

# CVD & ALD

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## CVD & ALD

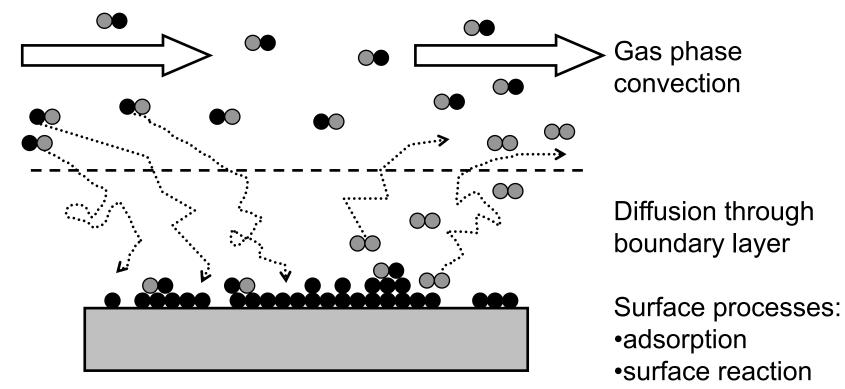
Chemical Vapor Deposition, CVD Atomic Layer Deposition, ALD

Alternatives to PVD, but only partially.

Major uses:

- -optical fiber fabrication
- -films in microelectronics & MEMS
- -optical coatings
- -solar cells
- -a-Si and poly-Si for flat panel displays

## **CVD** schematically



byproduct desorption

## **Thermal CVD reactions**

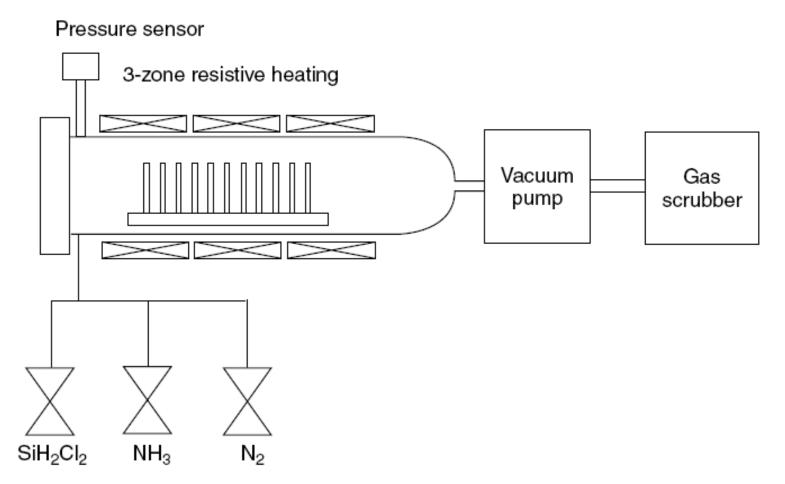
Gaseous precursor + surface reaction  $\rightarrow$  solid film + gaseous byproducts

pyrolysis reduction

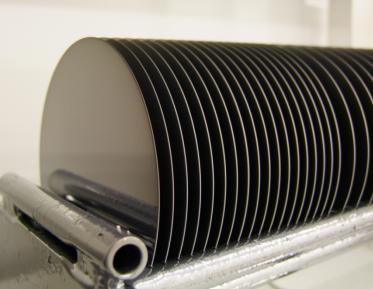
hydrolysis

compound formation  $\begin{array}{l} \text{SiH}_{4} (g) \rightarrow \text{Si} (s) + 2 \text{ H}_{2} (g) \\ \text{SiCl}_{4} (g) + 2 \text{ H}_{2} (g) \rightarrow \\ \text{Si} (s) + 4 \text{ HCl} (g) \\ \text{SiCl}_{4} (g) + 2 \text{ H}_{2} (g) + \text{O}_{2} (g) \rightarrow \\ \text{SiO}_{2} (s) + 4 \text{ HCl} (g) \\ \text{3 SiH}_{2}\text{Cl}_{2} (g) + 4 \text{ NH}_{3} (g) \rightarrow \\ \text{Si}_{3}\text{N}_{4} (s) + 6 \text{ H}_{2} (g) + 6 \text{ HCl} (g) \end{array}$ 

# Thermal CVD reactor: gaseous precursors, resistive heating





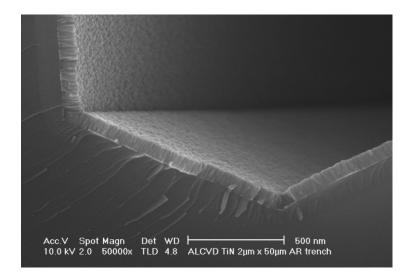


## Surface controlled reaction

Slow reaction rate (e.g. due to low temperature). Lots of gas available, and only a fraction of it has chance to react.

Because all surfaces are at same temperature, same deposition rate everywhere. Because surface controlled  $\rightarrow$  good step coverage.

ALD is a prime surface controlled reaction, excellent conformality.



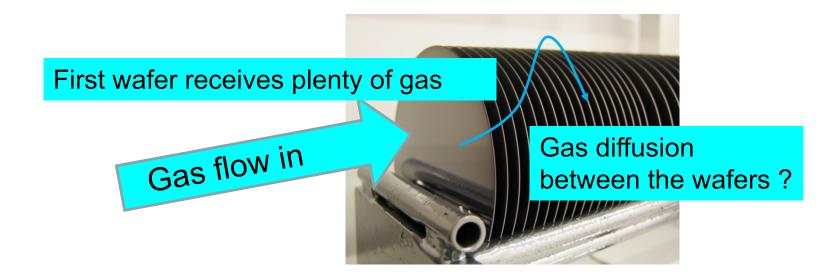
ALD TiN at the bottom of high aspect ratio groove.

# Mass transport limited reaction

Reaction rate is very fast at high temperatures (Arrhenius: rate is exponentially temperature-dependent).

All arriving gases react immediately  $\rightarrow$  need to ensure that gases arrive equally to all parts of reactor. If not, position dependent depo rate.

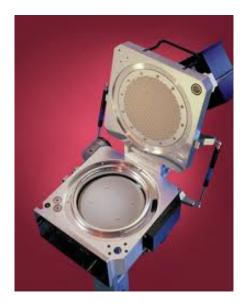
Reaction is in mass transport limited mode.



# Surface limited vs. mass transport limited reactions

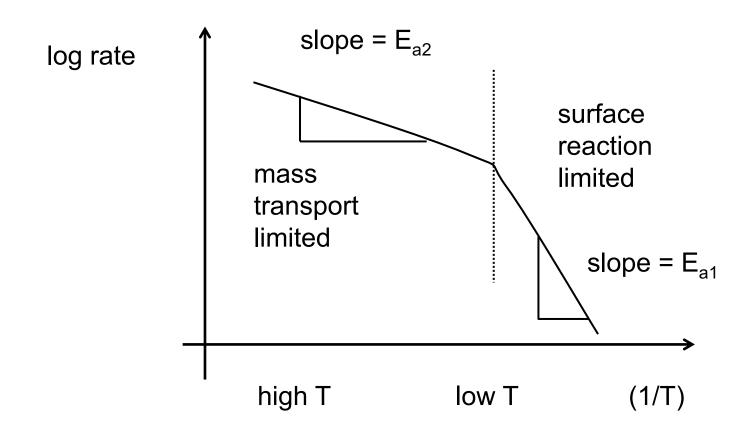


Surface reaction limited mode: -slow reaction rate -time for gases to diffuse -always extra gas available -can pack wafers tightly



A mass transport limited reactor: -all arriving gases react at once -therefore all wafers need to experience the same gas flow -easier to design uniform flow for single wafer reactors

# Surface limited vs. mass transport limited reactions



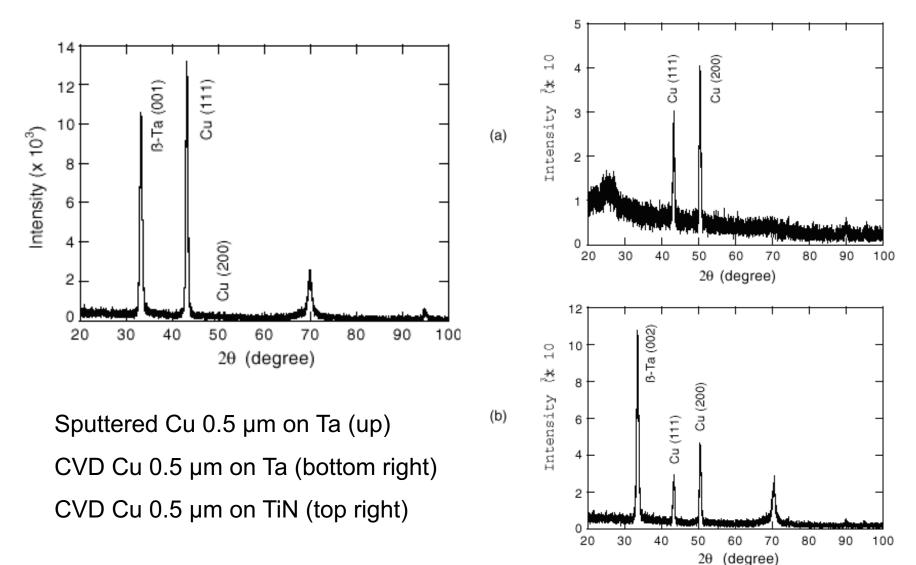
## **Common CVD reactions**

Material/method	Source gases	Temperature	Stability
LTO	$SiH_4 + O_2$	425 °C	Densifies
HTO	$SiCl_2H_2 + N_2O$	900 °C	Loses Cl
TEOS	$TEOS + O_2$	700 °C	Stable
PECVD OX	$SiH_4 + N_2O$	300 °C	Loses H
LPCVD poly	SiH <sub>4</sub>	620 °C	Grain growth
LPCVD a-Si	SiH <sub>4</sub>	570 °C	Crystallizes
LPCVD Si <sub>3</sub> N <sub>4</sub>	$SiH_2Cl_2 + NH_3$	800 °C	Stable
PECVD $SiN_x$	$SiH_4 + NH_3$	300 °C	Loses H
CVD-W	$WF_6 + SiH_4$	400 °C	Grain growth

 $LTO = Low-Temperature Oxide; HTO = High-Temperature Oxide; TEOS = TetraEthylOxySilane, Si(OC_2H_5)_4.$ 

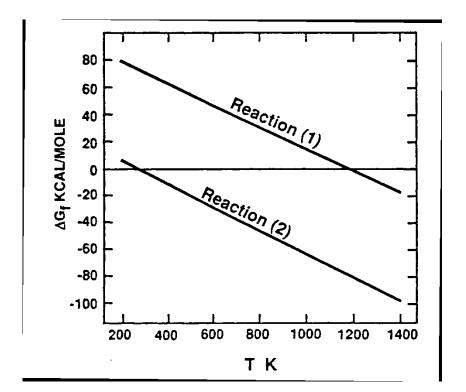
The precursor name TEOS has become synonymous with the resulting oxide film; it should be obvious which meaning is used.

## Copper: sputter vs. CVD



20

## Thermodynamics of CVD

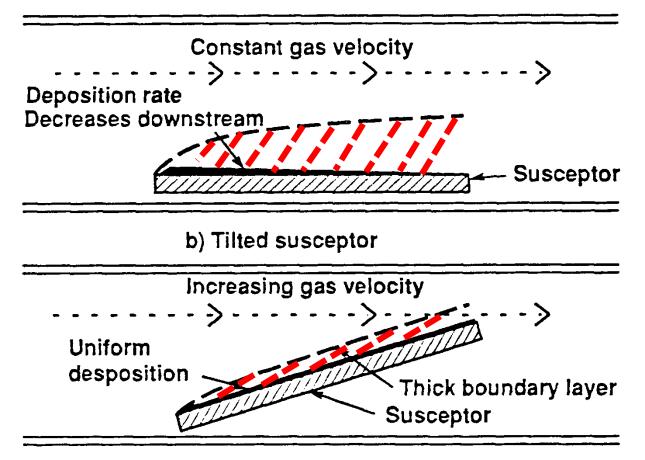


 $\Delta G < 0$  for reaction to take place

(1)  $\operatorname{TiCl}_4 + 2\operatorname{BCl}_3 \longrightarrow \operatorname{TiB}_2 + 10\operatorname{HCl}$ (2)  $\operatorname{TiCl}_4 + \operatorname{B}_2\operatorname{H}_6 \longrightarrow \operatorname{TiB}_2 + \operatorname{H}_2 + 4\operatorname{HCl}$ 

## Boundary layer = stagnant gas layer

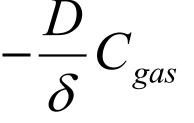
a) Horizontal susceptor



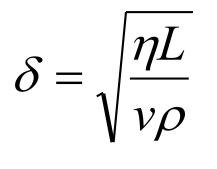
Pierson: Handbook of CVD, p. 36

## Rate modeling

 $J_{gas-to-surface} = -\frac{D}{S}C_{gas}$ 

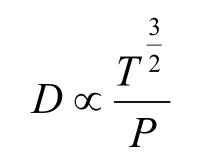


Diffusion of precursor gas from main flow to the surface.



Boundary layer thickness  $\delta$ .

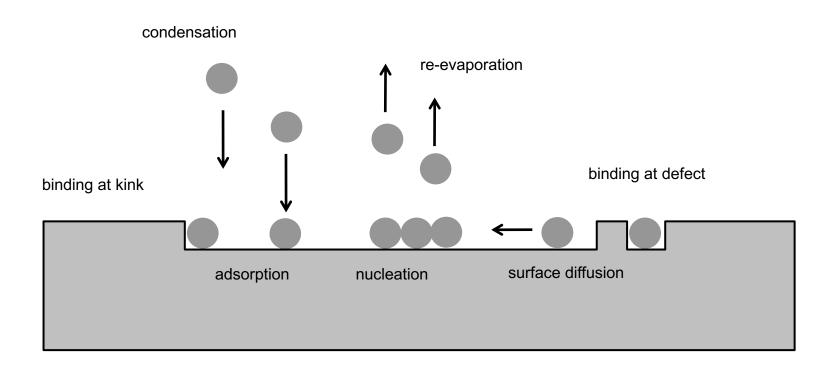
Raising temperature → density slightly increases and velocity increases



But lowering pressure, e.g. by 1000X → Diffusivity D increases 1000-fold.

 $\rightarrow$  J<sub>qas-to-surface</sub> increases dramatically

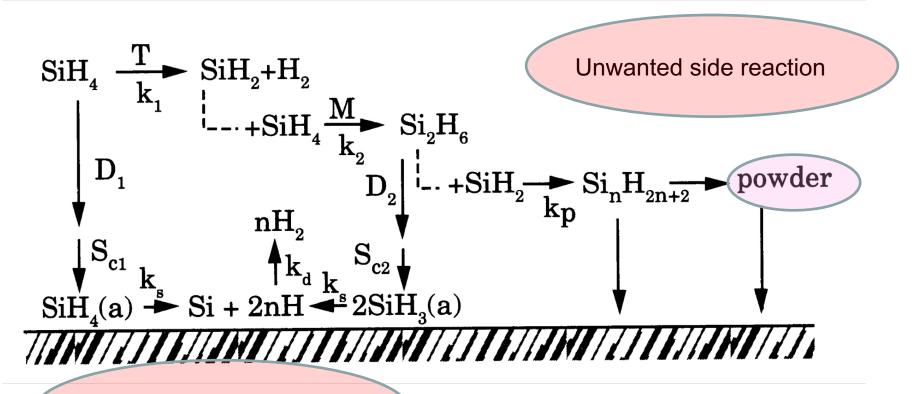
## Surface processes in deposition



## Adsorption processes

- $SiH_4(g) \rightarrow SiH_4(ad) \rightarrow Si(c) + 2H_2(g)$
- usual process: molecular adsorption
- Zn (g) + Se (a) → ZnSe (c)
- separate vapors adsorb strongly to the other specie
- passivation protects surface from reaction
- hydrogen typical passivation agent

# Silicon CVD from SiH<sub>4</sub>

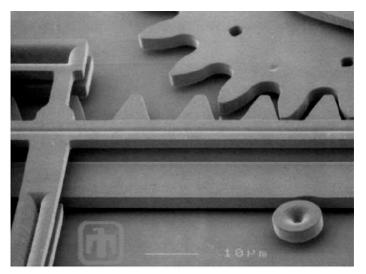


Wanted deposition reaction

Smith: Thin-film deposition

### Polysilicon

- SiH<sub>4</sub> (g) ==> Si (s) + 2 H<sub>2</sub> (g)
- Deposited by CVD at 625°C → true poly
- Can be deposited at 575°C → amorphous
- Anneal after deposition: a-Si → poly !
- Typical thickness 100 nm-2 μm



#### Structure and Properties of LPCVD Silicon Films

T. I. Kamins\*

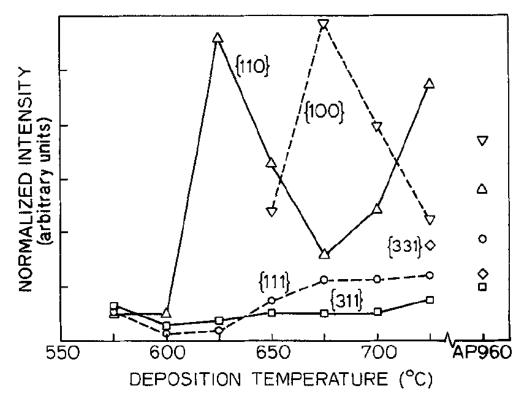
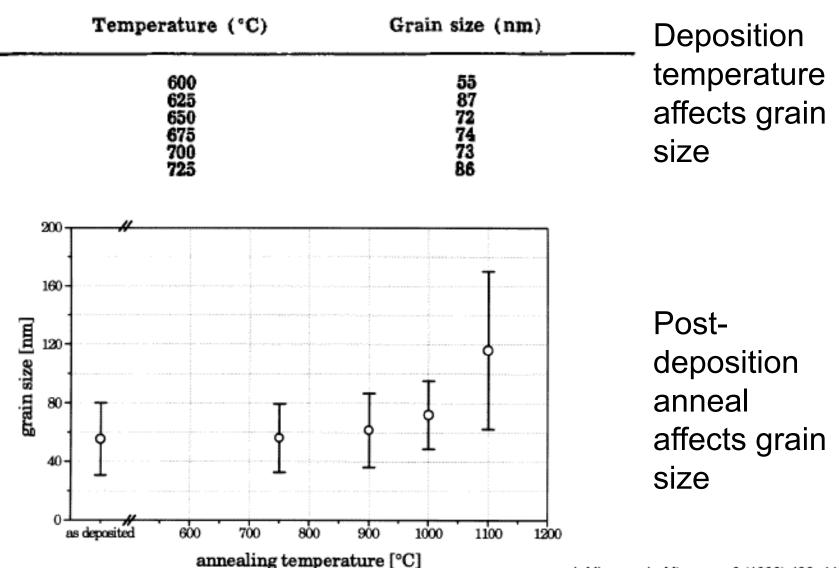


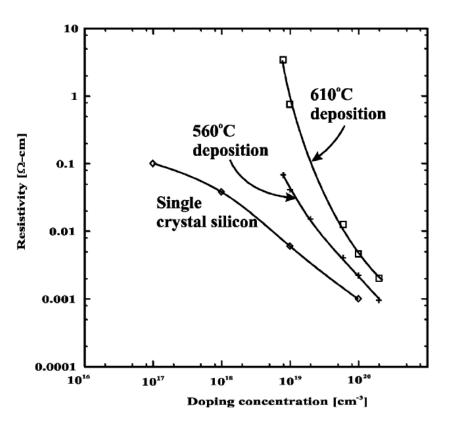
Fig. 1. X-ray texture as a function of deposition temperature for LPCVD silicon films and for an atmospheric pressure film.



#### Table I. Average grain size as a function of deposition temperature

J. Micromech. Microeng. 6 (1996) 436–446.

## Poly vs. <Si>



Poly resistivity always higher !

Density: same 2.3 g/cm3 Young's modulus: same 170 GPa CTE: same 2.5 ppm/K

Thermal conductivity:

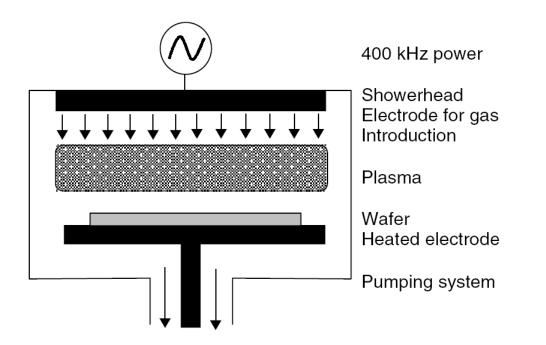
- <Si> 156 W/K\*m (at room temp)
- poly 32 W/K\*m

Carrier mobility: <Si> 100 cm<sup>2</sup>/Vs poly 10 cm<sup>2</sup>/Vs

# PECVD: Plasma Enhanced CVD

- Plasma aids in chemical reactions
- Can be done at low temperatures
- Wide deposition parameter range
- High rates (1-10 nm/s) (thermal 10% of this)

# PECVD @ 300°C: can be deposited on many materials



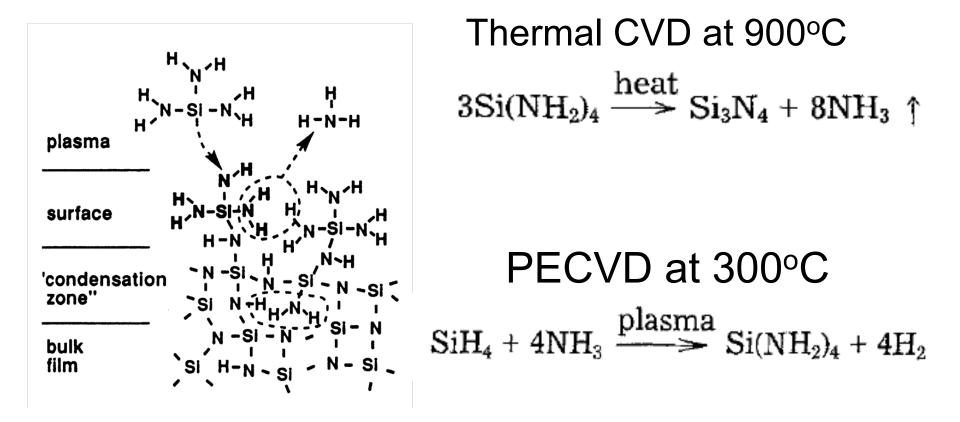
#### Oxide:

 $SiH_4$  (g) + N<sub>2</sub>O (g) ==>  $SiO_2$  + N<sub>2</sub> +  $2H_2$ 

#### Nitride:

 $3 \operatorname{SiH}_2\operatorname{Cl}_2(g) + 4 \operatorname{NH}_3(g) = > \operatorname{Si}_3\operatorname{N}_4(s) + 6 \operatorname{H}_2(g) + 6 \operatorname{HCl}(g)$ 

## SiN<sub>x</sub>:H: thermal vs. plasma



Smith: J.Electrochem.Soc. 137 (1990), p. 614

### Nitride thermal vs. PECVD (1)

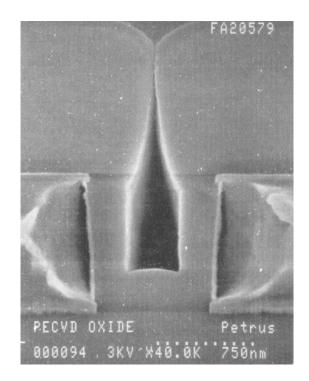
Property	High Temp. Nitride 900°C	Plasma Dep. Nitride 300°C
Composition	Si <sub>3</sub> N <sub>4</sub>	SiN <sub>x</sub>
Si/N Ratio	0.75	0.8 - 1.0
Solution Etch Rate Buffered HF 20-25°C 49% HF 23°C 85% H <sub>3</sub> PO₄ 155°C 85% H <sub>3</sub> PO₄ 180°C	10 - 15 Å/min 80 Å/min 15 Å/min 120 Å/min	200 - 300 Å/min 1500 - 3000 Å/min 100 - 200 Å/min 600 - 1000 Å/min
Plasma Etch Rate 82% $CF_4$ -8% $O_2$ , 700 W Na <sup>+</sup> Penetration IR Absorption	600 Å/min <100 Å	1000 Å/min <100 Å
Si-N max. SiH minor Density	~830 cm <sup>-1</sup> - 2.8 - 3.1 g/cm <sup>3</sup>	~830 cm- <sup>1</sup> 2,200 cm <sup>-1</sup> 2.5 - 2.8 g/cm <sup>3</sup>

### Nitride thermal vs. PECVD (2)

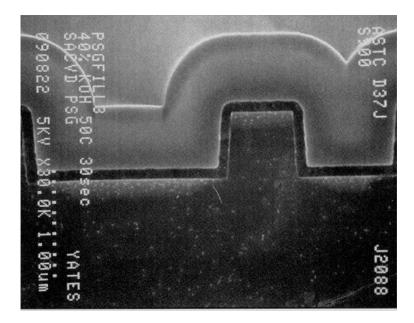
Property	High Temp. Nitride 900°C	Plasma Dep. Nitride 300°C
Refractive Index	2.0 - 2.1	2.0 - 2.1
Dielectric Constant	6 - 7	6 - 9
Dielectric Strength	1 x 10 <sup>7</sup> V/cm	6 x 10 <sup>6</sup> V/cm
Bulk Resistivity	10 <sup>15</sup> - 10 <sup>17</sup> Ω-cm	10 <sup>15</sup> Ω-cm
Surface Resistivity	>10 <sup>13</sup> Ω-cm	1 x 10 <sup>13</sup> Ω-cm
Intrinsic Stress	1.2 - 1.8 x 10 <sup>10</sup> dyn/cm <sup>2</sup>	1 - 8 x 10 <sup>9</sup> dyn/cm <sup>2</sup>
	Tensile	Compressive
Thermal Expansion	4 x 10 <sup>-6</sup> /°C	· •
Color, Transmitted	None	Yellow
Step Coverage	Good	Conformal
H <sub>2</sub> O Permeability	Zero	Low - None

## Half-time

## Step coverage in CVD



Quite OK, but the "keyhole" might be a minor problem (PECVD)

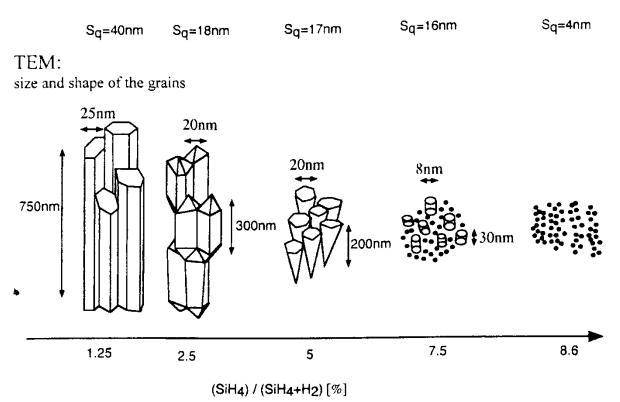


Conformal step coverage (thermal CVD processes)

Cote, D.R. et al: Low-temperature CVD processes and dielectrics, IBM J.Res.Dev. 39 (1995), p. 437

## Grain size & roughness

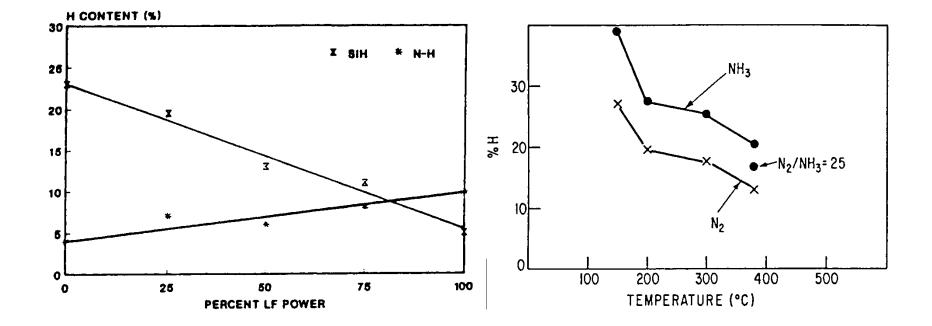
AFM: surface roughness



Vallat-Sauvain

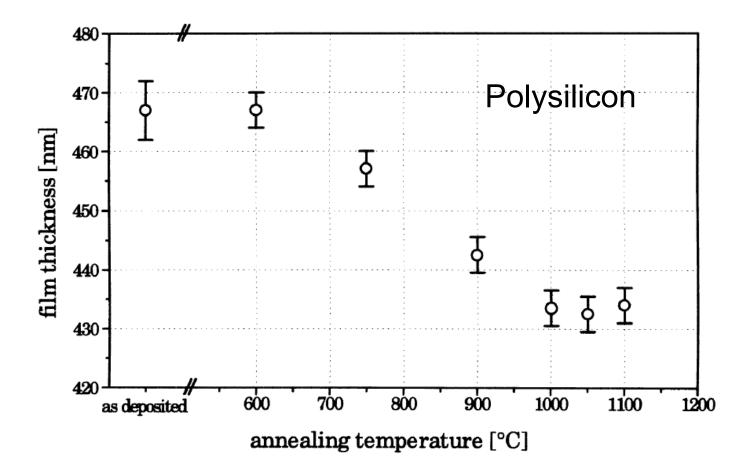
## Hydrogen in PECVD films

 $3SiH_2CI_2(g) + 4 NH_3(g) = > Si_3N_4(s) + 6 H_2(g) + 6 HCI(g)$ 



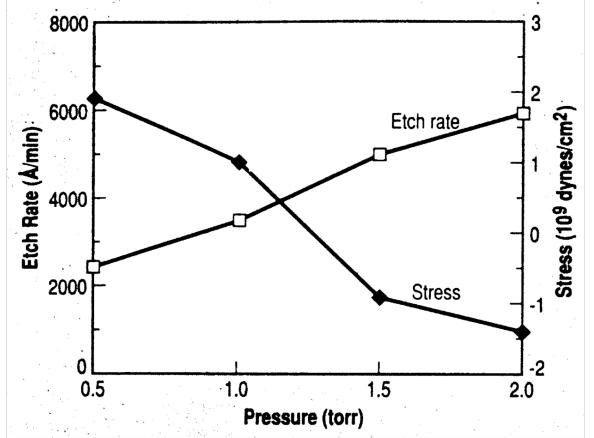
30 at% hydrogen is usual; 3-5% is as small as it gets

# CVD films lose thickness upon anneal (H<sub>2</sub> escapes)



J. Micromech. Microeng. 6 (1996) 436-446

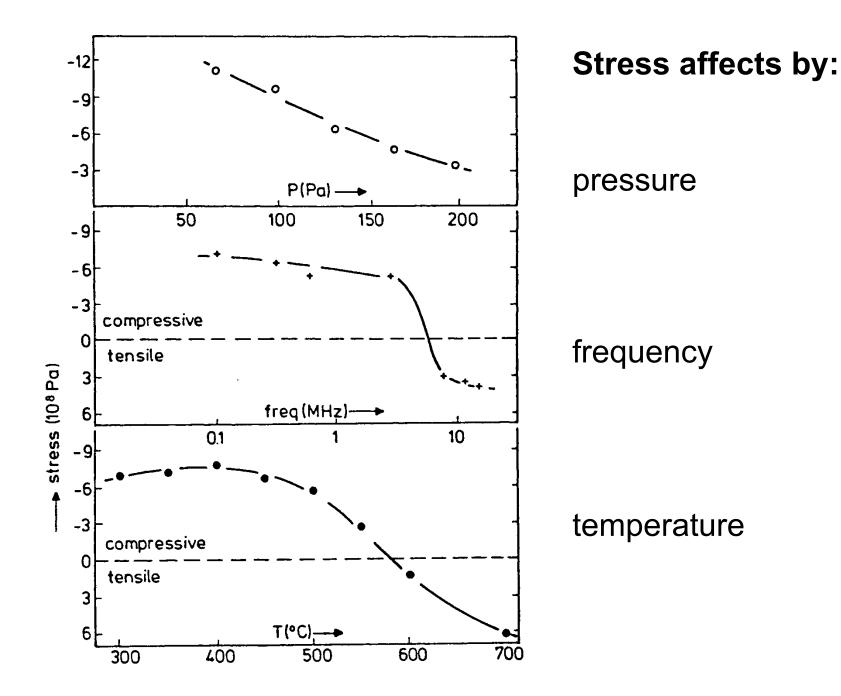
## Film quality: etch rate

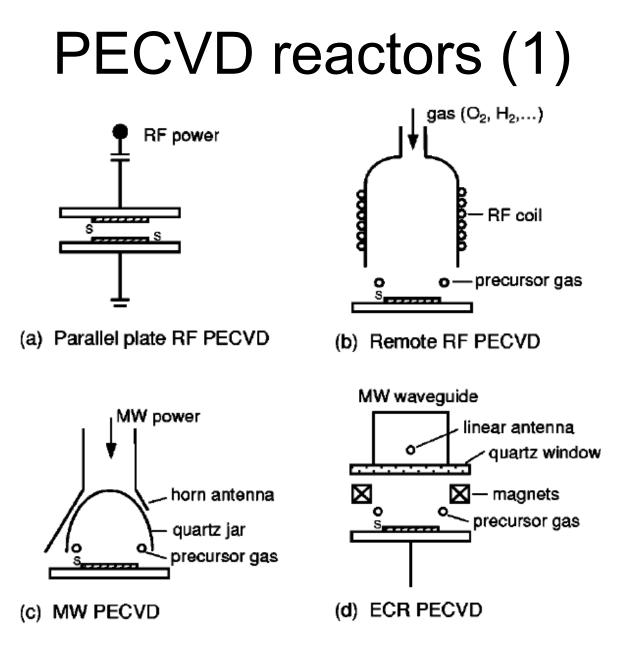


Low pressure equals more bombardment, and thus denser film, and compressive stress

With SiO<sub>2</sub>, BHF etch rate is a film quality measure (should be no more than 2X thermal oxide reference

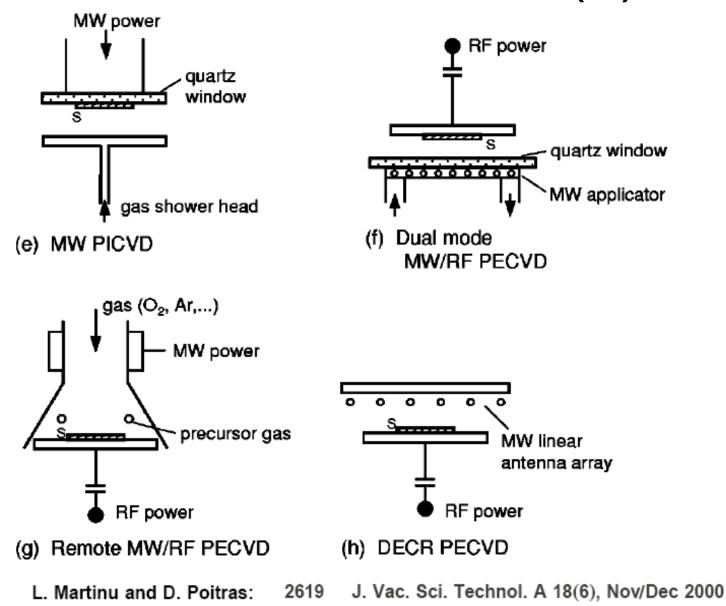
Hey et al: Solid State Technology April 1990, p. 139



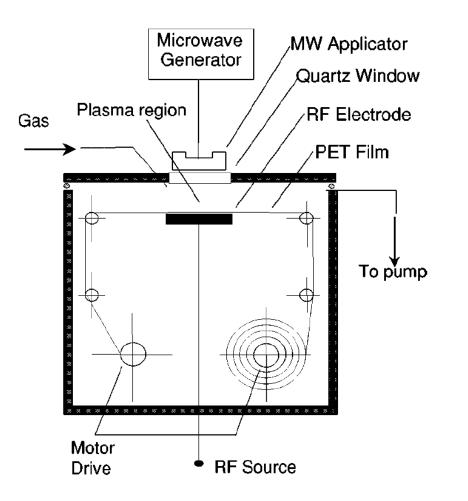


L. Martinu and D. Poitras: 2619 J. Vac. Sci. Technol. A 18(6), Nov/Dec 2000

## PECVD reactors (2)

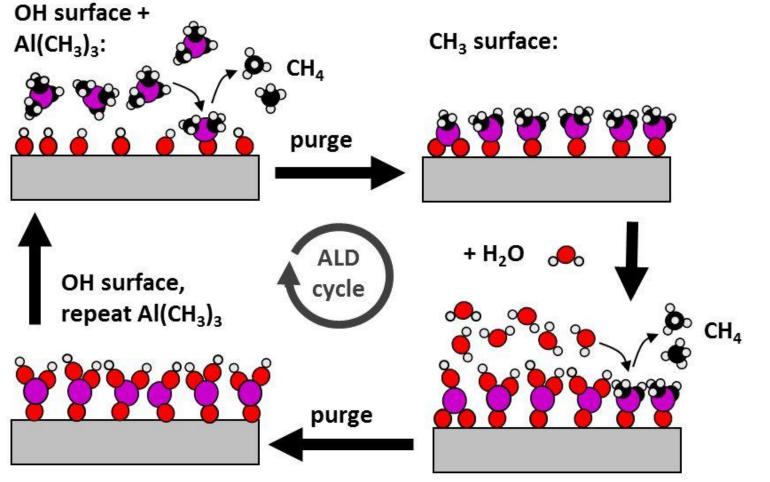


# Roll-to-roll PECVD



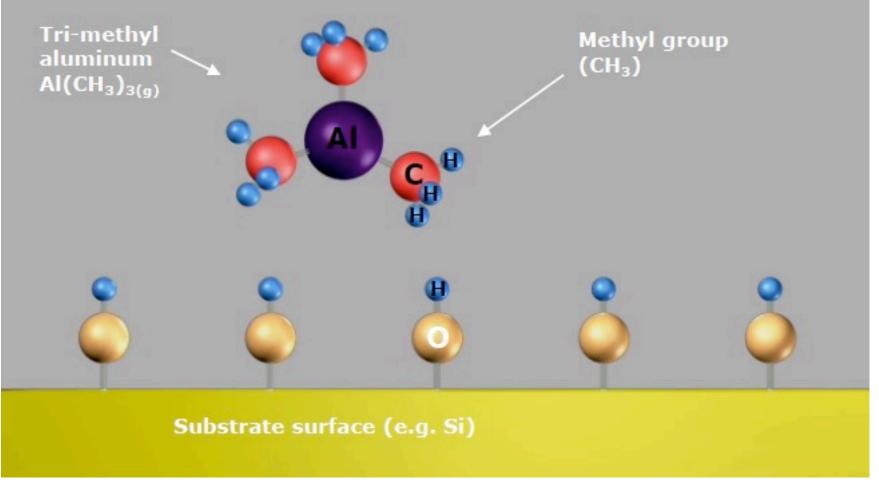
L. Martinu and D. Poitras

# ALD: Atomic Layer Deposition



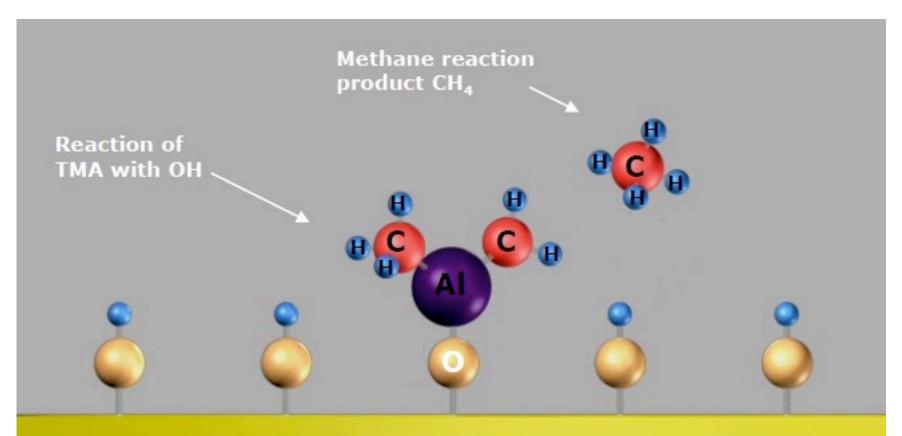
Precursors introduced in pulses, with purging inbetween

### Step 1: precursor 1 introduction



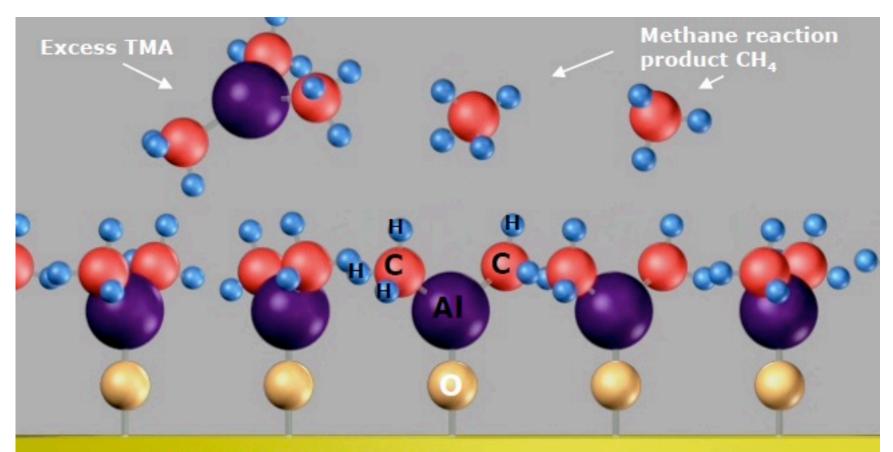
J Provine & Michelle Rincon, Stanford University 2012

### Reaction with hydroxyl moieties



Substrate surface (e.g. Si)

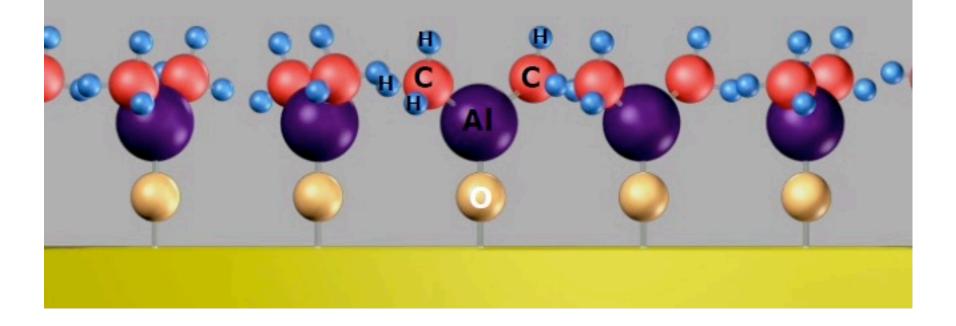
### Saturating reaction -> monolayer



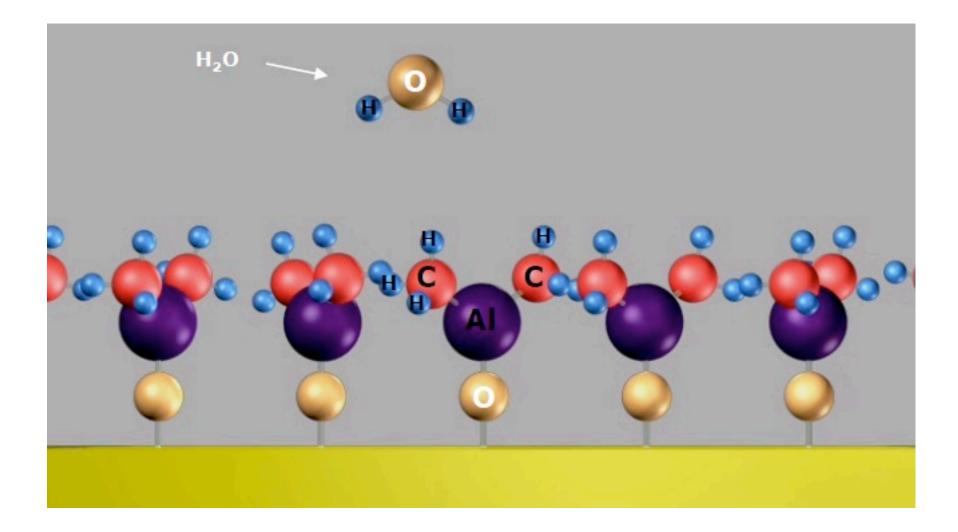
Substrate surface (e.g. Si)

# Step 2: purge

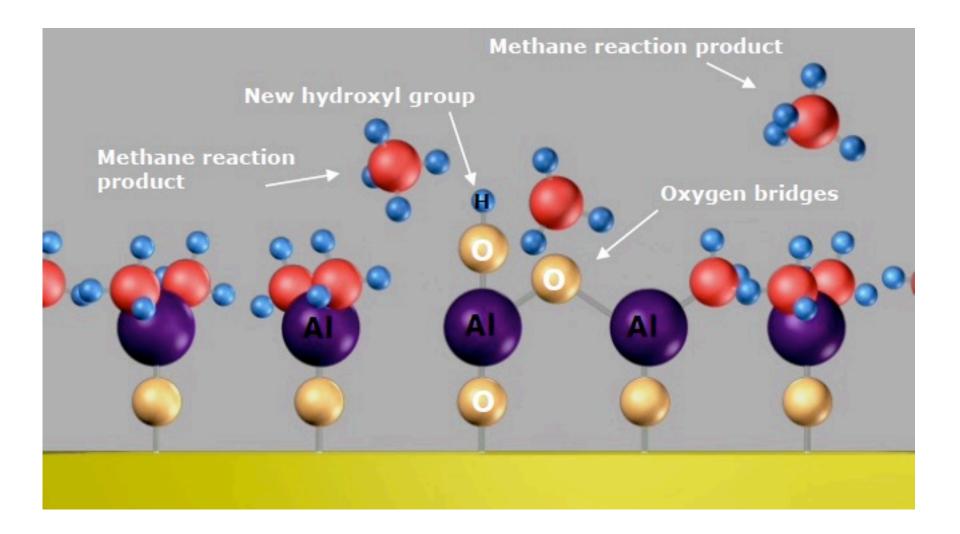
Flush away precursor molecules and reaction products



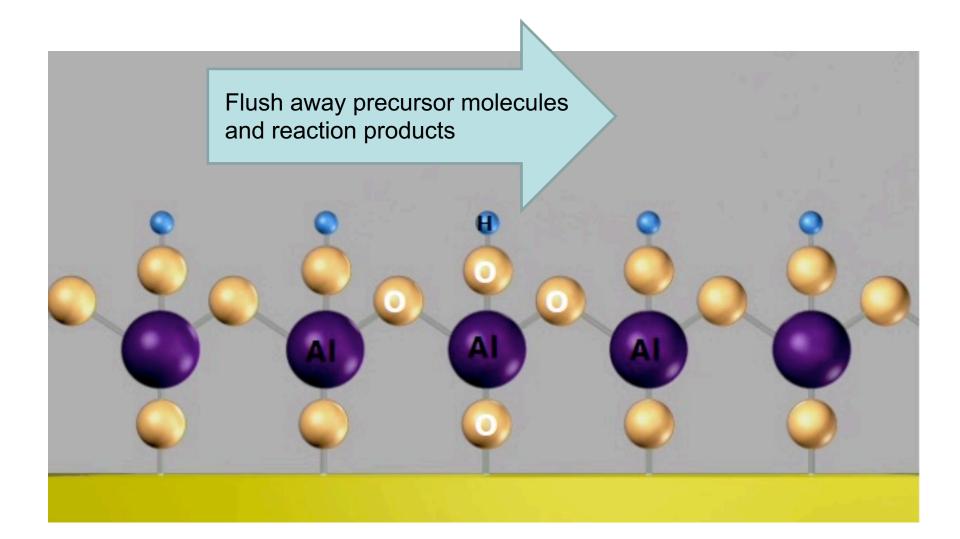
### Step 3: second precursor



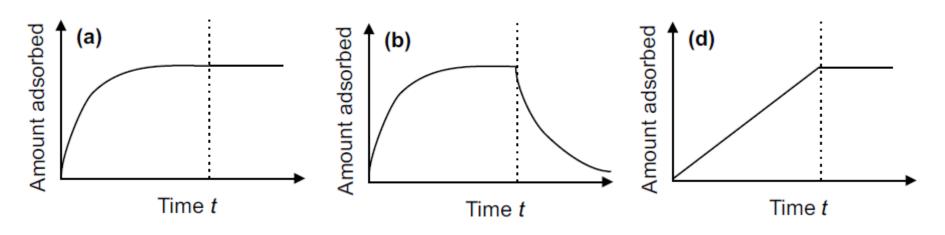
## Second half-reaction



# Step 4: purge again



### Surface saturation



Irreversible saturation ALD reactions:

Surface saturates with a monolayer of precursor, strong chemisorption (=chemical bonds formed) Reversible saturation:

Physisorption only (weak bonds like van der Waals): once precursor flux is stopped, surface specie will desorb. Irreversible nonsaturating.

CVD regime: more reactants in, more film is deposited (continuosly)

# ALD process based on:

- Chemisorption
  - Suitable temperature for chemical bonding, no thermal decomposition
  - Covalent bonding ⇒ excellent adhesion

Saturation

- Sufficient dosing of precursor material
- Self-terminating reactions ⇒ extremely precise dosing not required

### Surface controlled reactions

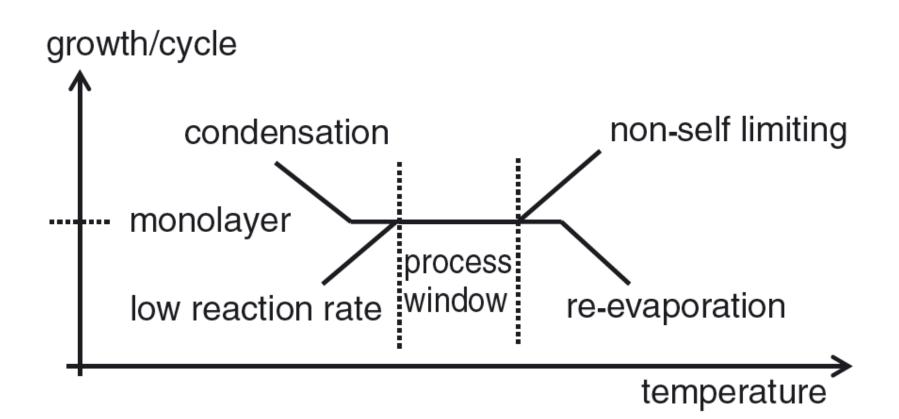
Film thickness is independent of substrate geometry 

 ⇒ conformal film
 onto deep trenches and 3D structures

Sequential

- Digital growth
- Sufficient purging needed between pulses
- Good flow dynamics required to ensure rapid gas changes

## ALD window



# **Deposition rate**

Basically one atomic layer per pulse

In practise less than an atomic layer because:

a) Inactive surface sites

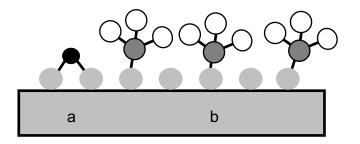
 b) Steric hindrance: a large precursor molecule prevents another precursor molecule from approaching the reactive site

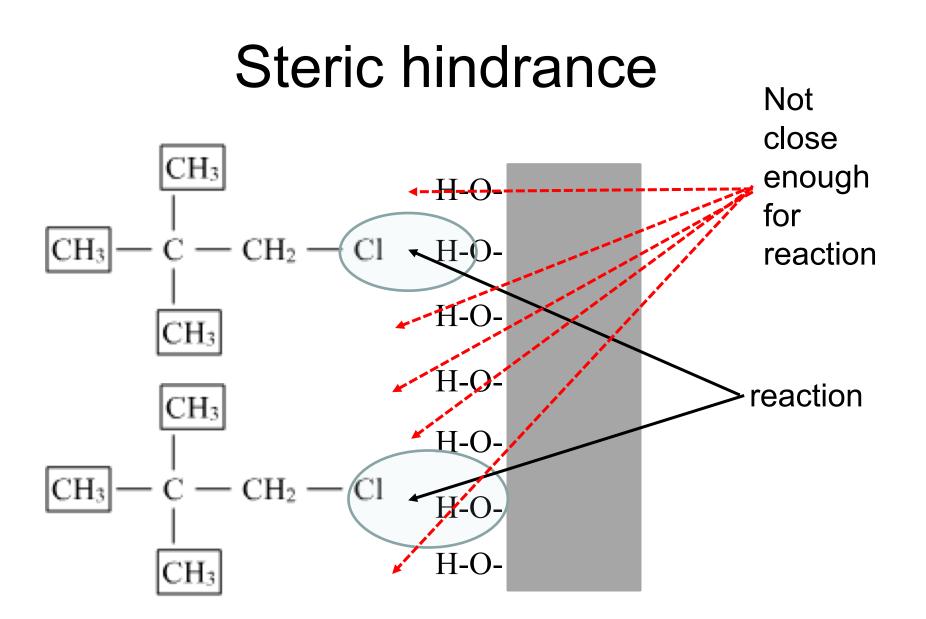
Al<sub>2</sub>O<sub>3</sub> deposition it is 1.1 Å/cycle (0.11 nm/cycle)

TiN it is 0.2 Å/cycle

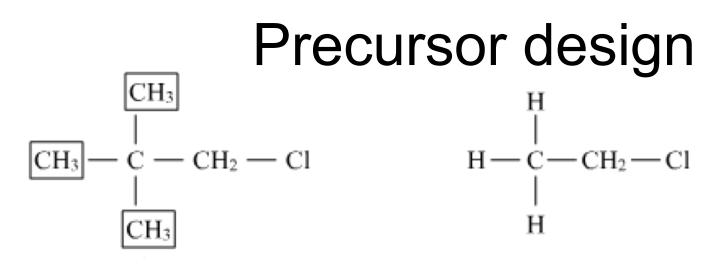
If pulses are one second  $\rightarrow$  15\*monolayer thickness/minute ~ 2 nm/min

If 0.1 second pulses → 20 nm/min max.





Most often: not an atomic layer, but less



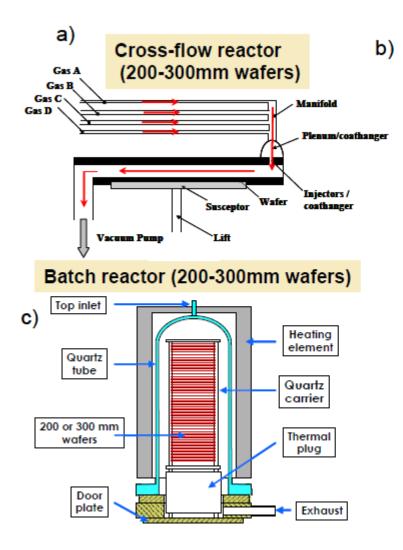
Large size, Steric hindrance

Small size, No steric hindrance.

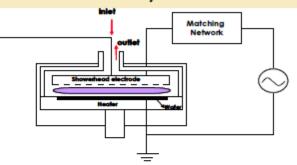
But you also need to consider:

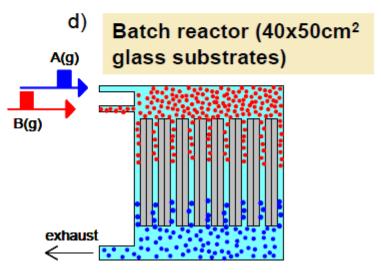
-thermal stability -vapour pressure -toxicity -price...

## ALD reactors



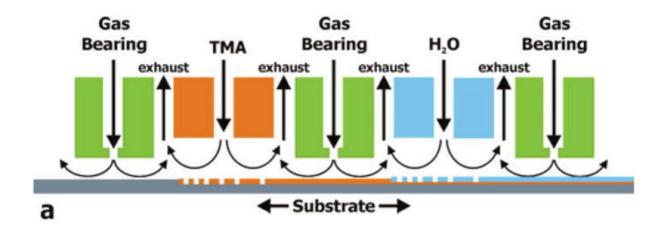
Showerhead ALD or PEALD reactor (200-300mm wafers)





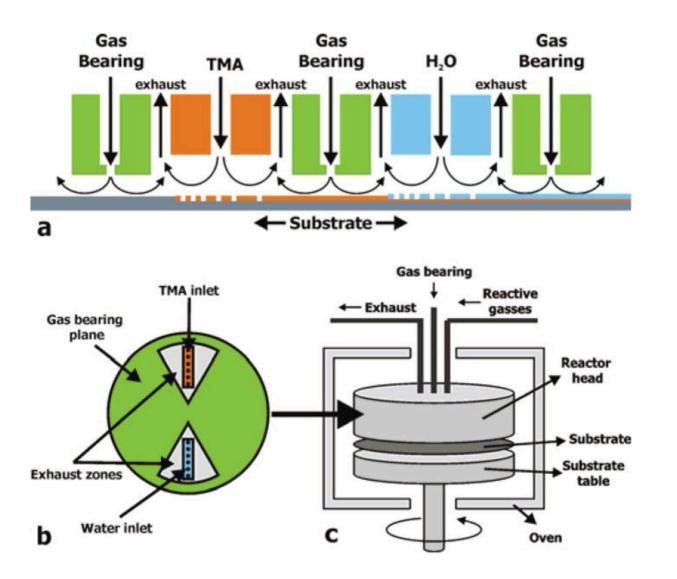
Suvi Haukka 2005

# **Spatial ALD**





# **Spatial ALD**



#### jes.ecsdl.org

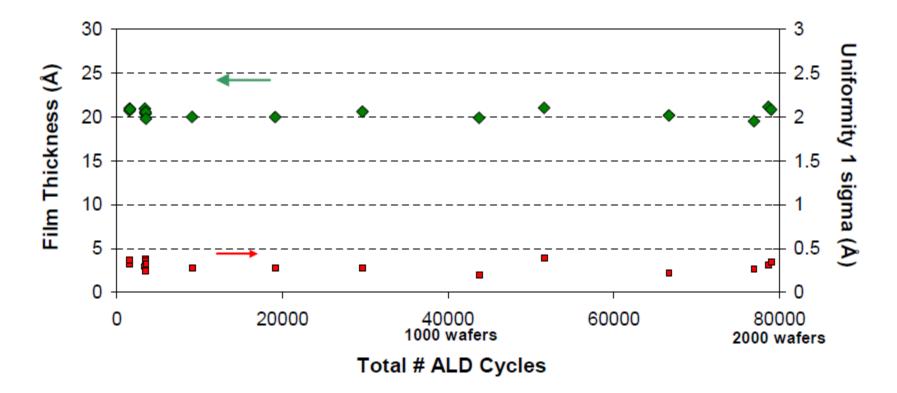
# **Spatial ALD reactors**



### Deposition upto 10 nm Al<sub>2</sub>O<sub>3</sub>

www.blog.baldengineering.com

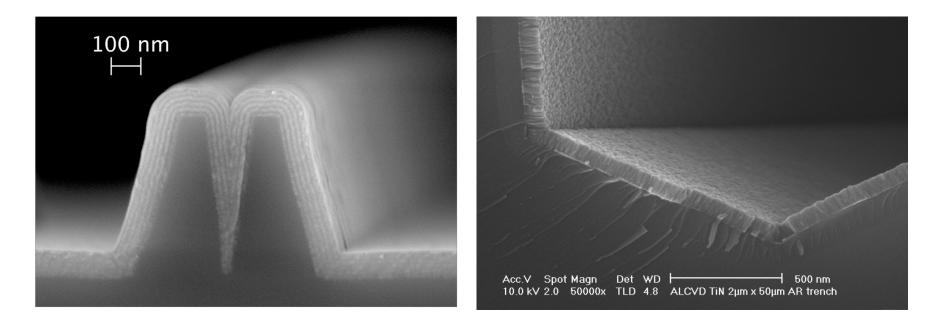
# ALD uniformity (=thickness across the wafer)



HfO<sub>2</sub> dielectric marathon test: 2000 wafers

# ALD conformality (=step coverage in microstructures)

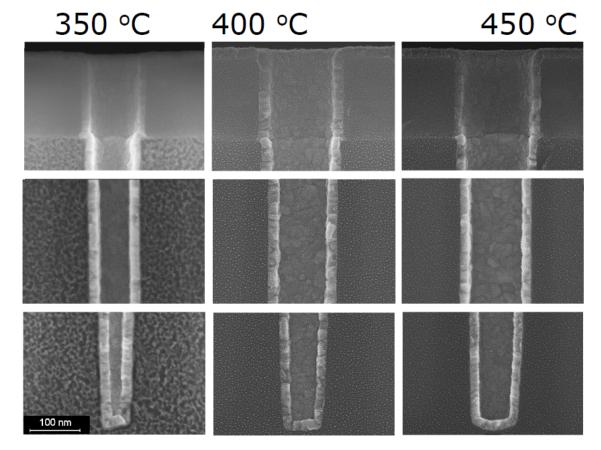
Excellent conformality: all surfaces coated by diffusing gaseous precursors in the surface reaction limited mode.



Al<sub>2</sub>O<sub>3</sub>/TiO<sub>2</sub> nanolaminate

#### **TiN** barrier

# Step coverage (2)

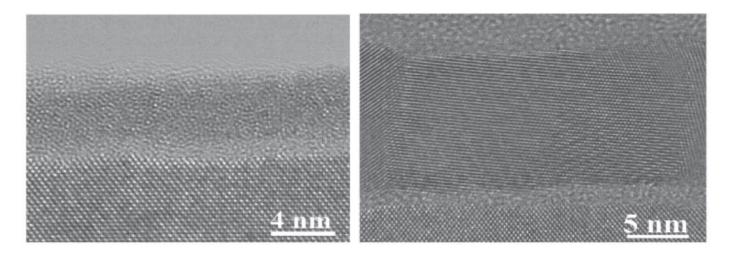


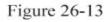
Step coverage good also in high aspect ratio grooves,

BUT pulse lenghts have to to be increased

(in coating porous materials, pulses last for minutes !!).

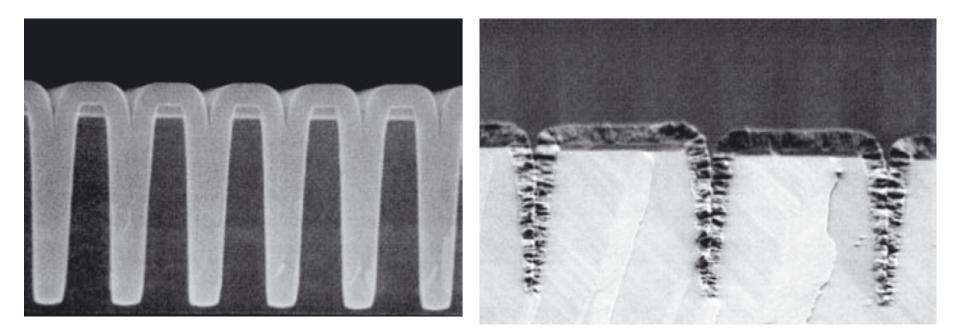
# Structure: amorphous vs. polycrystalline ?





ALD ZrO<sub>2</sub>: the 4 nm thick film is amorphous but the 12 nm thick film is polycrystalline. Reproduced from Kukli *et al.* (2007), copyright 2007, Elsevier.

# Crystallinity



amorphous aluminum oxide

Polycrystalline strontium titanate

Vehkamäki et al. (2001)

# Materials deposited by ALD

Nitrides	AlN TaN <sub>x</sub> NbN TiN MoN W <sub>x</sub> N ZrN HfN GaN InN
Carbides	TIC NbC TaC
Elements	Pt Ru Ir Pd Rh Cu Fe Mo Co Ni W
Sulfides	ZnS SrS CaS PbS
Fluorides	CaF <sub>2</sub> SrF <sub>2</sub> ZnF <sub>2</sub> MgF <sub>2</sub> LaF <sub>3</sub>
Oxides	Al <sub>2</sub> O <sub>3</sub> TiO <sub>2</sub> Ta <sub>2</sub> O <sub>5</sub> Nb <sub>2</sub> O <sub>5</sub> HfO <sub>2</sub> ZrO <sub>2</sub> SiO <sub>2</sub> ZnO MgO La <sub>2</sub> O <sub>3</sub> Y <sub>2</sub> O <sub>3</sub> Sc <sub>2</sub> O <sub>3</sub> Er <sub>2</sub> O <sub>3</sub> V <sub>2</sub> O <sub>5</sub> CeO <sub>2</sub> SnO <sub>2</sub>
	$Er_2O_3 V_2O_5 CeO_2 SnO_2 \dots$

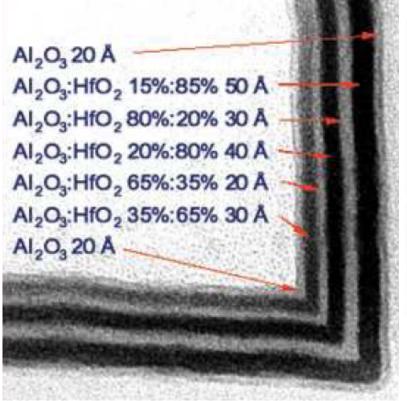
Important missing materials: •silicon

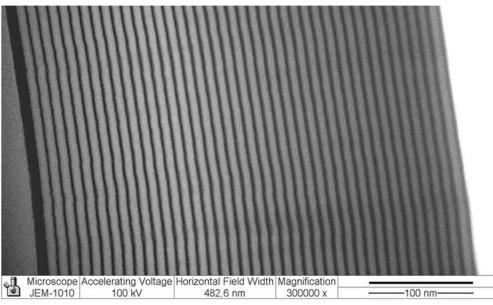
•silicon nitride

# ALD applications

- 1 nm thick catalysts (Pt, Pd)
- 2 nm thick TiN barrier layers underneath copper
- 6 nm thick CMOS gate oxides like HfO<sub>2</sub>
- 10 nm thick etch masks for plasma etching  $(Al_2O_3)$
- 30 nm thick antireflection coatings in solar cells  $(AI_2O_3)$
- 200 nm thick barrier layers in flat panel displays (Al<sub>2</sub>O<sub>3</sub>)

### New materials: nanolaminates

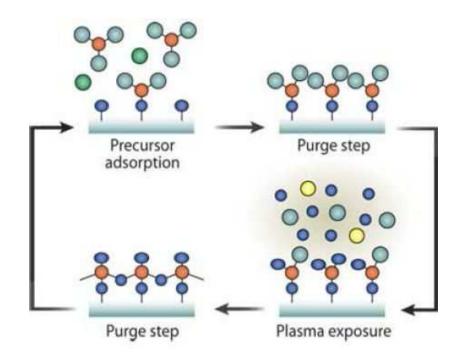


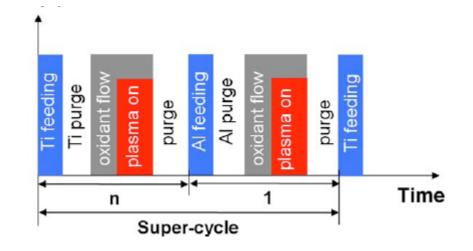


### Al2O3 and Ta2O5

Adriana Szeghalmi, Stephan Senz, Mario Bretschneider, Ulrich Gösele, and Mato Knez, APL 2009

# Plasma ALD (PEALD)





Gyu-Jin Choi, Seong Keun Kim,<sup>a</sup> Seok-Jun Won, Hyeong Joon Kim, and Cheol Seong Hwang<sup>\*,z</sup>

Journal of The Electrochemical Society, 156 (9) G138-G143 (2009)

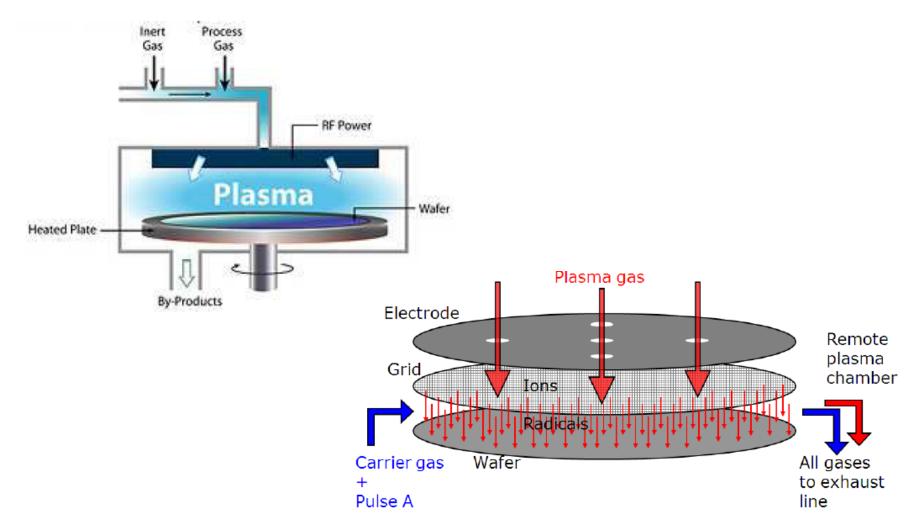
# Plasma ALD benefits

Plasma can break down precursors at lower temperature

New precursors become available because plasma can break down precursors that could not be used in thermal ALD

lons can kick of loosely bound specie from surface, densifying the film

## **PEALD** equipment



# ALD process development



with a new gate dielectric

- 2 1 years
  - ▲ Gamma phase: a production system 10000 wafers without interruption
    10-20 wafers/h
- 3 2 years

ŧ

Beta phase: Only one deposition process for each module

in an integrated system, production specs, 1000

consecutive wafers, <1% (1sigma or even 3 sigma)

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- 4 – 3 years
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Alpha phase: Integrated system 300mm, close to production system,

only one deposition process for each module, complete process flow, 250 wafers

(minimarathon), less than 10 particles < 0.075um, 2% (1 sigma)

#### - 6 – 5 years

<u>Concept and feasibility</u>: Scale-up of the precursor synthesis (50-100g), 1 to 3 different precursor chemicals, designed precursor vessel, process studies: 25 consecutive wafers (200-300mm), low particle counts, uniformity less than 5% (1 sigma), real device.

- 10 – 7 years

**Exploring phase:** Universities and research institutes, chemical suppliers: Development and synthesis of a number of new precursors and first ALD process studies on small substrates.

### PVD

Atoms as source material

Solid source materials

Vacuum/high vacuum

Elemental films mostly

Room temperature

Alloy films easily (W:N)

One process, many materials

### CVD & ALD

Molecules as source materials

Solid, liquid, gas precursors

Fluid dynamics important

Molecular/compound films mostly, Chemical bonds broken & formed

Needs elevated temperatures (or plasma activation)

Elements and compounds OK, alloys more difficult

any materials Each process materials specific

 $SiO_2$ ,  $Si_3N_4$ ,  $AI_2O_3$ ,  $HfO_2$ , ... Si, W

AI, Au, Cu, Pt, ...  $SiO_2$ 

# Summary

- Thermal CVD: excellent film quality
- PECVD: reasonable film quality at low T
- ALD: excellent film quality at low T
- Thermal CVD: high temperature needed
- PECVD: very high rate possible
- ALD: best for very thin films