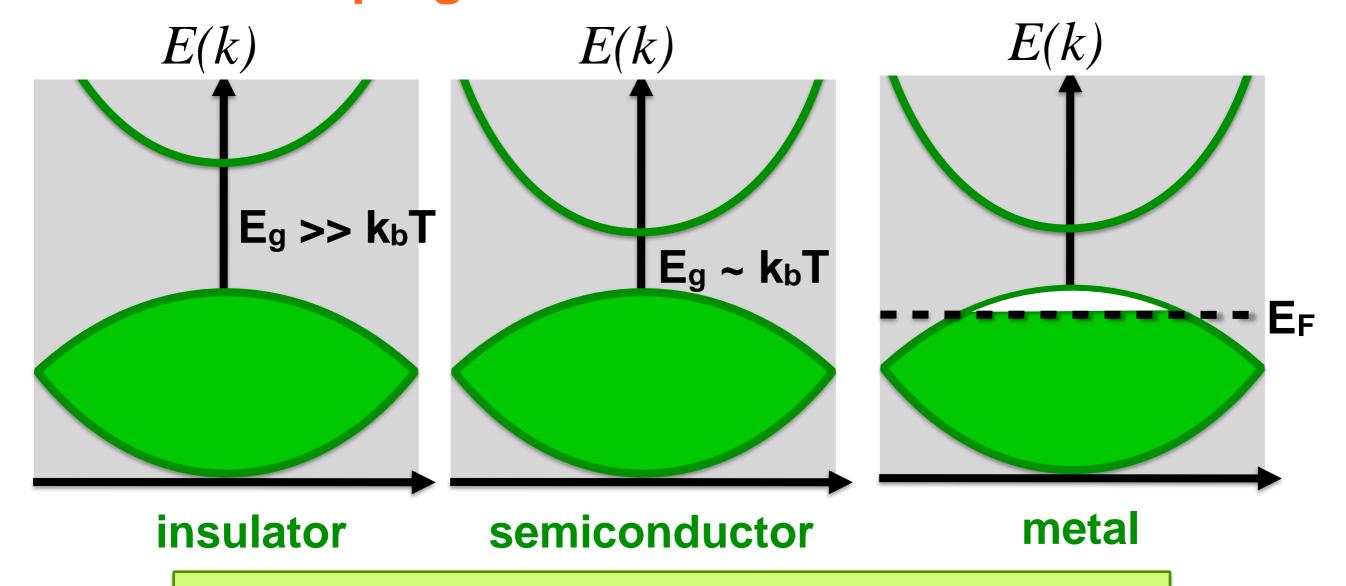
$$\psi(\mathbf{r}) = e^{i\mathbf{k}\mathbf{r}} u(\mathbf{r})$$

Phase factor

$$u(\mathbf{r} + \mathbf{R}) = u(\mathbf{r})$$

Periodically repeating part

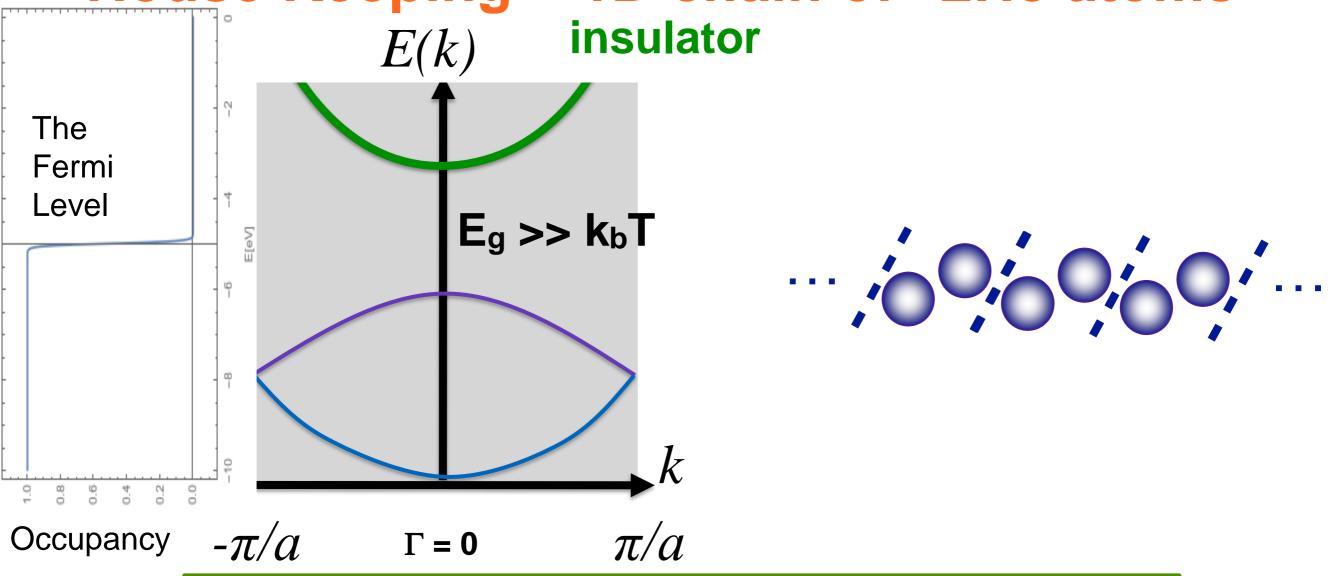
House Keeping – Band structure confusion



The Fermi energy marks the energy of the highest occupied state.



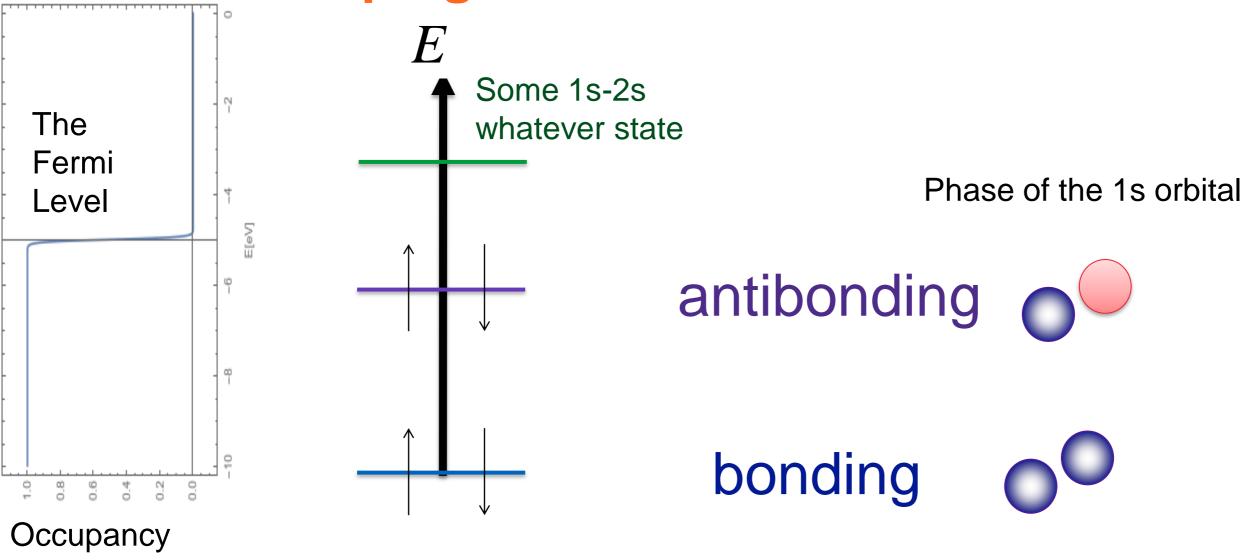
House Keeping – 1D chain of "2He atoms"



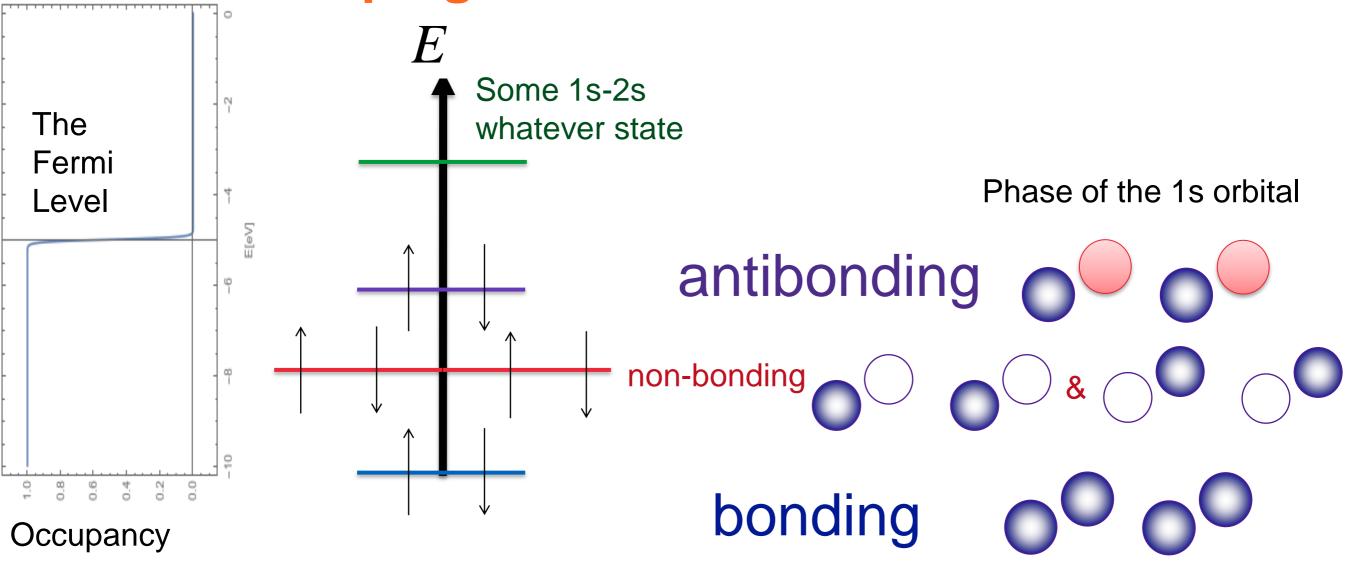
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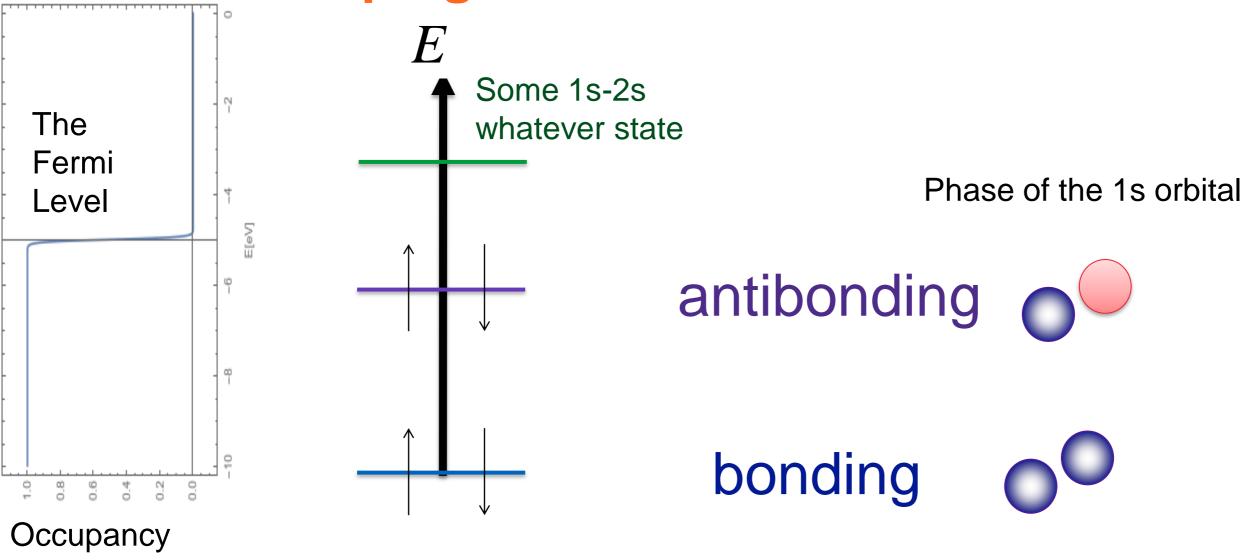
House Keeping – Back to molecule "2He atoms"



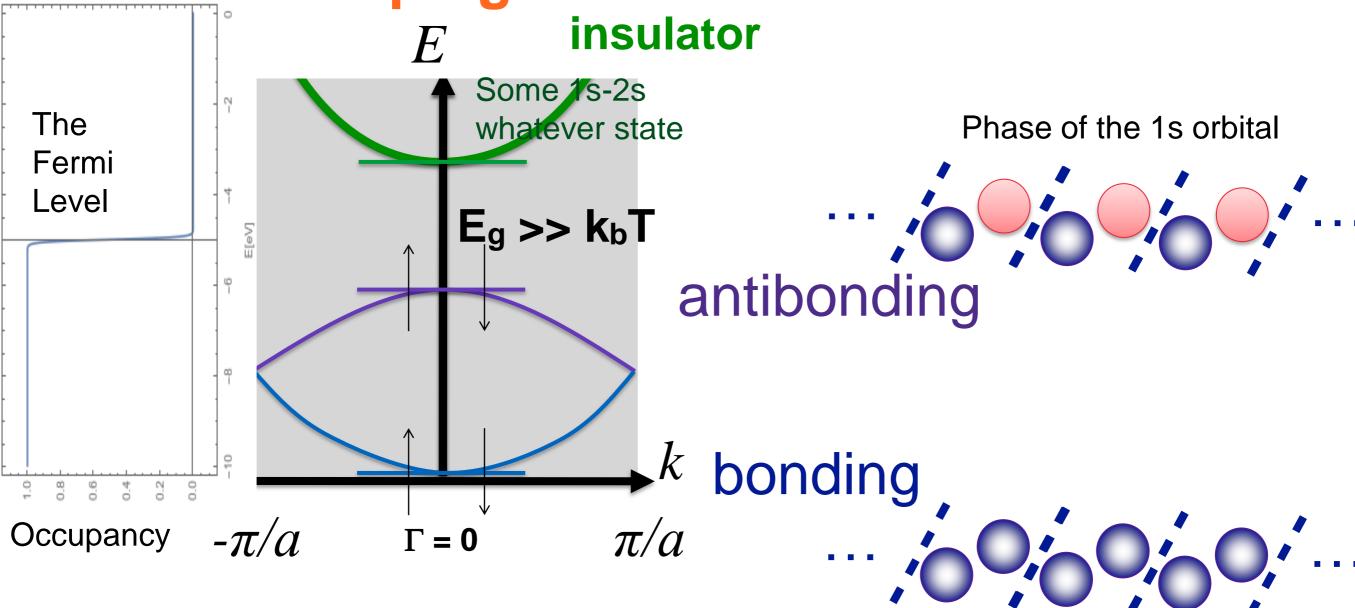
House Keeping – Back to molecule "4He atoms"



House Keeping – Back to molecule "2He atoms"



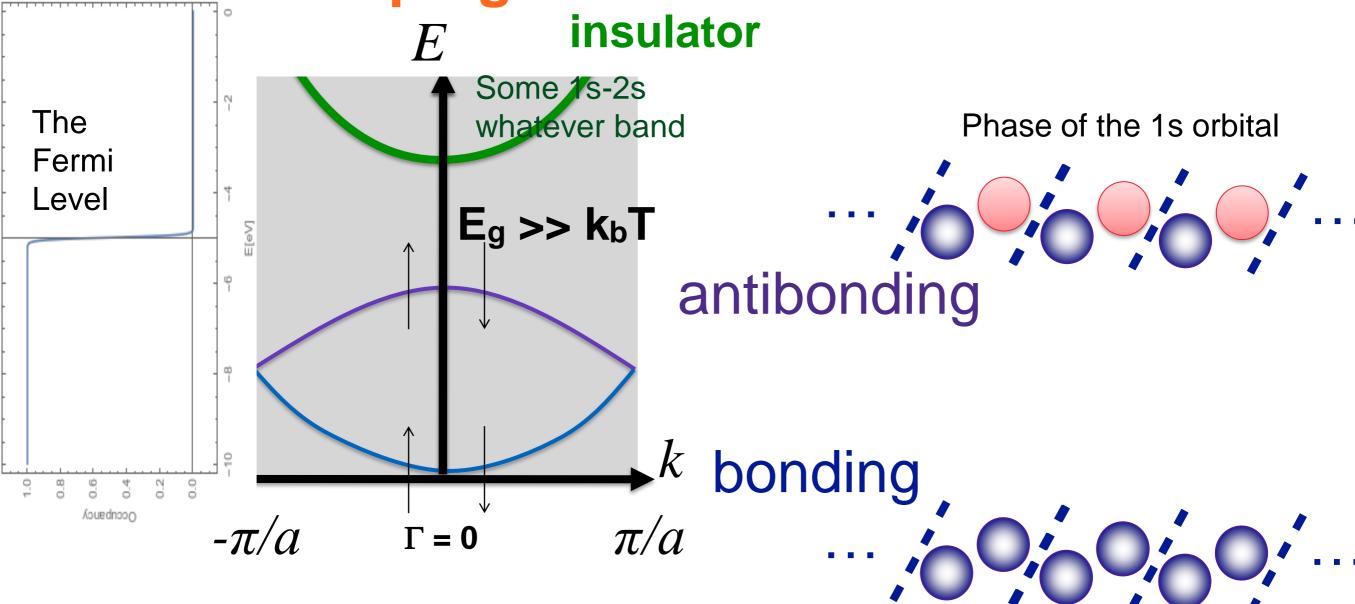
House Keeping – 1D chain of "2He atoms"



Only 2 atoms in the unit cells - 4 e



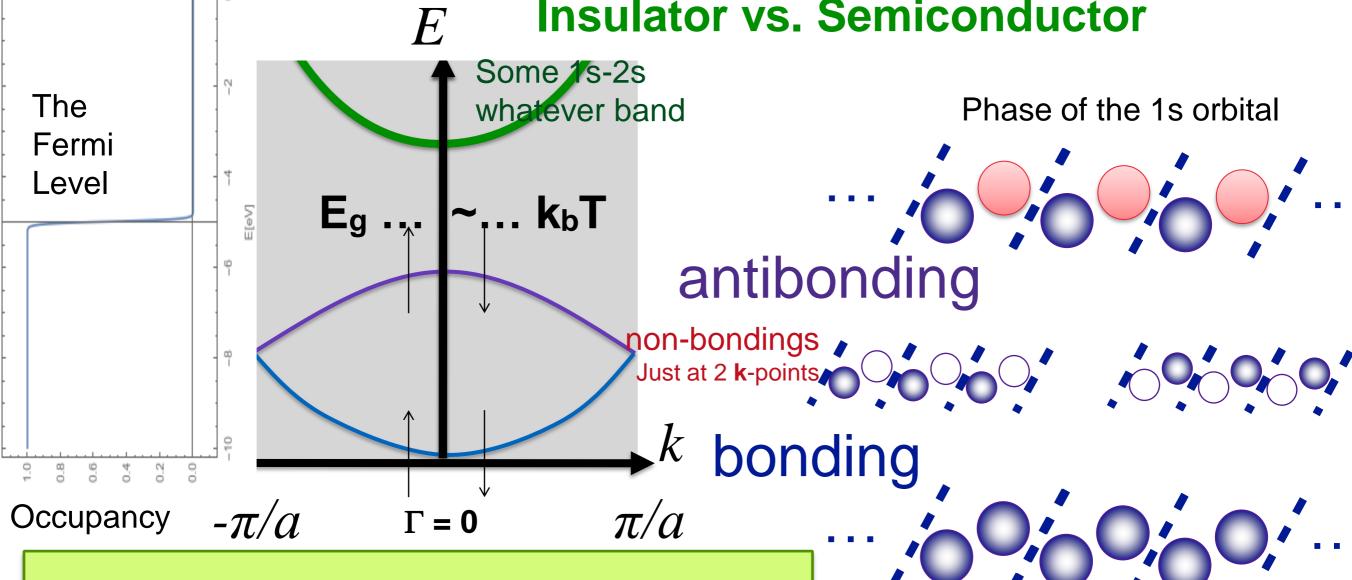
House Keeping – 1D chain of "2He atoms"



Only 2 atoms in the unit cells – 4 e⁻¹



House Keeping – 1D chain of "2He atoms" *E* Insulator vs. Semiconductor



Insulator vs. semiconductor is only divided by the size of gap



House Keeping – 1D chain of "2He atoms" Insulator vs. Semiconductor Some 1s-2s The whateve band Phase of the 1s orbital Fermi Level .. k_bT **E**g ..↑. antibonding non-bondings 🔪 Just at 1 k-point bonding

 π/a

Insulator vs. semiconductor is only divided by the size of gap

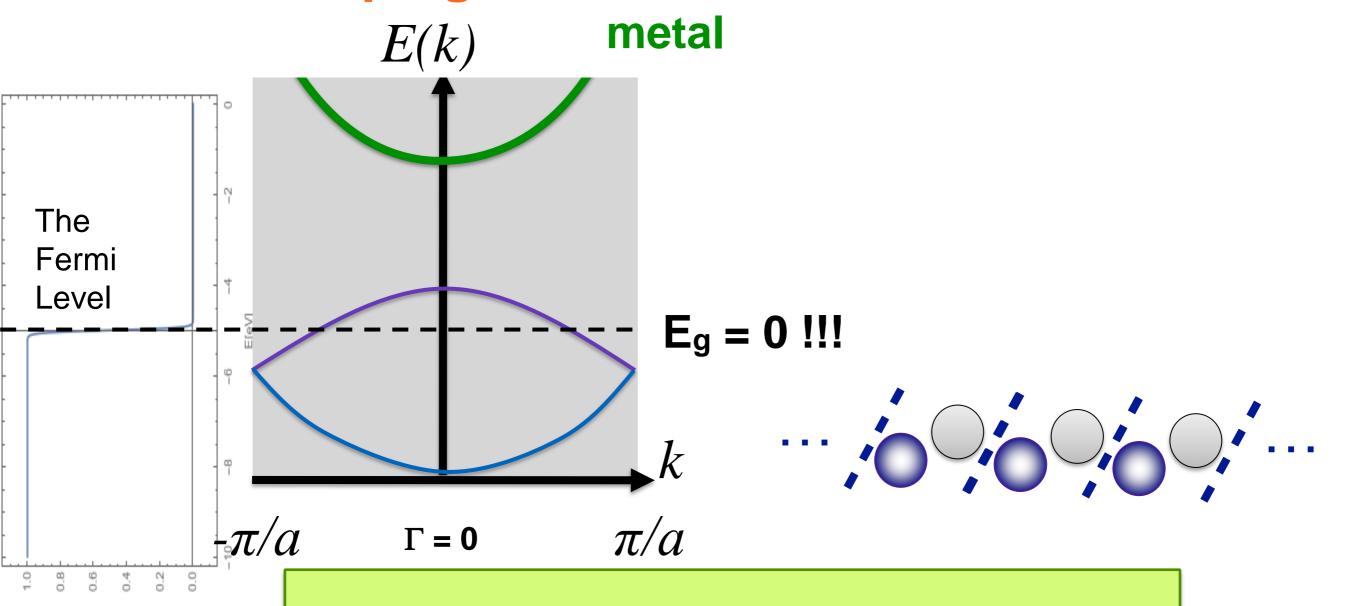
 $\Gamma = \mathbf{0}$



 $-\pi/a$

Occupancy

House Keeping – 1D chain of "He-H atoms"

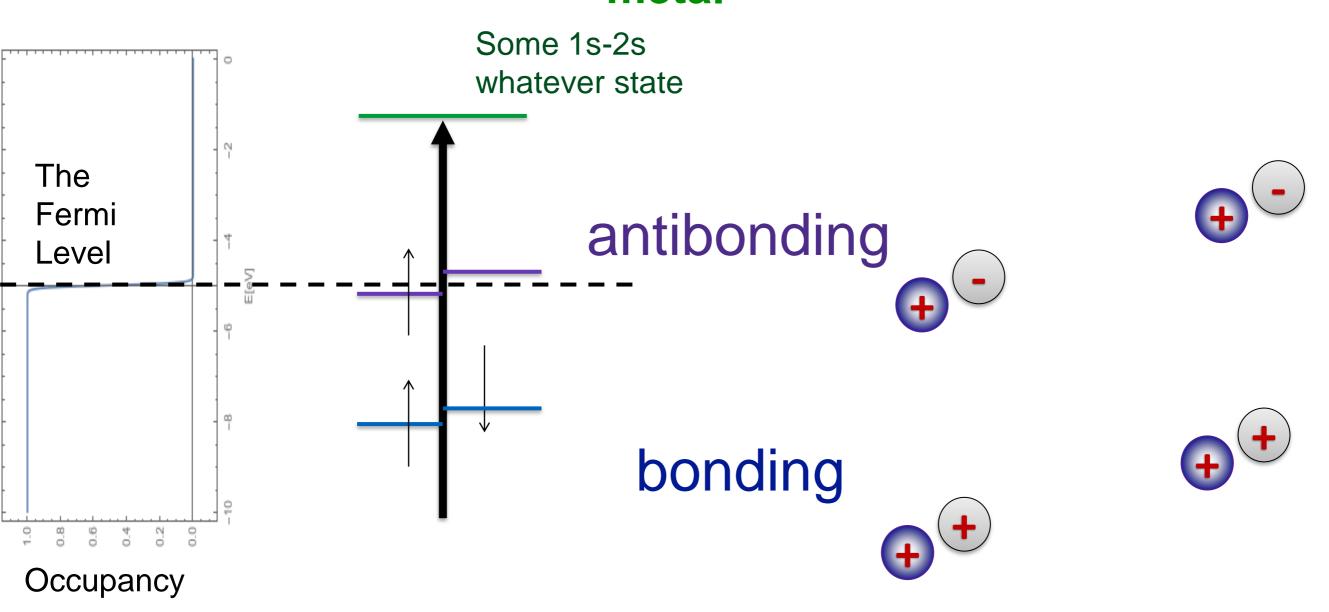


Occupancy

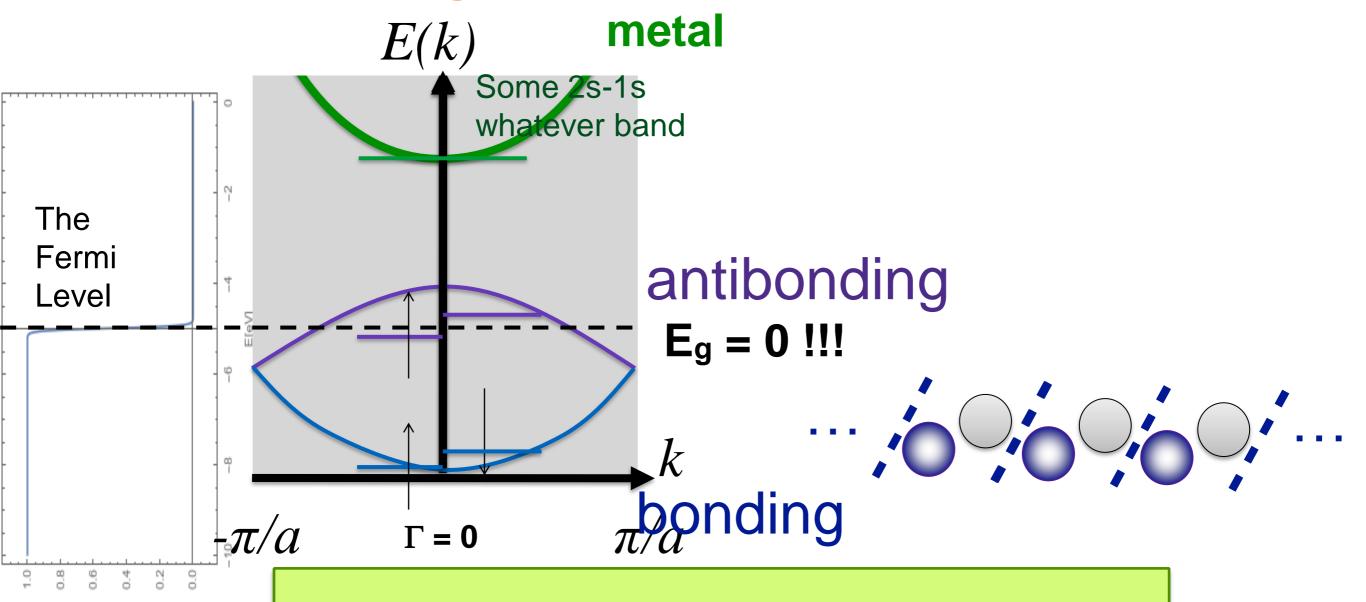
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House Keeping – Back to molecule "He-H atoms" metal



House Keeping – 1D chain of "He-H atoms"



Occupancy

No gap – electron can anytime be excited into the conduction part of the band



House Keeping – Measurables: occ. states spectroscopy

gas molecule "He₂": 1D "He₂" solid: No momentum Momentum resolved × momentum integrating E(k)The Fermi Level $-\pi/a$ Occupancy



House Keeping – Measurables: occ. states spectroscopy

gas molecule "He2": 1D "He₂" solid: No momentum Momentum resolved × momentum integrating E(k)The Fermi Level QUESTIONS? – Before going to >1 D Occupancy



House Keeping – 2D band structure

Graphene 1st Brillouin zone Real space lattice: k_y 1D projection: δ_3 k_{x} М The Fermi 2D band structure: Level -10



House Keeping – 3D band structure

Silicon 1D path Real space lattice: 1st Brillouin zone Along 3D band structure *no real standardization E vs. top of the valence band The Fermi Level 3.4 eV © bilbao crystallographic server -8 -10X U.K. Occupancy

https://www.cryst.ehu.es/cryst/get_kvec.html - add there Fd-3m = 227



Real-space lattice: https://materialsproject.org/materials/mp-149
1D band projection: https://myengineerings.com/crystalline-silicon/

wavevector **k**

House Keeping – 2D crystal calculations

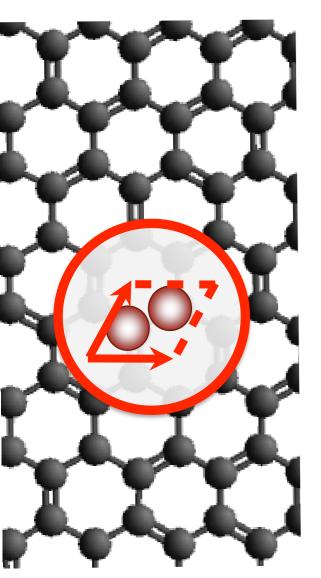
Graphene Real space lattice:

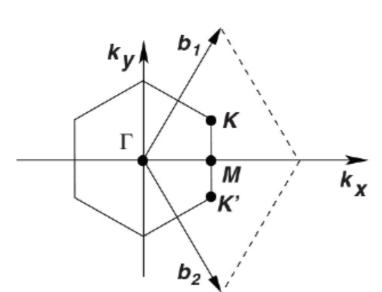
1st Brillouin zone

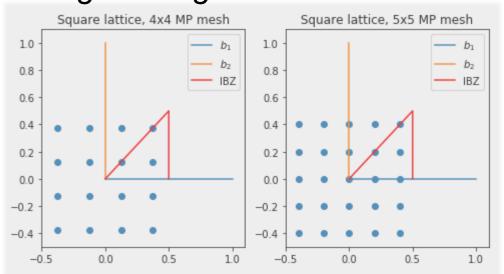
1st BZ:

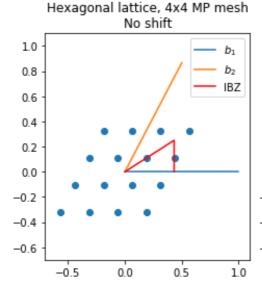
Has to be represented on a GRID!!!

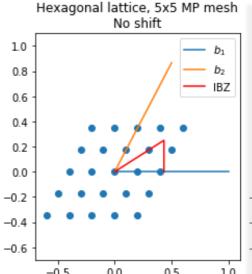
Discreet grid – e.g Monkhorst-Pack mesh

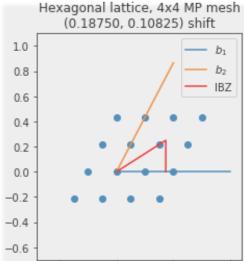












You only need to specify the density allong axis.



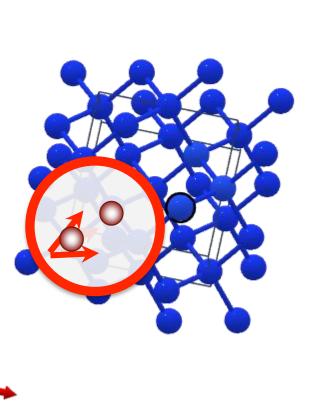
MP mesh: https://sites.psu.edu/dftap/2019/02/01/shifting-of-kpoints-in-hexagonal-lattices/

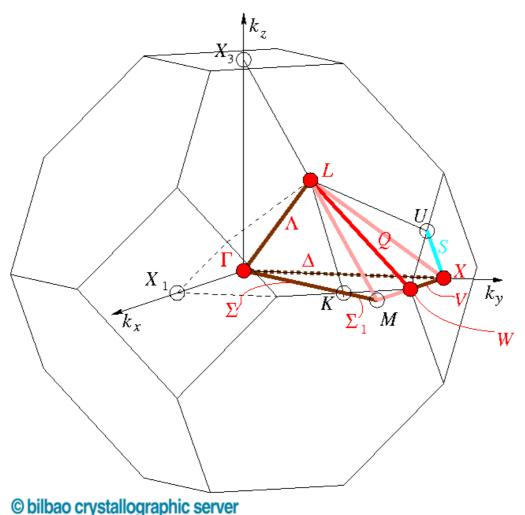
House Keeping – 3D crystal calculations

Silicon
Real space lattice:

1st Brillouin zone

1st BZ: Has to be represented on a GRID!!! 1 type of 3D representation:







You only need to specify the density allong axis.

http://www.cryst.ehu.es

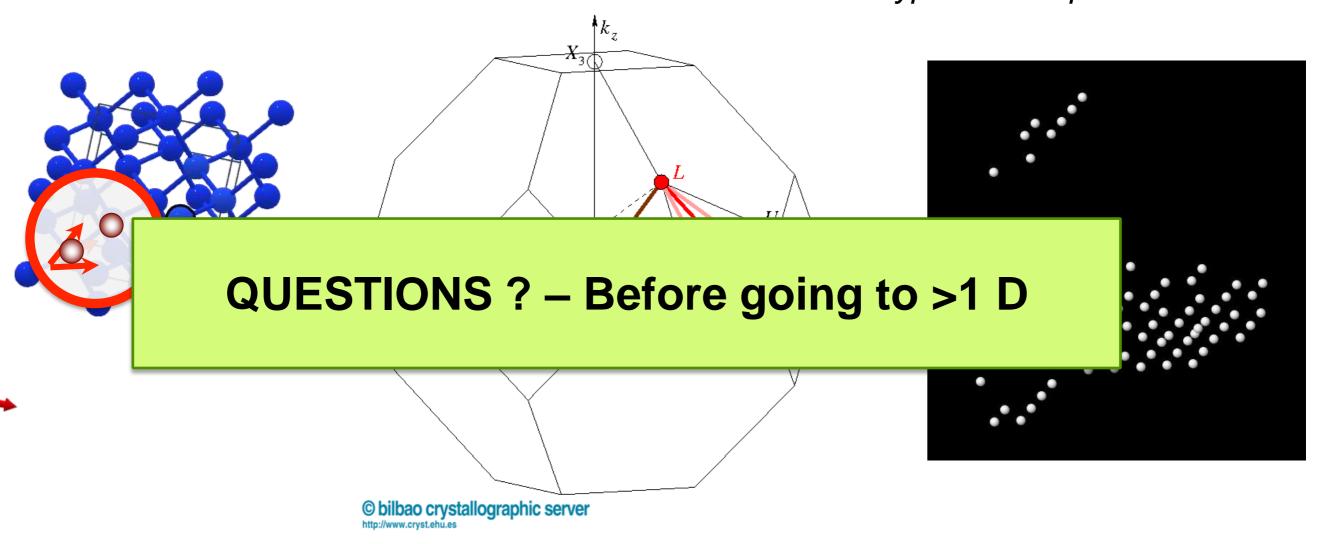


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Silicon Real space lattice:

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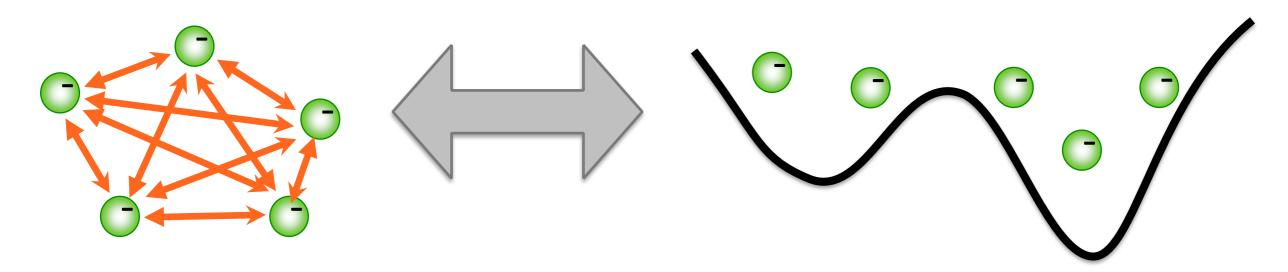


You only need to specify the density allong axis.



Last week at the same time

$$v_{\mathrm{ext}}(\mathbf{r}) \Leftrightarrow n_0(\mathbf{r}) = \sum_{i=1}^{N_e} |\phi_i(\mathbf{r})|^2$$



Kohn-Sham DFT:

- interacting system can be mapped onto fictitious non-interacting system.
- this makes DFT tractable.

Kohn-Sham (KS) equations

Now we can solve coupled equations:

$$v_{
m aux}[n]({f r}) = v_{
m ext}({f r}) + v_{
m Hartree}[n]({f r}) + v_{
m xc}[n]({f r})$$
 KS potential KS orbitals/wave functions



$$\left[-\frac{\nabla_{\mathbf{r}}^2}{2} + v_{\text{aux}}[n](\mathbf{r}) \right] \phi_i(\mathbf{r}) = \epsilon_i \phi_i(\mathbf{r})$$

$$n(\mathbf{r}) = \sum_i^{N_e} |\phi_i(\mathbf{r})|^2$$
 KS eigenvalues

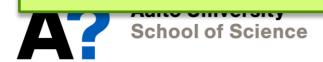
Basis functions

Let's expand Kohn-Sham states in a basis.

$$\phi_j(\mathbf{r}) = \sum_i c_{ij} \varphi_i(\mathbf{r})$$

Many basis choices:

- plane waves or real space
 - linear augmented plane waves (LAPW)
 - projector augmented plane waves (PAW)
- local orbitals (LCAO)
 - Gaussian orbitals
 - numeric atom centered orbitals (NAOs)



In FHI-aims: flexible basis function choice

$$\varphi_{i[lm]}(\mathbf{r}) = \underbrace{\frac{u_i(r)}{r} Y_{lm}(\Omega)}_{r}$$
 spherical harmonic

solution of a radial Schrödinger equation:

$$\left[-\frac{d^2}{dr^2} + \frac{l(l+1)}{r^2} + v_i(r) + v_{\text{cut}}(r) \right] u_i(r) = \epsilon_i u_i(r)$$

- free atom like: $v_i(r) = v_{\mathrm{free\ atom}}^{\mathrm{DFT}}(r)$
- Hydrogen like: $v_i(r) = Z/r$
- free ions, harm. osc. (Gaussians), ...

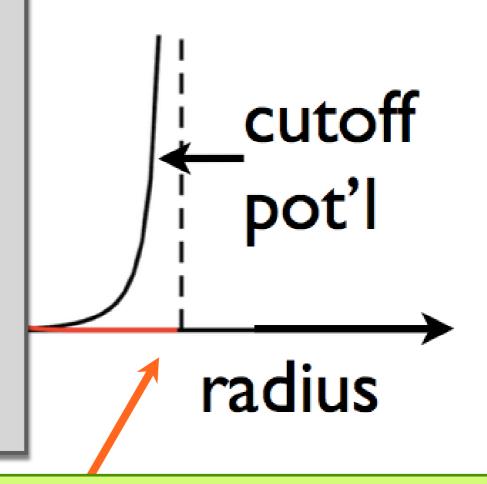
$$\left[-\frac{d^2}{dr^2} + \frac{l(l+1)}{r^2} + v_i(r) + v_{\text{cut}}(r) \right] u_i(r) = \epsilon_i u_i(r)$$

FHI-aims technical detail:

specification of the cut-off potential

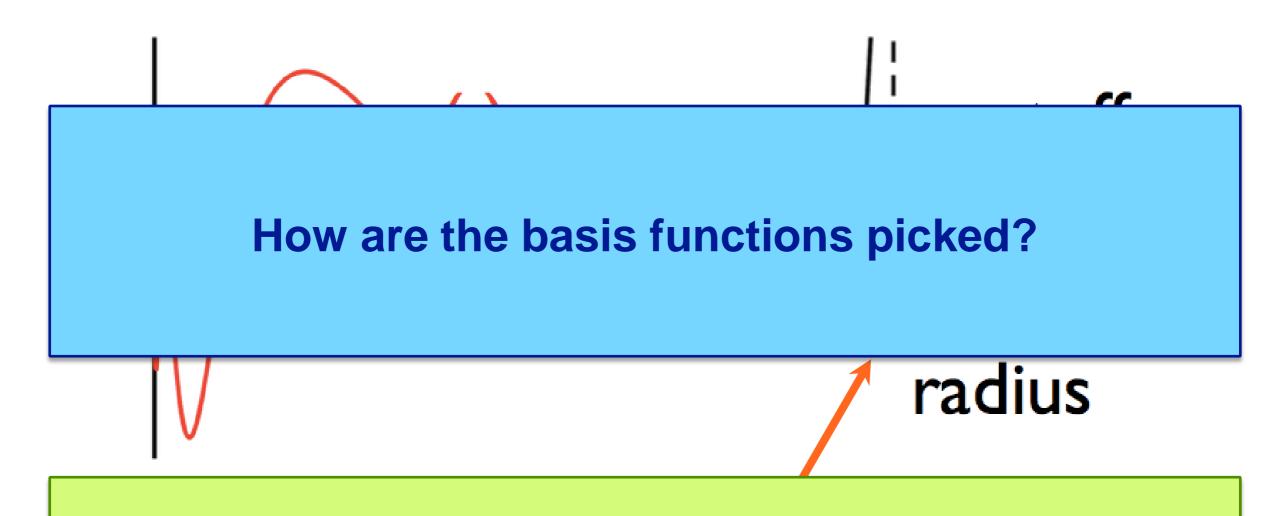
cut_pot onset width scale

e.g. cut_pot 3.5 1.5 1.0



Necessary to localise basis functions in space ⇒ improves computational scaling!





Necessary to localise basis functions in space ⇒ improves computational scaling!



Wishlist:

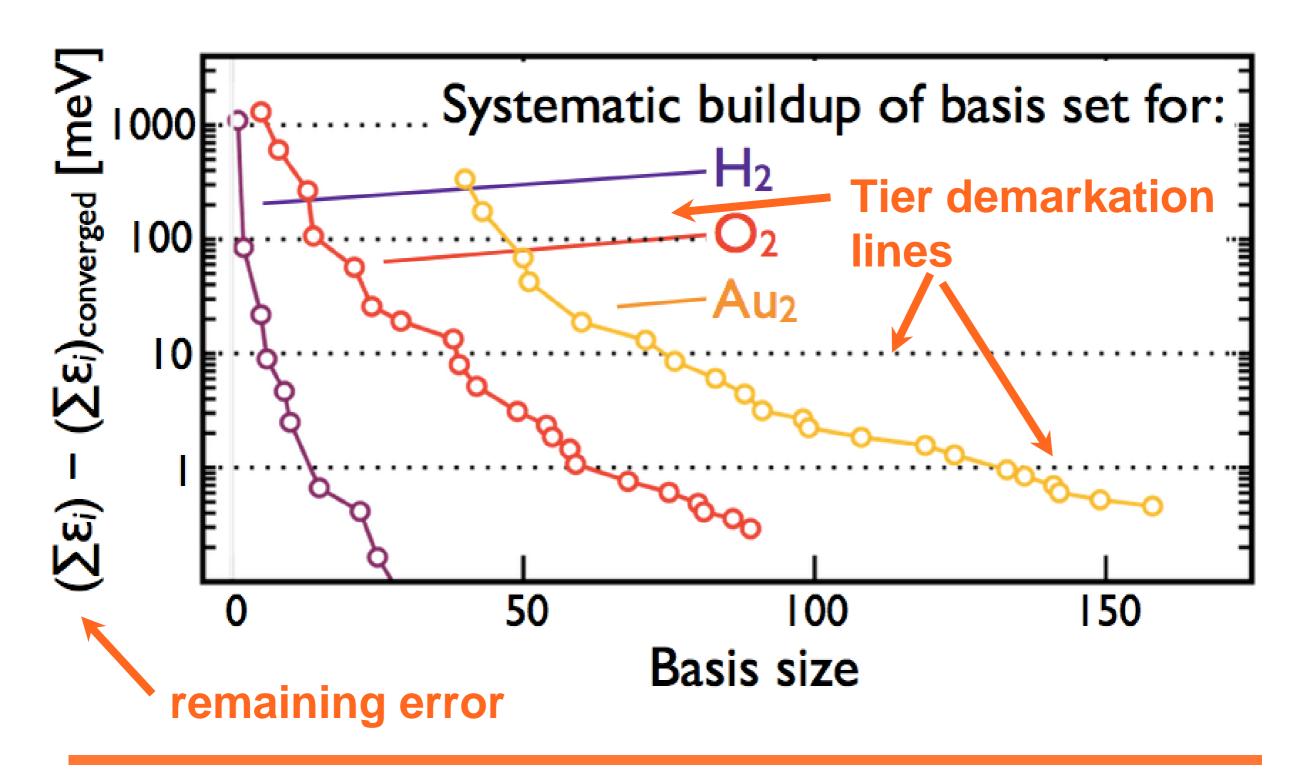
- element specific
- transferable
- from fast qualitative to meV-converged total energy accuracy

Good news! – FHI-aims developers prepared set of "Tiers" to systematically improve the accuracy!*

- free atom like: $v_i(r) = v_{
 m free\ atom}^{
 m DFT}(r)$ Hydrogen like: $v_i(r) = Z/r$
- free ions, harm. osc. (Gaussians), ...



Basis function selection in FHI-aims





Result: Hierarchical Basis Set Library for All Elements

| | H | C | O | Au |
|---------|-----------------------|----------------------|------------------------|------------------------|
| minimal | 1s | [He]+2s2p | $[\mathrm{He}] + 2s2p$ | [Xe] + 6s5d4f |
| Tier 1 | H(2s,2.1) | $\mathrm{H}(2p,1.7)$ | H(2p,1.8) | Au ²⁺ (6p) |
| | $\mathrm{H}(2p, 3.5)$ | H(3d,6.0) | H(3d, 7.6) | H(4f, 7.4) |
| | | $\mathrm{H}(2s,4.9)$ | H(3s,6.4) | $\mathrm{Au^{2+}(6s)}$ |
| | | | | H(5g,10) |
| | | | | H(6h,12.8) |
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| | H(2s,1.2) | H(3s, 4.3) | H(3d,5.6) | $\mathrm{H}(3p,3.3)$ |
| | H(3d,7.0) | H(5g,14.4) | H(5g,17.6) | H(1s,0.45) |
| | | H(3d,6.2) | H(1s,0.75) | H(5g,16.4) |
| | | | | H(6h, 13.6) |
| Tier 3 | H(4f,11.2) | H(2p,5.6) | $O^{2+}(2p)$ | $H(4f,5.2)^*$ |
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| | ••• | ••• | ••• | ••• |

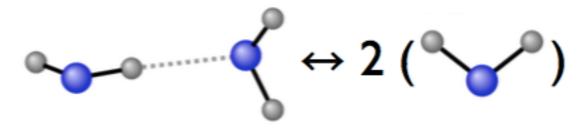
Systematic hierarchy of basis (sub)sets, iterative automated construction based on dimers

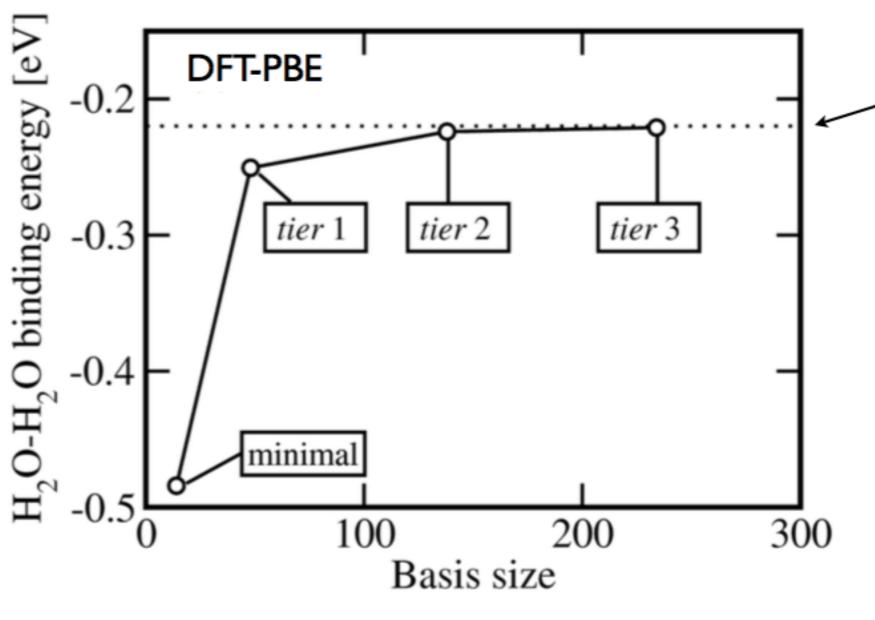
"First tier (level)"

"Second tier"

"Third tier"

Accuracy: (H₂O)₂ Hydrogen Bond Energy





Basis set limit (independent): $E_{Hb} = -219.8 \text{ meV}$

Basis sets: Radial fn. character

| | Н | C,N,O |
|---------|---------|-----------|
| minimal | ls | [He]+2s2p |
| tier | s,þ | s,p,d |
| tier 2 | s,p,s,d | s,p,d,f,g |
| tier 3 | s,p,d,f | s,p,d,f |



Kohn-Sham equations in a basis

Pick basis set {φ_i} and expand:

$$\phi_n(\mathbf{r}) = \sum_i c_{ni} \varphi_i(\mathbf{r})$$

Solve generalised eigenvalue problem:

$$egin{aligned} \mathbf{h} \mathbf{c}_n &= \epsilon_n \mathbf{s} \mathbf{c}_n \ h_{ij} &= \langle arphi_i | \hat{h} | arphi_j
angle \ s_{ij} &= \langle arphi_i | arphi_j
angle \end{aligned}$$



Kohn-Sham equations in a basis

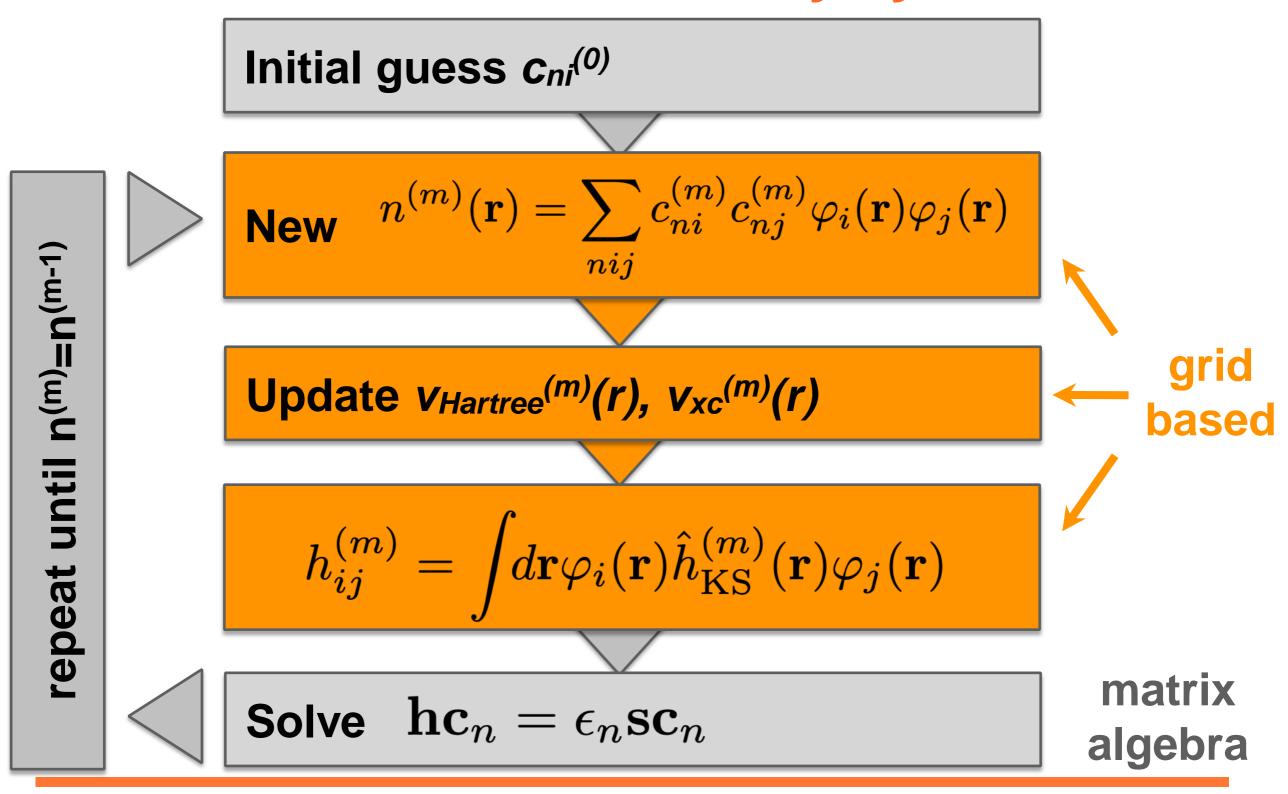
Two advantages:

- 1. number of $\{\phi_i\}$ << number of r points
 - ⇒ smaller matrices
- 2. matrix algebra very efficient on computers

$$\phi_n(\mathbf{r}) = \sum_i c_{ni} \varphi_i(\mathbf{r})$$
 $\mathbf{hc}_n = \epsilon_n \mathbf{sc}_n$
 $h_{ij} = \langle \varphi_i | \hat{h} | \varphi_j \rangle$
 $s_{ij} = \langle \varphi_i | \varphi_j \rangle$



Kohn-Sham self-consistency cycle





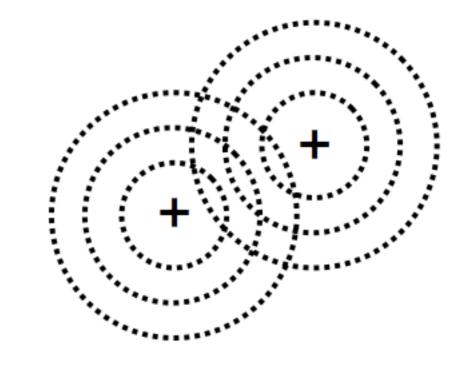
Grid-based operations

$$h_{ij}^{(m)} = \int d\mathbf{r} \varphi_i(\mathbf{r}) \hat{h}_{KS}^{(m)}(\mathbf{r}) \varphi_j(\mathbf{r})$$

Discretise to integration grid:

$$\int \!\! d\mathbf{r} f(\mathbf{r}) \to \sum_{\mathbf{r}} w(\mathbf{r}) f(\mathbf{r})$$

FHI-aims uses overlapping atom-centered grids:



FHI-aims technical detail:

FHI-aims uses logarithmic grids with the following settings:

points

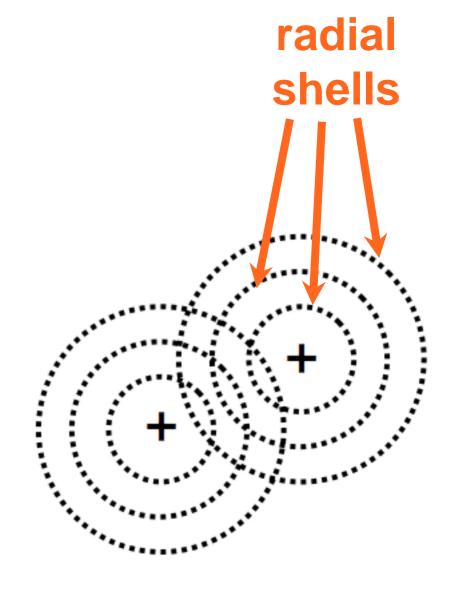
radial_base : 36 5.0 ← max.

radial_multip: 1 rad

lier

radius multiplies # points

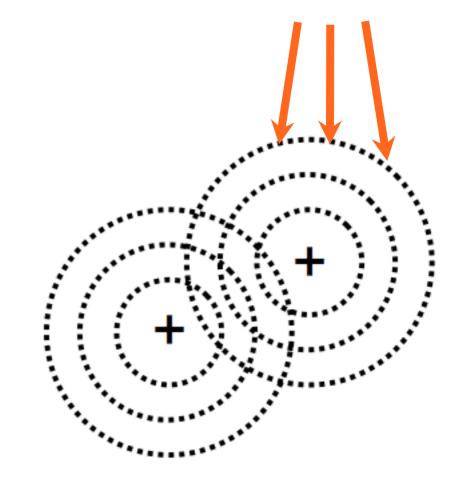
FHI-aims uses overlapping atom-centered grids:





FHI-aims technical detail: Lebedev grids can be made denser in outer regions: # points angular_grids specified division 0.2659 50 division 0.4451 110 division 0.6052 194 division 0.7543 302 outer_grid 194

Specific angular point distribution ("Lebedev")





radius

FHI-aims grid accuracy settings

FHI-aims technical detail:

grid settings can be pre-defined:

light

: computationally cheap, out-of-the-box settings for fast prerelaxations, structure searches, etc.

intermediate: For *hybrid-XC*: improved accuracy (derived from tight), but with lower computation costs.

tight

: safe, guaranteeing meV-level accurate energy differences also for large structures.

really tight

: for most purposes, strongly overconverged settings for convergence verification beyond tight.



FHI-aims basis set overview

| | light | intermediate | tight | really tight |
|--------|-------|--------------|-------|--------------|
| Tier 1 | | | | |
| Tier 2 | | Increasedad | Cur | |
| Tier 3 | | | Alach | |
| Tier 4 | | | | |



FHI-aims basis set overview

| | light | intermediate | tight | really tight |
|--------|--|--|--------------------|--------------|
| Tier 1 | these combinations don't make much sense | New – for hybrid XC with improved accuracy, but keeping the computational cost down | | |
| Tier 2 | | | Usually sufficient | |
| Tier 3 | | | | |
| Tier 4 | | This combination don't make much sense | | |



Questions?

Enjoy our "solid-state & surface modelling intro" tutorial:

- Download from Mycourses
- /work/courses/unix/PHYS/E0546/TUTORIALS
- Ask for printouts

Interesting links related to the tutorial:

Bilbao crystallographic server – https://www.cryst.ehu.es/

Materials project - https://materialsproject.org

Web of Elements - https://www.webelements.com/

Don't forget to put your name in the list of attendance!



Basis functions in FHI-aims

Wishlist:

- element specific
- transferable
- from fast qualitative to meV-converged total energy accuracy

Maybe the computer can help us pick a good basis set.

- free atom like: $v_i(r) = v_{
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- free ions, harm. osc. (Gaussians), ...

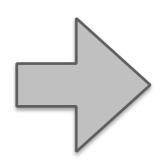


Basis function selection in FHI-aims

Robust iterative selection strategy (e.g., Delley 1990)

Initial set

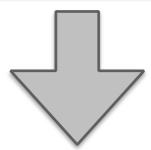
Occupied free atom orbitals $\{u\}^{(0)} = \{u_{free}\}$



Pool of candidates

- search pool *u*_{trial}
- find $u_{opt}^{(n)}$ that minimises $E^{(n)} = E[\{u\}^{(n-1)} \bigoplus u_{trial}]$





repeat until

 $E^{(n-1)}-E^{(n)} < threshold$



Increase basis set

$$\{u\}^{(n)} = \{u\}^{(n-1)} \bigoplus u_{opt}^{(n)}$$



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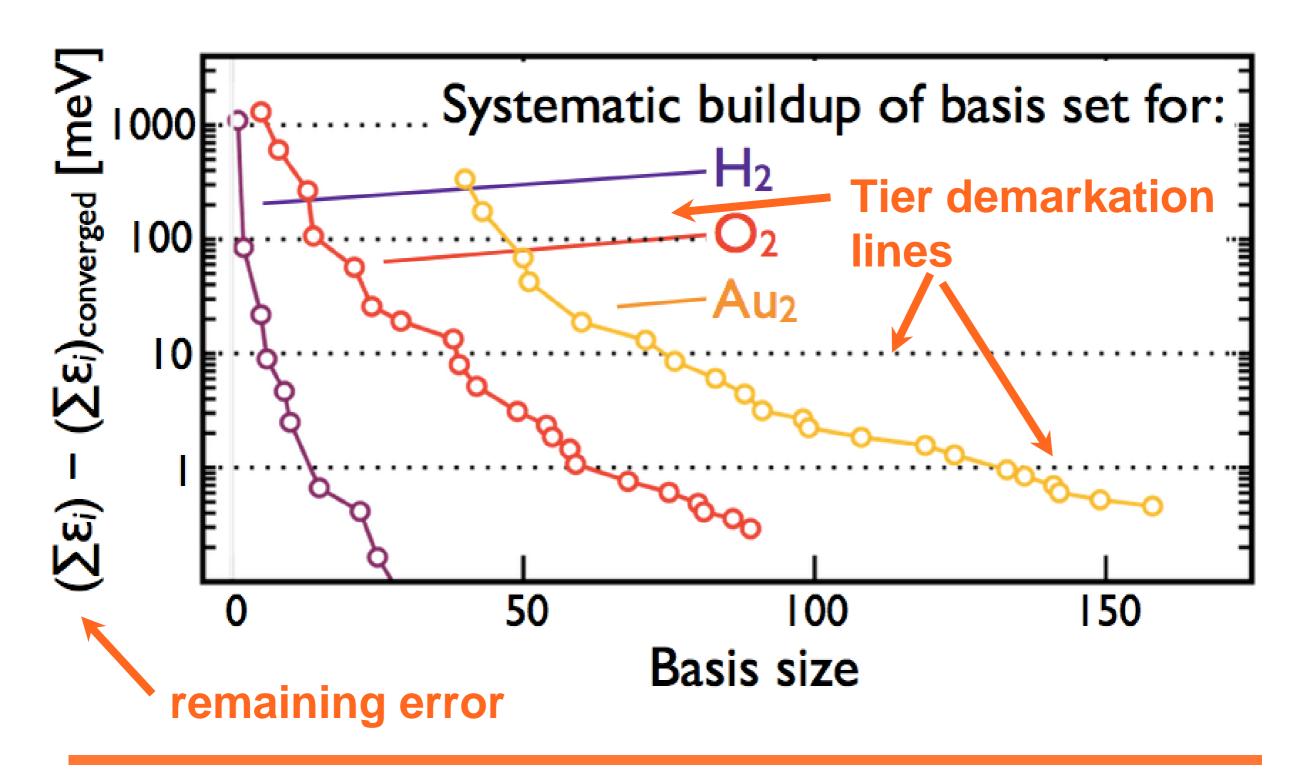
Pool of trial functions u(r):

- 2+ ionic
- Hydrogen-like for z=0.1-20

Optimisation target:

Non-self-consistent symmetric dimers, averaged for different bonding distances

Basis function selection in FHI-aims





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Systematic hierarchy of basis (sub)sets, iterative automated construction based on dimers

"First tier (level)"

"Second tier"

"Third tier"

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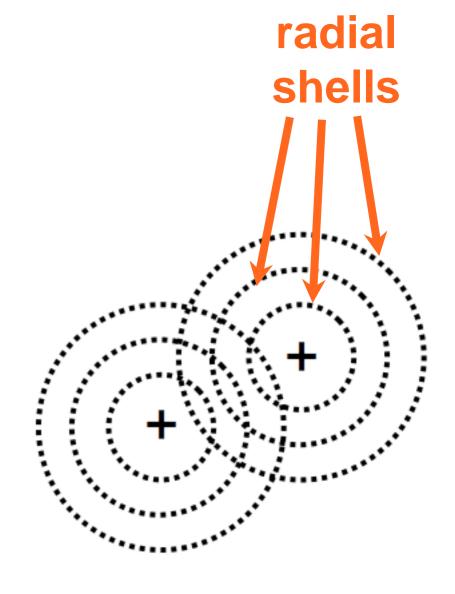
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radial_multip: 1 radius

lier

multiplies # points

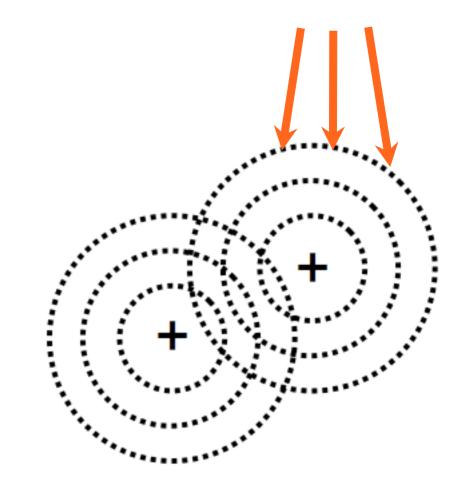
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